# **EVOLUTION OF THE LHC BEAM SCREEN SURFACE CONDITIONING UPON ELECTRON IRRADIATION**

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### Abstract

For the vacuum scientists and the accelerator community, finding solutions to mitigate pressure rises induced by electron, photon and ion desorption, and also beam instabilities induced by ion and electron clouds is a major issue. Moreover, it is worth noting that the OFE copper beam screen of the LHC are initially cleaned with standard industrial processes, leading to a residual chemical contamination. Along the time, changes in the surface chemistry of vacuum chambers are observed during beam operations, leading to modifications of: outgassing rates, stimulated desorption processes and secondary emission yields (SEY). The impact of ions on molecule desorption and electron production was investigated to identify their influence on the global pressure rises and to quantify the ion conditioning effect on copper surfaces: (i) SEY evolution was measured to understand the changes of surface conditioning upon particle irradiation; (ii) surface chemistry evolution after electron irradiation was investigated by XPS and TOF-SIMS analyses using the both ANDROMEDE facility at IJCLab. Finally, the relationship between the surface chemistry and the conditioning phenomenon will be discussed.

#### **INTRODUCTION**

Electron emission is a major phenomenon involved in the formation of the electron-cloud (EC) in the LHC. EC effects have been identified among the major performance limitations for the Large Hadron Collider (LHC). The EC is induced by the electron multipacting process related to the secondary electron yield (SEY) of the surfaces exposed to electron irradiation. The SEY depends strongly on the nature of materials and on the surface chemistry, and consequently the EC formation is determined by the surface properties. Nevertheless, electron bombardment of the inner surface of the beam pipes is also responsible for a beneficial effect called "surface conditioning" (or the "scrubbing effect") leading to a reduction of SEY and therefore inducing a mitigation of the multipacting process and electron cloud build up.

The aim of this study was first to validate the set-up developed in our laboratory and devoted to SEY measurements, by reproducing the results that exist elsewhere. Then, we proposed an alternative method for the analysis of the surface chemistry to the X-ray Photoelectron Spec-

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troscopy which is traditionally used: Time-of-Flight Secondary Ion Mass Spectrometry with high energy gold nanoparticles to probe the surface. We investigated the conditioning of copper beam screen induced by electron bombardment and identified the role played by the modifications of the surface chemistry in this phenomenon.

#### **EXPERIMENTAL DESCRIPTION**

The experimental set up for SEY measurements is an inhouse build set-up developed at IJCLab. It consists of a single UHV chamber (base pressure:  $7x10^{-10}$  mbar) equipped with an electron gun providing a pulsed electron beam (with a pulse length of 30 ms) in the energy range 10 to 1500 eV, with an intensity from few nanoamperes to 10  $\mu$ A. The sample is carried by a single manipulator allowing for a precise positioning of the sample in the chamber.

The SEY was measured by the sample bias method that is a two-step SEY measurement. First, the primary current  $I_p$  is acquired for each primary electron energy E by applying a positive sample bias (V = +50 V). In this case, the current measured on the sample is  $I_M(+50V) = I_P$ . The emitted secondary electrons (SE) are trapped and recaptured by the sample. The sample polarity is then switched to a negative value (V = -20 V) and I<sub>M</sub> is acquired while shooting with the electron gun on the sample with the same energy settings as during I<sub>p</sub> acquisition. The SE current (I<sub>SE</sub>) is given by:

$$I_{SE} = I_p - I_M(-20V).$$
(1)

The SEY  $\delta$  is then obtained by:

$$\delta = \frac{I_{SE}}{I_p} = \frac{I_p - I_M(-20V)}{I_p} = 1 - \frac{I_M(-20V)}{I_M(+50V)}.$$
 (2)

In the present work, we used the ANDROMEDE facility [1] to analyse samples of copper beam screen. 12-MeV Au<sub>400</sub><sup>4+</sup> ion beams are accelerated by a NEC Pelletron® 4MV electrostatic accelerator to bombard samples. The emitted secondary ions (both positive and negative ions) are analysed with the ToF spectrometer EVE In this set-up, it is possible to record a mass spectrum with only one single ion impact. Consequently, the secondary ion emission yield is strongly enhanced and the sensitivity of chemical surface analysis is improved. A high detection efficiency for molecules deposited in very small quantities on metal surfaces (lower than the monolayer) is reached. Moreover, the limited number of primary ions (about 107 nanoparticles/cm2) required to analysis the samples prevents excessive damage of the surface. Only the extreme surface (i.e.

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the molecules deposited on the surface and the materials surface corresponding to a depth of  $\approx 10$  nm) is analysed.

The experimental system used for XPS analysis in the present work is the K-ALPHA system from Thermo-Fisher, equipped with a monochromatic Al ka X-ray source (hv = 1486.6 eV) and an Ar<sup>+</sup> ion gun for XPS sputter depth profiling. For each analysis, a full energy spectrum is acquired for identification of the constituent elements and additional sweeps are then performed in narrower energy regions to obtain high energy-resolution spectra. Spectra displayed in the following in normalized intensity were obtained after normalization of the signal to its maximum intensity.

# SECONDARY ELECTRON YIELD **EVOLUTION**

The evolution of the secondary electron yield of an asreceived beam screen sample during its irradiation at room temperature by 500 eV electrons is shown in Fig. 1. SEY is observed to decrease over the full considered energy range when the irradiation dose increases. The maximum SEY, initially equal to 2.67, decreases down to 1.13 after a dose of 4.8 10<sup>-2</sup> C/mm<sup>2</sup>. For larger irradiation doses, no further evolution of SEY is observed. These results are in good agreement with those previously published in the literature: the initial value of the SEY for an as-received copper is reported to be between 2.5 and 2.1 and for a conditioned sample the SEY decreases down to a value between 1.2 and 1.1 [2-4].



Figure 1: Secondary electron yield curves as a function of primary electron energy, for an as-received copper beam screen conditioned with a 500 eV electron beam at different irradiation doses.

# SURFACE CHEMISTRY EVOLUTION

In the as-received state, the C 1s line shows a main peak at 284.8 eV ascribed to C-C bonds (sp3 hybridization). A second peak at 286 eV corresponding to C-O bonds, and a third at 288.2 eV related to O-C=O groups are visible (Fig. 2). These latter are due to an adventitious carbon contamination and comes from the different compounds forming the adsorbed laver. In the fully conditioning state (Fig. 2 (b)), the XPS analysis indicates that the decrease of the secondary electron yield corresponds to significant modifications of the surface chemistry. The disappearing of the peak at 288.2 eV on the C1s line implies the removal of O-C=O groups. The main C 1s peak is shifted towards 284.8 eV, indicating the transformation of the adventitious carbon layer (sp3) into a more graphitic form (sp2), as already observed in previous studies, for instance in [2]. This phenomenon is called sometimes "graphitization". Figure 2 (c) shows the XPS analysis of a carbon coating layer provided by the CERN: we found the main C 1s peak related to the sp2 hybridization.

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Figure 2: SEY curves and C1s XPS spectra measured on a LHC Cu beam screen sample: (a) "as received"; (b) fully conditioned; (c) on an as received amorphous carbon coating.

From the analysis of the negative secondary ion spectra, a significant increase of carbon (Cn<sup>-</sup>) and carbon cluster (C<sub>n</sub>H<sup>-</sup>) emission for the conditioned sample is observed (Fig. 3). Comparison of the conditioned sample spectrum with a graphene spectrum obtained under the same conditions suggests that a film of hydrogenated graphene covers the irradiated surface as the carbon cluster distribution is similar. Thus, in agreement with the XPS results, the ToF-SIMS analysis shows that a carbon layer whose bonds are that of graphene (sp2 hybridization) is formed on copper surface under electron bombardment. It is worth noting that this carbon layer is not composed only of carbon, but it contains also high amounts of hydrogen and oxygen.

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Figure 3: Negative secondary ion mass spectrum for a copper beam screen after electron conditioning process displaying intense carbon cluster peaks. (top); Negative secondary ion mass spectrum for a 6-8 monolayer graphene foil (bottom); in both cases, the samples were bombarded by 12 MeV  $Au_{400}^{4+}$  nanoparticles and the recorded spectra correspond to a dose of  $3 \times 10^7$  nanoparticles/cm<sup>2</sup>.

#### **CONCLUSION**

In conclusion, the investigation of the copper beam screen conditioning under irradiation with 500 eV electrons evidenced two processes: (i) a surface cleaning by electron stimulated desorption; (ii) the formation of a graphene-like layer. Both processes lead to the decrease of the copper SEY down to value close to unity. Complementary surface analysis (Tof SIMS and XPS) showed that this thin carbon layer (less than 0.5 nm) contained high amount of hydrogen and oxygen. Further investigations are needed to understand why such a thin graphite layer can have such a significant effect.

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