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RESISTIVE MEASUREMENTS ON AN IMPROVED Nb-A1-Ge SUPERCONDUCTING RIBBON*

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An improved version of the Nb-Al-Ge ribbon which was developed in our laboratory¹ has been produced. This ribbon is considerably less porous and more flexible than its predecessor. Resistive measurements at 4.2° - 16° K in steady transverse magnetic fields up to 150 kOe are presented. Preliminary results on other Nb₃Al-based compounds in ribbon form will also be presented.

Introduction

Because Nb₃Al and Nb₃Al-based ternary materials have been shown to be capable in principle of carrying large currents at temperatures above 12° K and fields up to 15T $^{1-3}$, we have directed some recent efforts 1 , 4 , 5 at producing ribbons with desirable technical properties. This task has been difficult, and improvement comes in a series of small steps, one of which is reported here.

Experimental

The basic fabrication steps are as reported previously¹, with the exception that annealing was performed in argon gas at 1/2 atmospheres. Compositions used were Nb3Al, Nb3Al.75Ge.25, Nb3Al.9Sf.1 and Nb3Al.7Si.3. The large amounts of porosity noted in the previous work¹ were not present, and although quantitative measurements of bendability were not made, there was an obvious increase in flexibility and resilience. Measurements of J_c vs. H were made in transverse fields in a manner identical to our previous work¹, at the Francis Bitter National Magnet Laboratory.

Results Figures 1 and 2 give the results for Nb3(A1,S1)



Figure 1 J_c vs. H for Nb3(Al.9Si.1) ribbon; H transverse to J, parallel to plane of ribbon; at 4.2°, 12° and 16°, in He gas.



Figure 2 J_c vs. H for Nb3(Al.7S1.3) ribbon; H transverse to J, parallel to plane of ribbon; at 4.2°, 12° and 16°, in He gas.

alloys. As can be seen the critical current drops off rapidly. Figure 3 shows the Nb₃Al_75Ge.25



Figure 3 J_c vs. H for Nb3Al.75Ge.25 ribbon; H as in Figs. 1 and 2; at 4.2°, 12° and 15° in He gas and 4.2° in He liquid. Note that the liquid environment increases J_c by ca. 30%.

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sample. This sample differed from that previously reported1 only in the heat treatment under one half atmosphere pressure and subsequent reduced porosity. The critical current at low fields is similar to the previous Nb3A1,75Ge.25 ribbons but greatly reduced at 4.2°K at higher fields (>10T); it is possible that the J_c peak has simply moved to fields well in excess of 15T, out of the range of our measurements. All samples were tested in a helium gas environment except the dashed line of 4.2°K in Figure 3. In this case the sample, after testing at 4.2, 12 and 15°K in He gas, was uncovered and replaced without change in liquid helium at 4.2°K. As can be seen, the improved heat transfer has raised $J_{\rm c}$ by about 30%. Figure 4 shows the results for Nb3Al. In this instance, J_C was found to be the highest of all samples at all fields. In the sample heat treated 24 hours at 750°C $J_{c}(4.2^{\circ}K)$ is $>10^4$ A/cm² at all fields to 15T than any other Nb3Al alloy tested.



Figure 4 J_c vs. H for Nb3Al ribbon; H as in Figures 1 and 2; at 4.2°K and 12°K for ribbon without 750°C ordering anneal, and 4.2°K with 24 hour anneal at 750°C.

Since longer times *up to 48 hours) at 750°C will improve the transition temperature even further⁵ it would be worthwhile to test such samples in the future. Not only would they have higher J_c but a higher T_c also, which is opposite to the Nb3(Al.75Ge.25) samples previously reported¹, where higher T_c meant lower J_c .

In conclusion, it should be pointed out that while these ribbons have very interesting properties up to 12 to 14°K, the critical currents are still too small for practical application. Further Nb3Al alloys are no more ductile than Nb3Sn which presents serious engineering disadvantages. However, Nb3Al is not any worse than Nb3Sn in so far as mechanical properties are concerned and provided that its critical current properties can be improved, it could one day become the successor to Nb3Sn until something better than both is found.

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References

- R. Lohberg, T. W. Eagar, I. M. Puffer and R. M. Rose, Appl. Phys. Lett. 22, 69 (1973).
- M. A. Janacko, J. R. Gavaler and C. K. Jones, J. Vacuum Sci. Tech. <u>9</u>, 341 (1971).
- R. H. Hammond, K. M. Ralls, C. H. Meyer, Jr., D. P. Snowden, G. M. Kelly and J. H. Perene, Jr., J. Appl. Phys. 43, 2407 (1972).
- T. W. Eagar, Bachelor's Thesis, Department of Metallurgy and Materials Science, M.I.T., Cambridge, Mass. 02139, January, 1972.
- J. G. Kohr, T. W. Eagar and R. M. Rose, Met. Trans. 3, 1177 (1972).