

# In-situ observation of phase transformation during ion beam irradiation

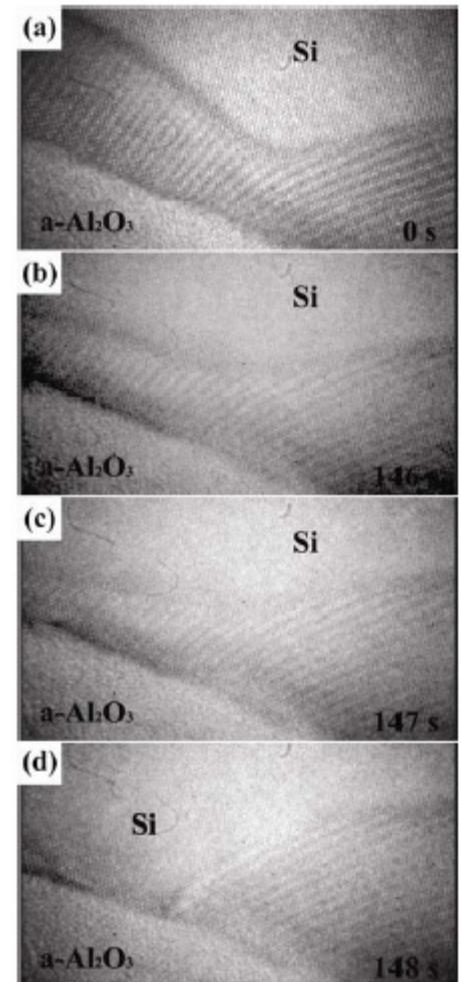
Jonghan Won<sup>1</sup>, James A. Valdez<sup>2</sup>

<sup>1</sup>Advanced Nano Surface Team, Korea Basic Science Institute, KOREA

<sup>2</sup>MST-8, Los Alamos National Laboratory, USA

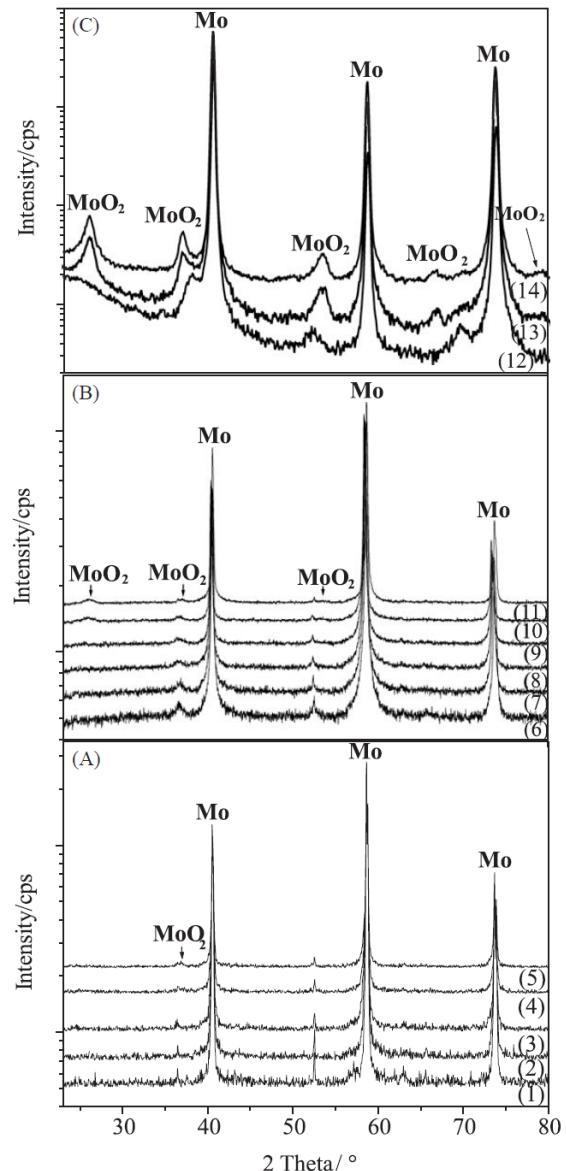
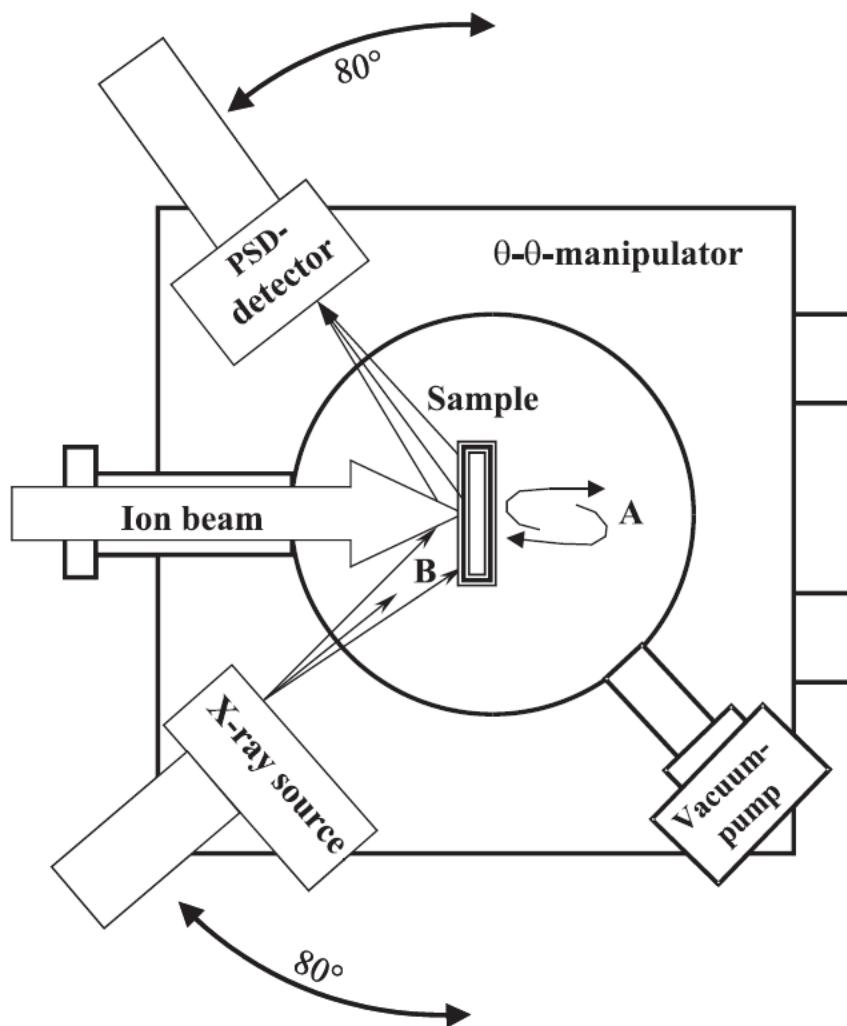
## ► What is In-situ observation ?

1. Real-time data acquisition  
- providing a “live” image or graph
2. Keeping the exact same position during observation
3. Saving time and effort



\* J.Won, Doctoral dissertation

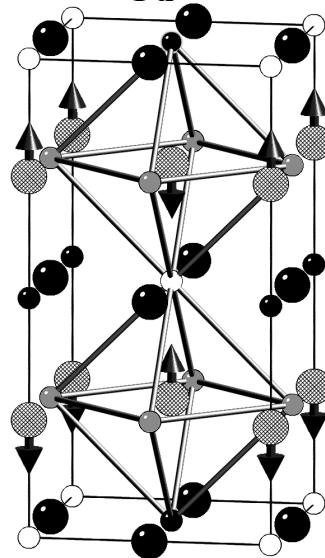
# In-situ observation during ion beam irradiation (XRD)



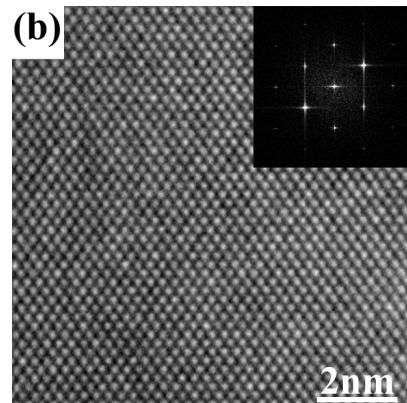
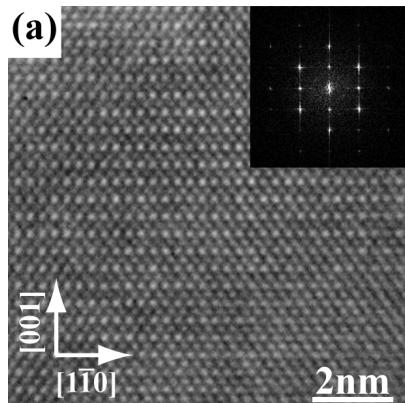
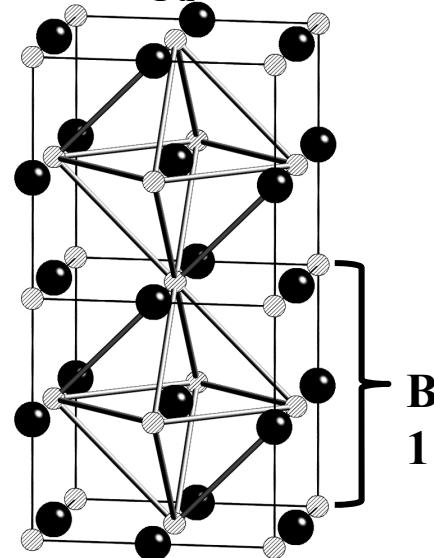
\* Y. Bohne et al. Vacuum 76 (2004)

# In-situ TEM observation during e<sup>-</sup> beam irradiation

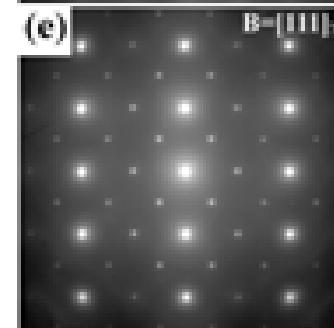
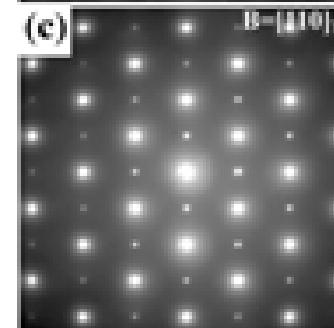
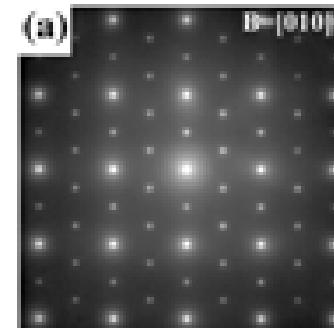
$\gamma\text{-NbN}_{1-x}$  (before)



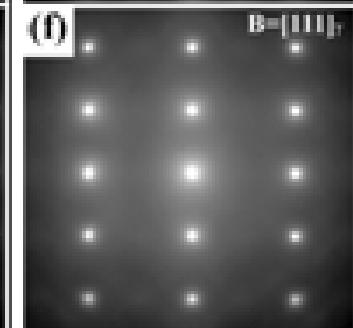
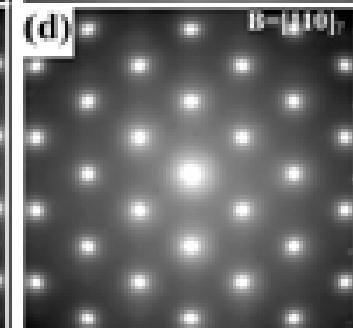
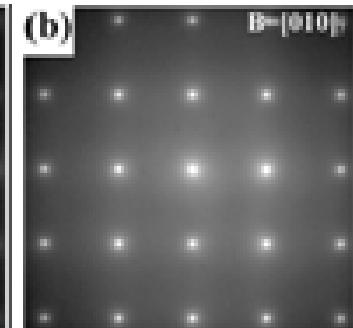
$\delta\text{-NbN}_{1-x}$  (after)



Before



After



# Reflection high-energy electron diffraction (RHEED)

- A high-energy electron beam (3-100 keV) is directed at the sample surface at a grazing angle of incidence
- It is possible to use both kinematic (elastic) and dynamic (inelastic) scattering

## Advantages

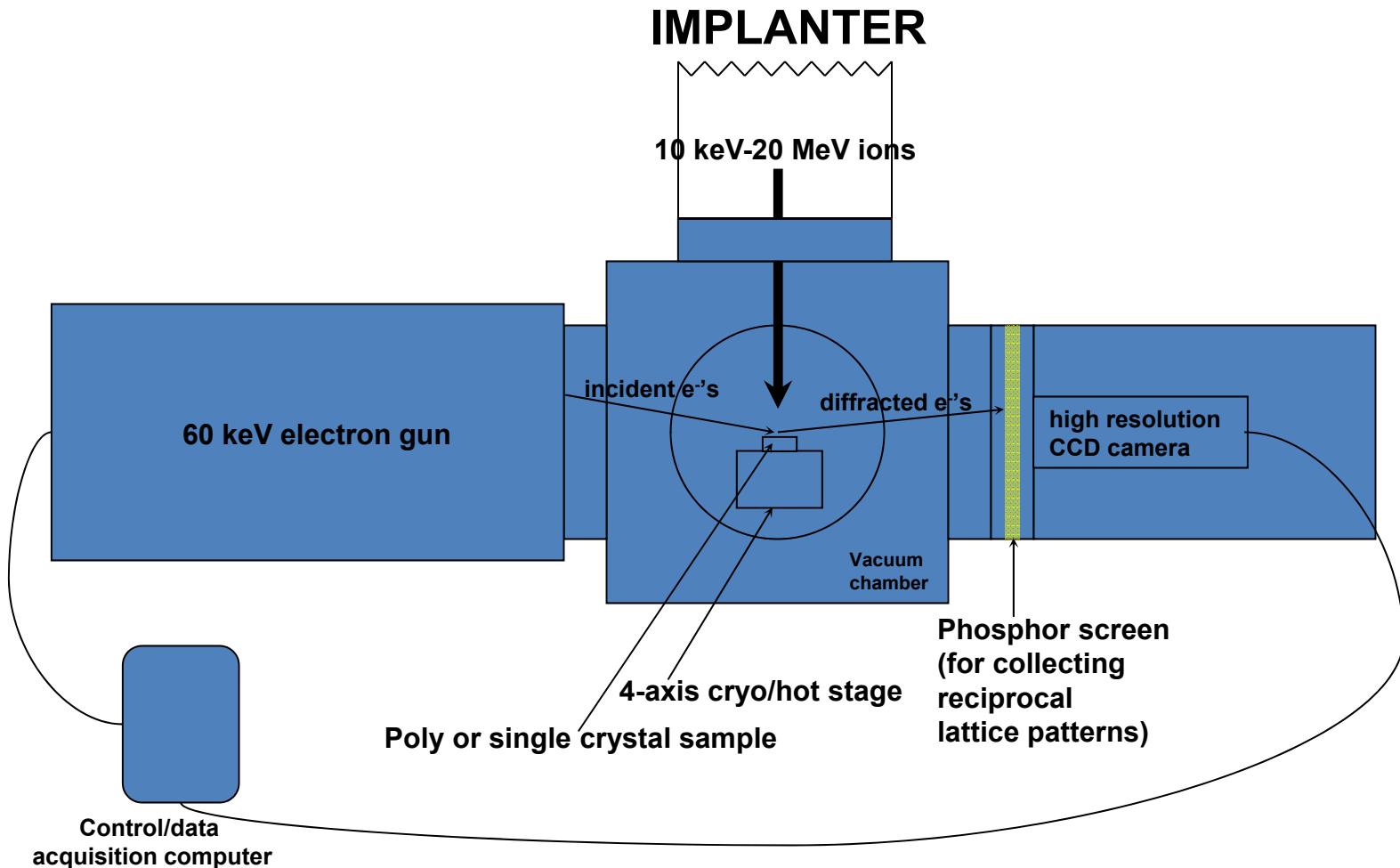
- High surface sensitivity
  - ability to measure very few atomic layers beneath the surface
- In-situ observation
  - phosphor screen with charge-coupled device (CCD) camera
  - real time sequential images and movie can be generated

	XRD	TEM	RHEED
Observation area	large	very small	large
Observation depth	deep	>100 nm	better than TEM
Instrument cost	high	high	Low
nondestructive	yes	no	yes

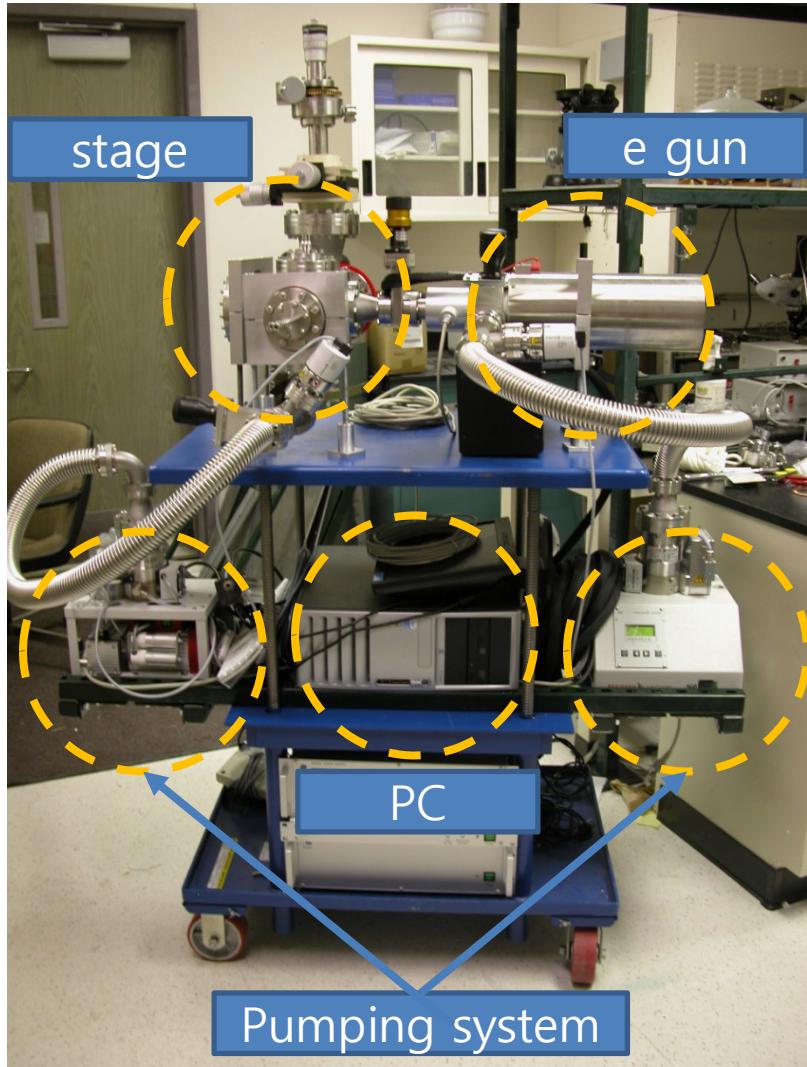
# Experimental procedure

- ▶ **Specimen** : LaAlO<sub>3</sub> single crystal (MTI corporation)
  - Rhombohedral at 25 °C and Cubic at > 435 °C, (100) surface normal
- ▶ **Ion implanter**: 200 kV Varian Implanter, 100 keV Ar<sup>++</sup> source
- ▶ **RHEED**: Guns system from STAIB Instruments (up to 60 keV)
  - Imaging system (kSA 400 analytical RHEED system)  
from k-Space associates, Inc.
- ▶ **Characterization**
  - GIXRD: Bruker AXS D8 Advance X-ray diffractometer  
(Cu-K<sub>α</sub> and θ–2θ geometry)
  - TEM : Sampling-combination of mechanical grinding and ion milling (PIPS)  
Observation- FEI Technai F30 TEM/STEM

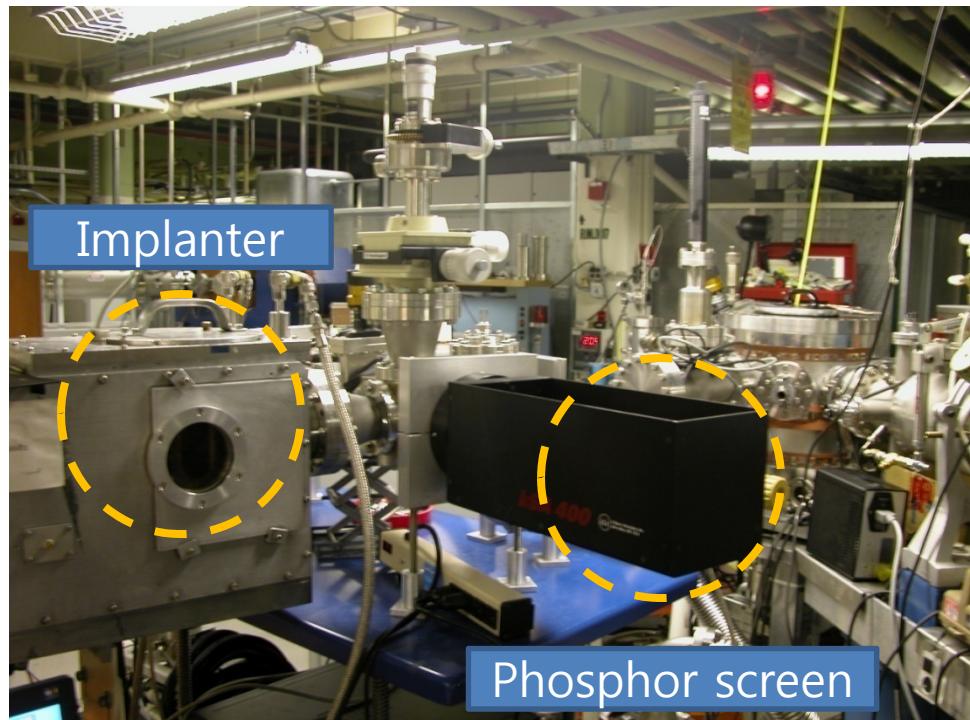
# System configuration



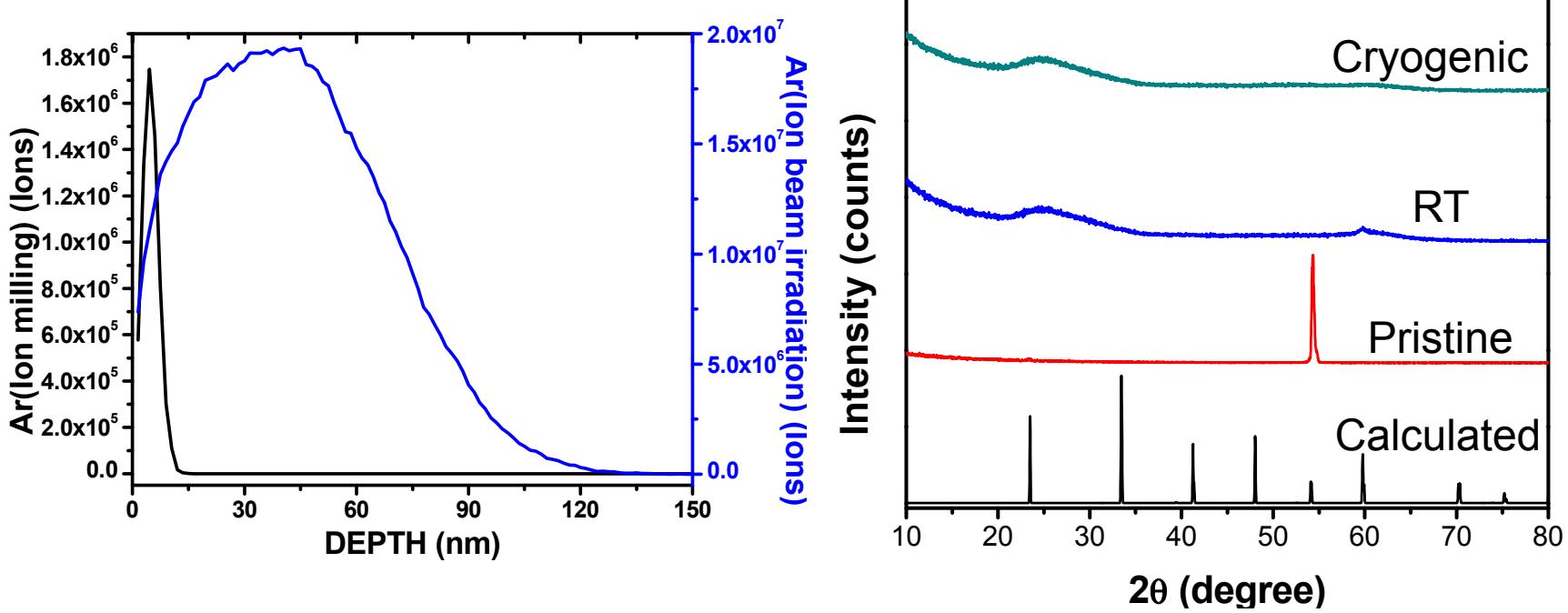
# System configuration



1. on implanter
2. RHEED
  - electron gun
  - sample stage
  - phosphor screen and CCD camera
  - control/data acquisition PC

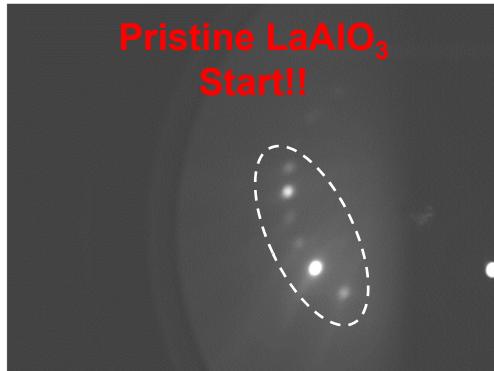


# Ion penetration depth calculated by SRIM2009 and GIXRD patterns from LaAlO<sub>3</sub>

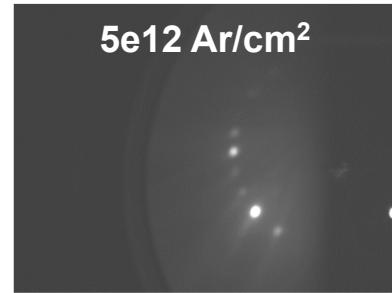
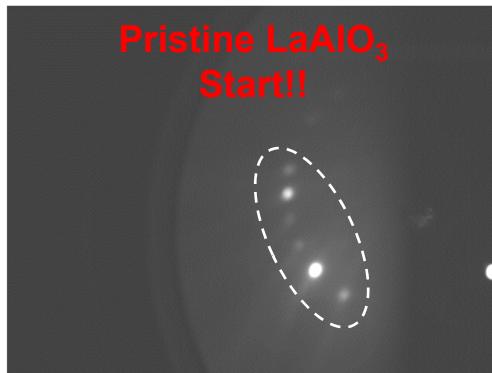


- Monte Carlo simulation show that the penetration depth is ~ 120 nm from the surface
- Calculated X-ray penetration depth is < 10 nm at a grazing angle of 0.25°
- After Ar irradiation, the LaAlO<sub>3</sub> single crystal phase near the surface is transformed into an amorphous phase

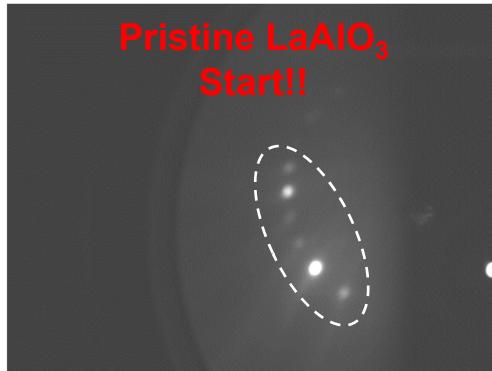
# In-situ RHEED sequential images for 100keV Ar<sup>2+</sup>ions irradiation of c-LaAlO<sub>3</sub> at RT



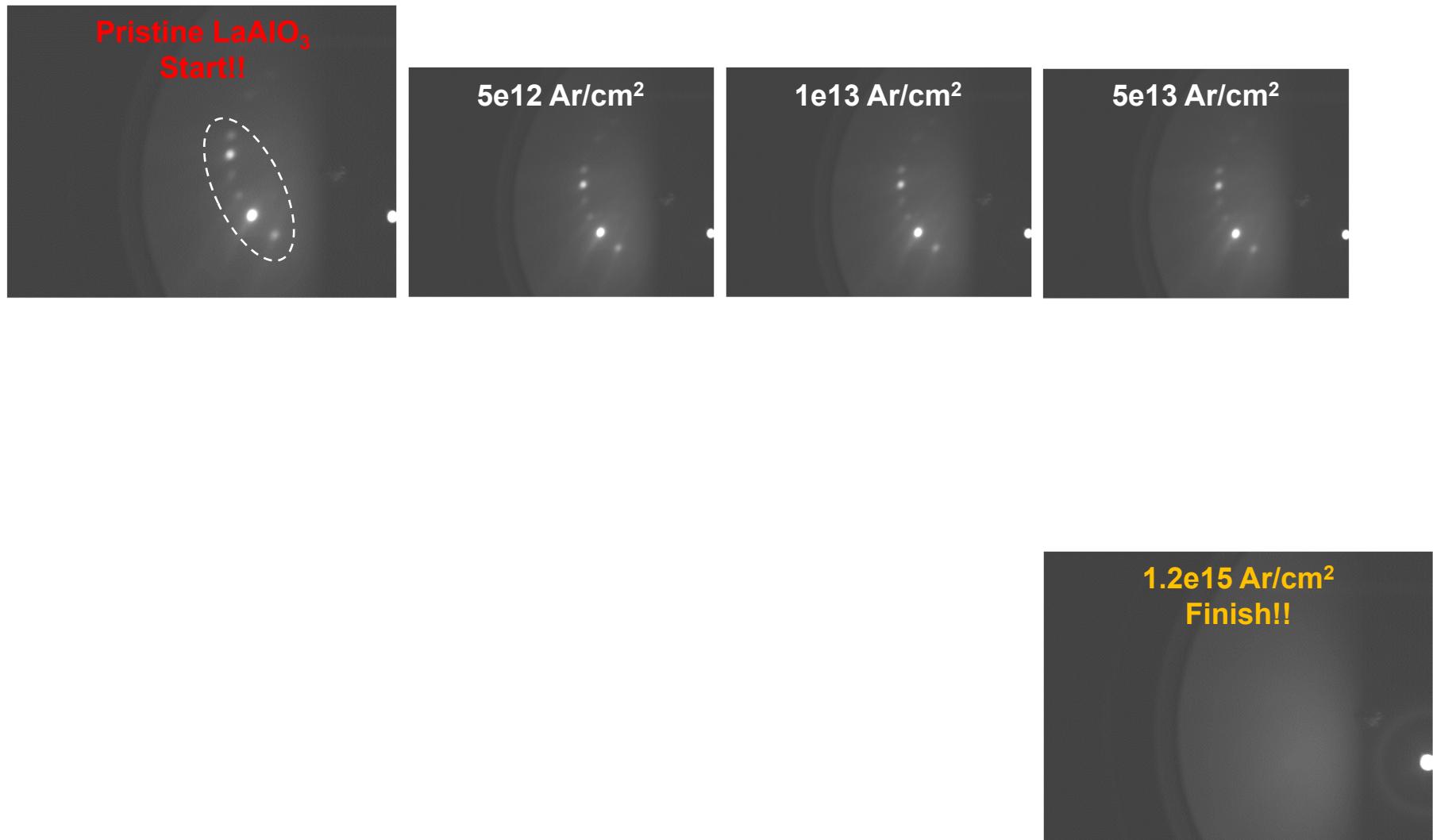
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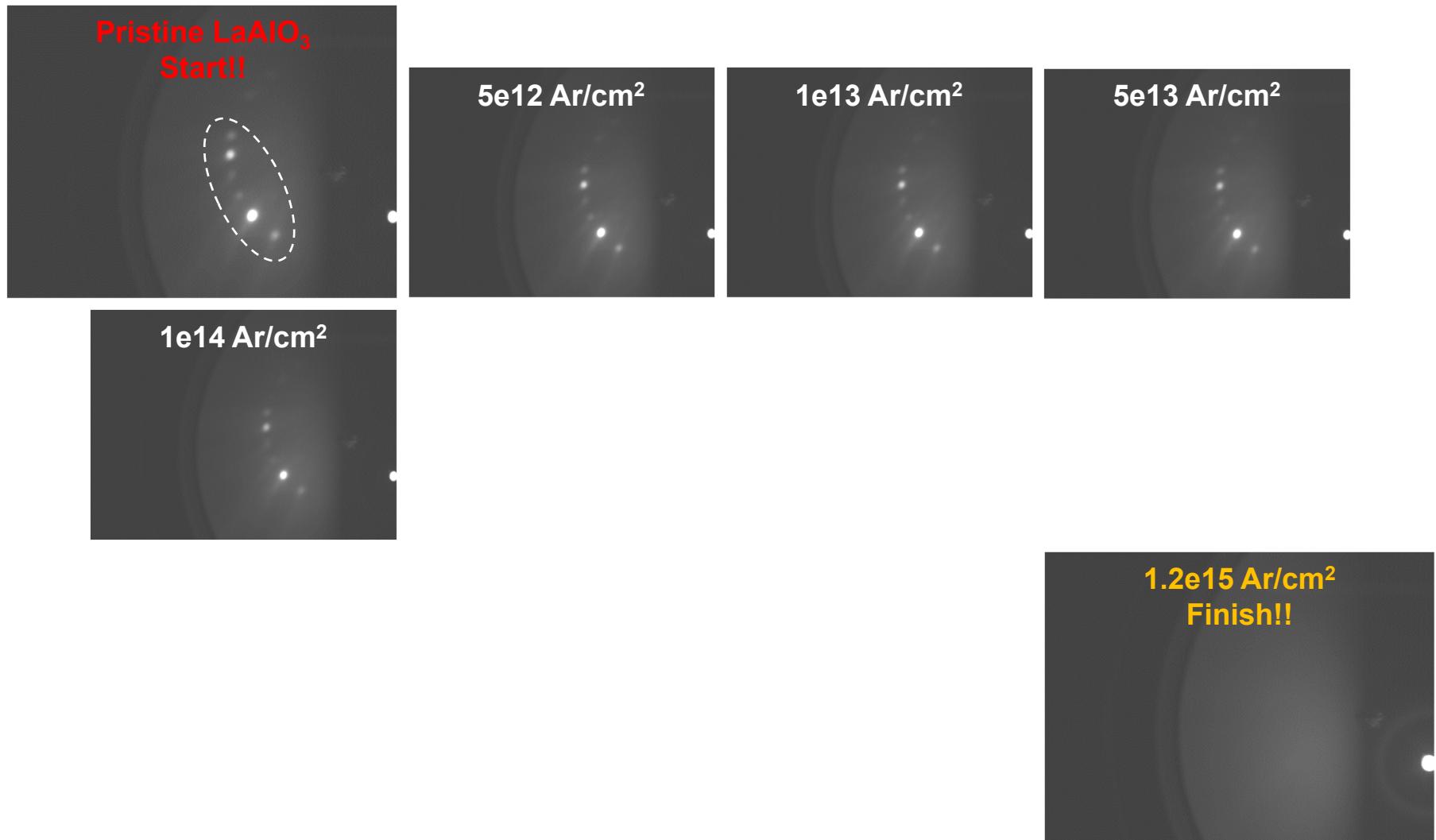
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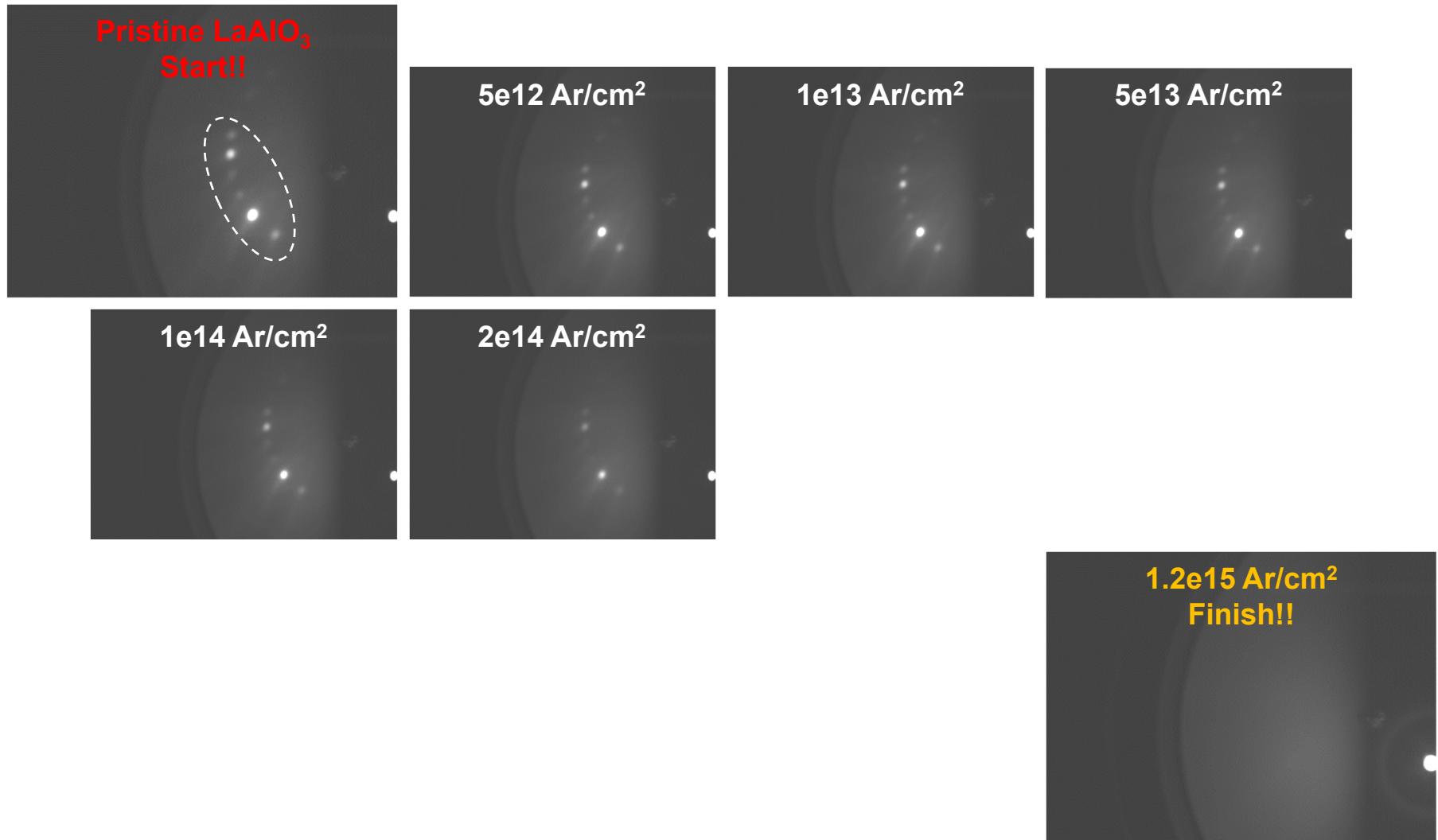
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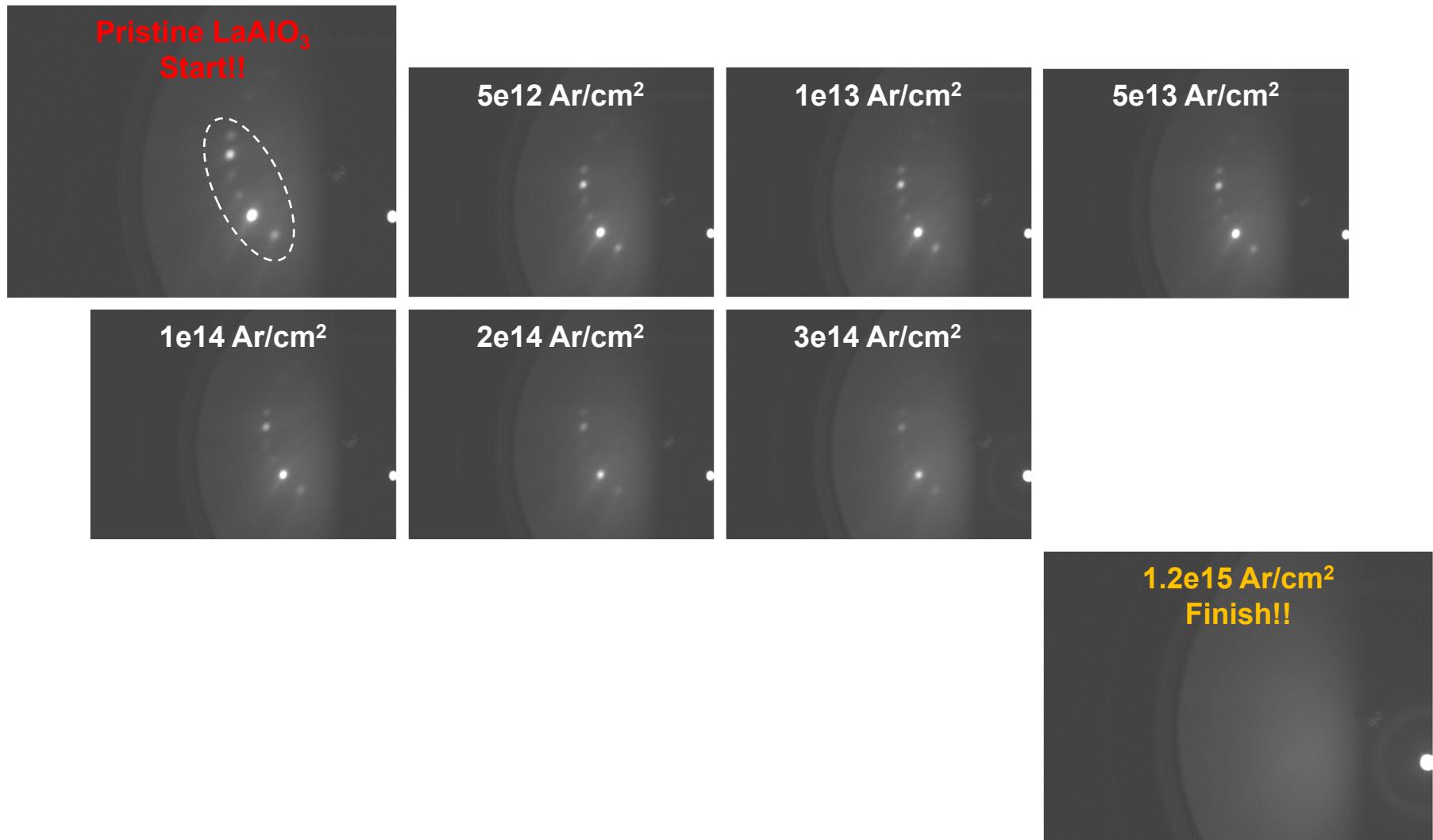
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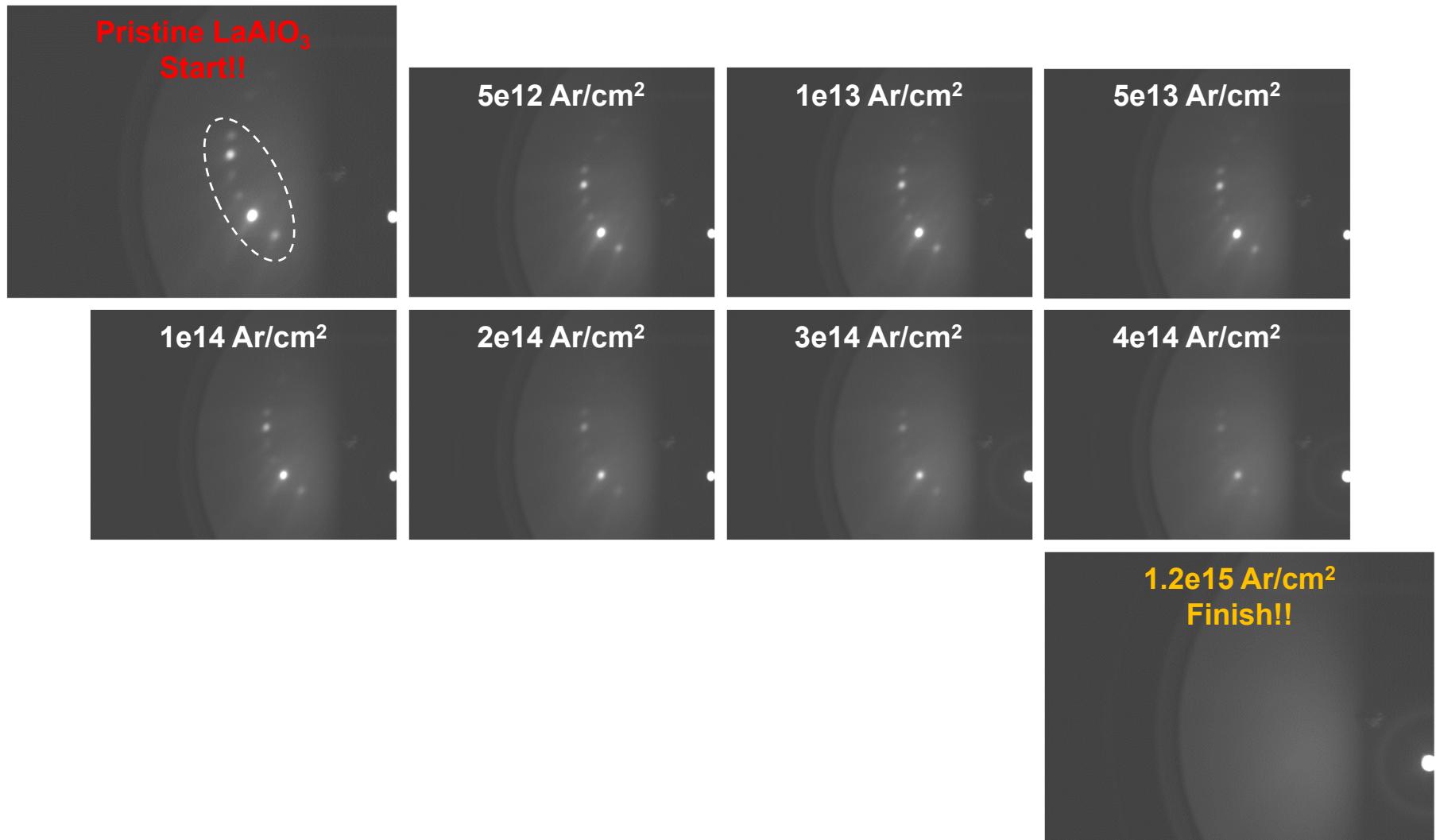
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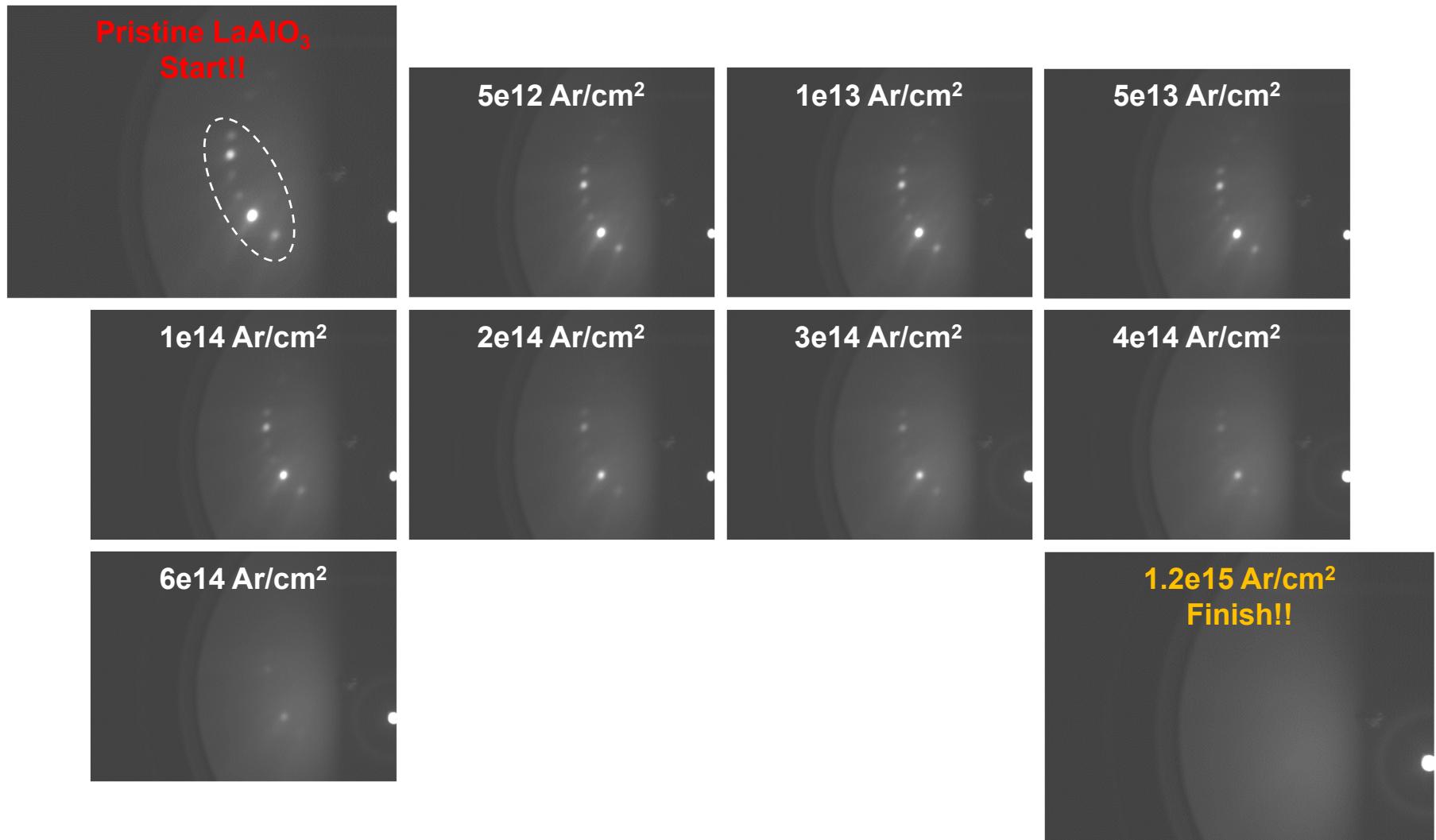
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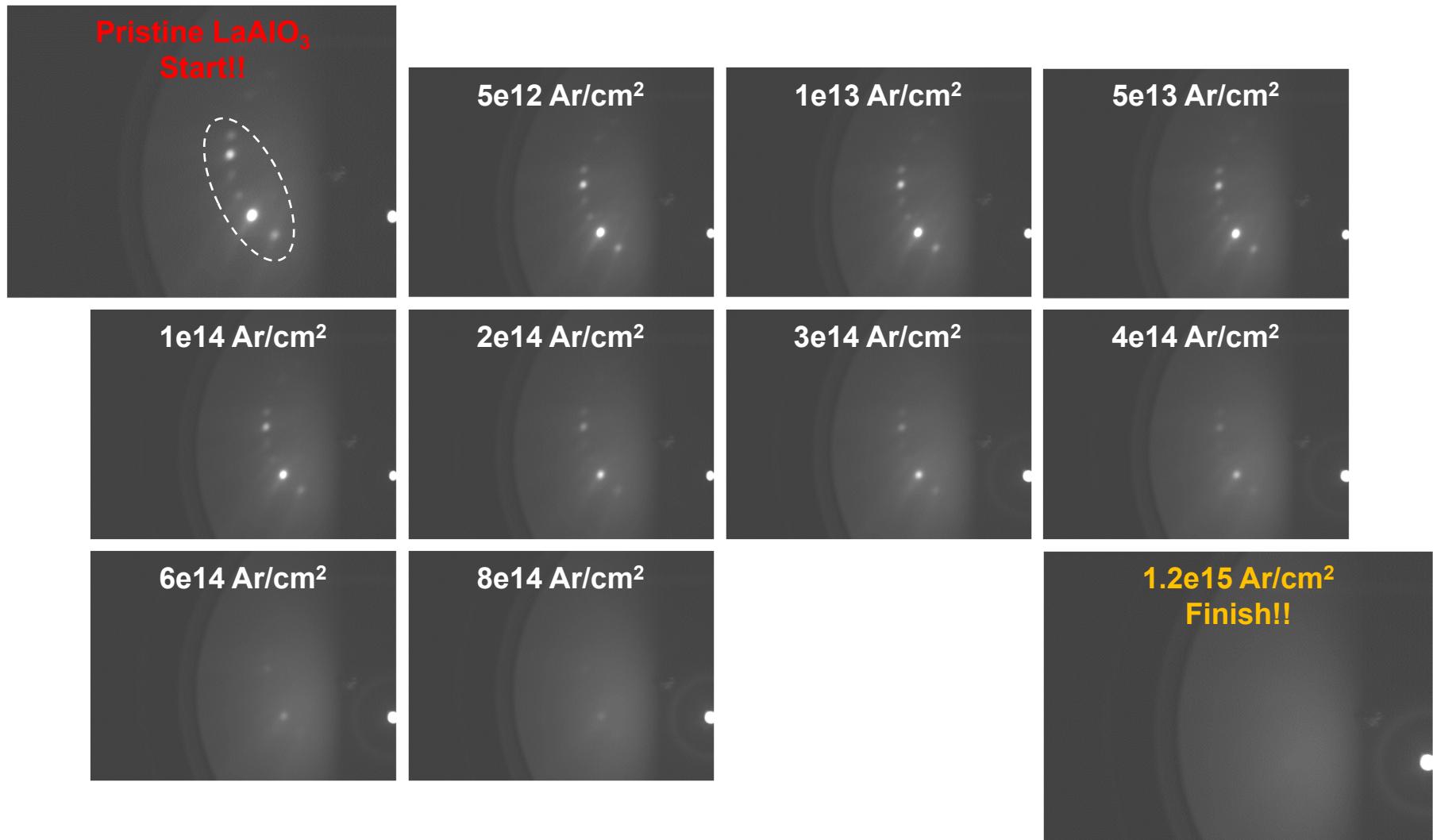
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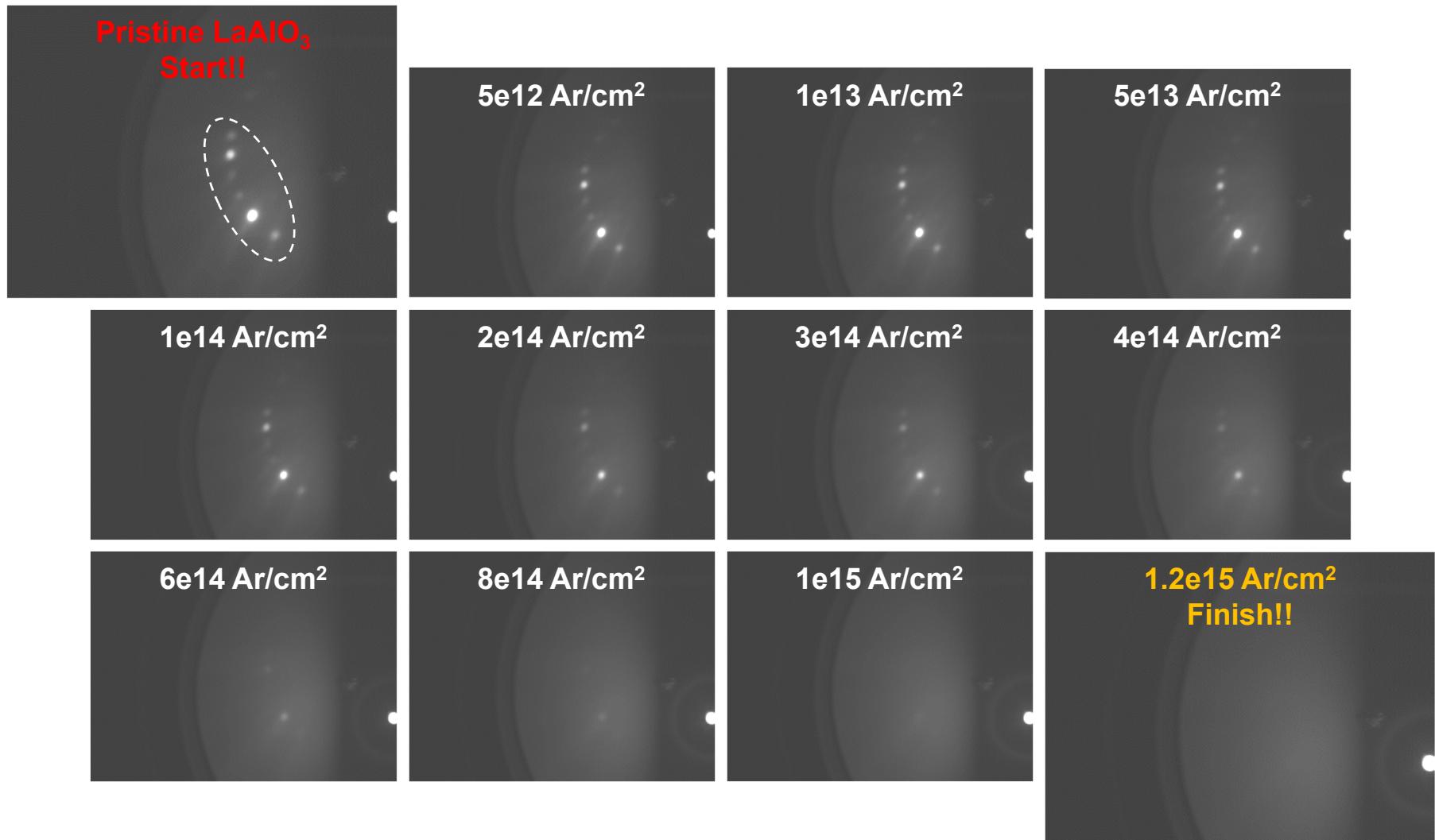
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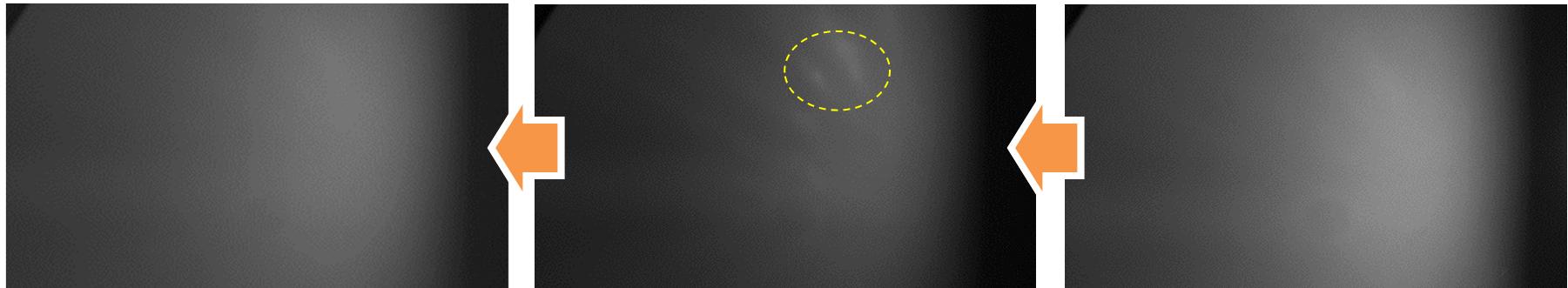
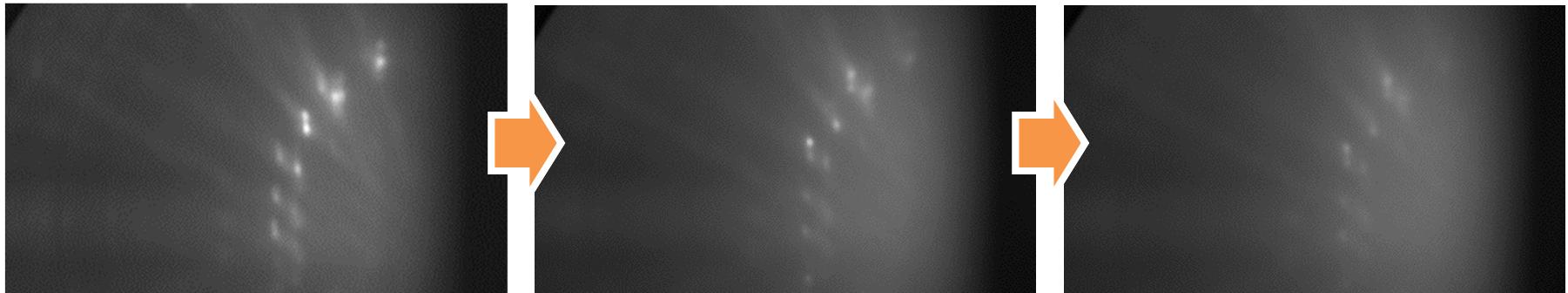


# Readjusted e<sup>-</sup> beam incident angle probe

Pristine- LaAlO<sub>3</sub>

5e13 Ar/cm<sup>2</sup>

2.38e14 Ar/cm<sup>2</sup>

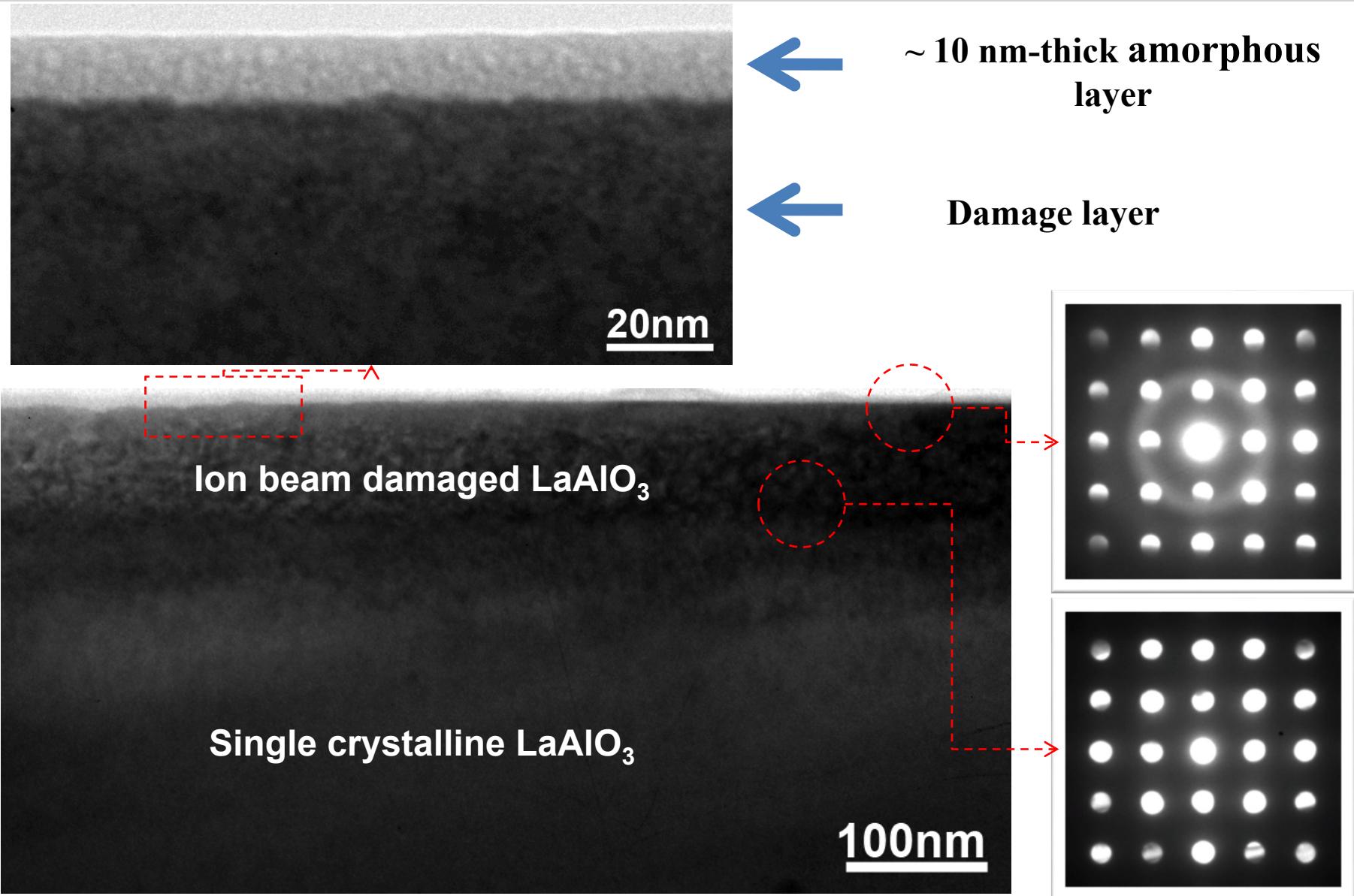


6.5e14 Ar/cm<sup>2</sup>

Readjusted e<sup>-</sup> incidence  
angle probe deeper  
into sample 3.5e14 Ar/cm<sup>2</sup>

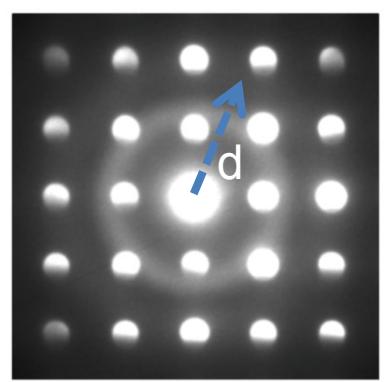
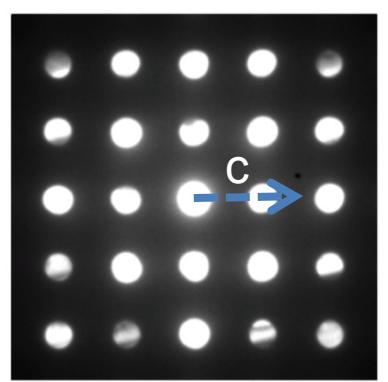
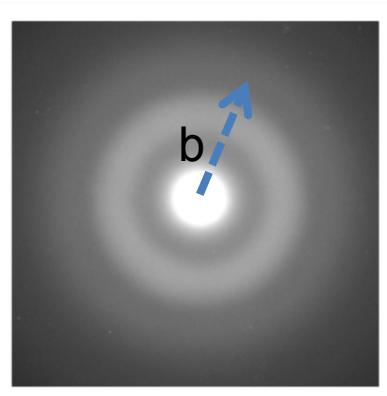
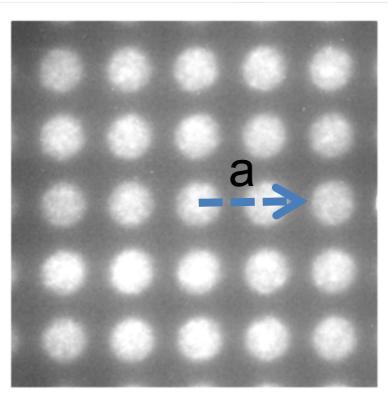
3.5e14 Ar/cm<sup>2</sup>

# TEM and MBD analysis of 100kev Ar<sup>2+</sup> ions irradiated sc-LaAlO<sub>3</sub> at RT



# MDP analyses of ion beam induced a-LaAlO<sub>3</sub>

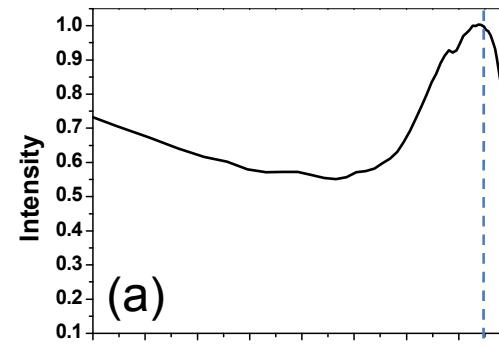
400 kV Xe irradiation



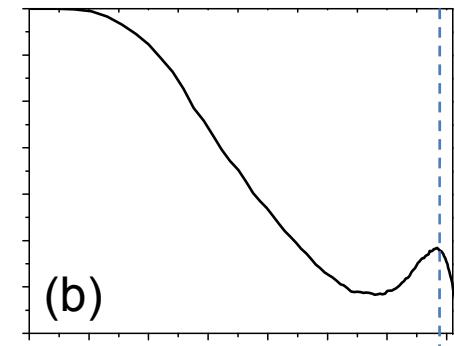
100kev Ar irradiation

Calculated reciprocal lattice distances from Xe (upper) and Ar (lower) are identical  
→ Crystalline (c) LaAlO<sub>3</sub> is transformed to an amorphous (a) phase by ion irradiation

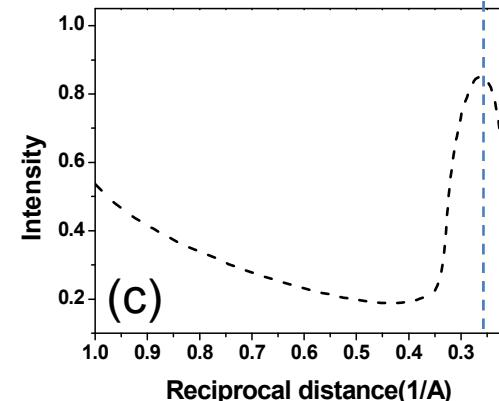
Spot spacing



(a)



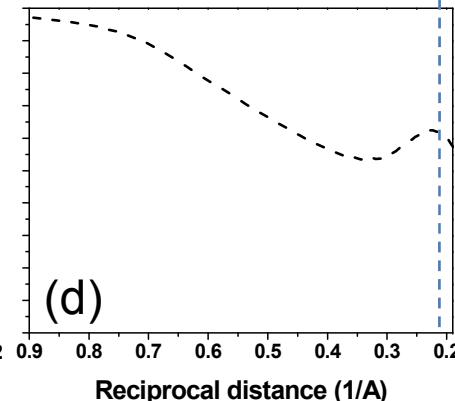
(b)



Intensity

Reciprocal distance(1/ $\text{\AA}$ )

(c)

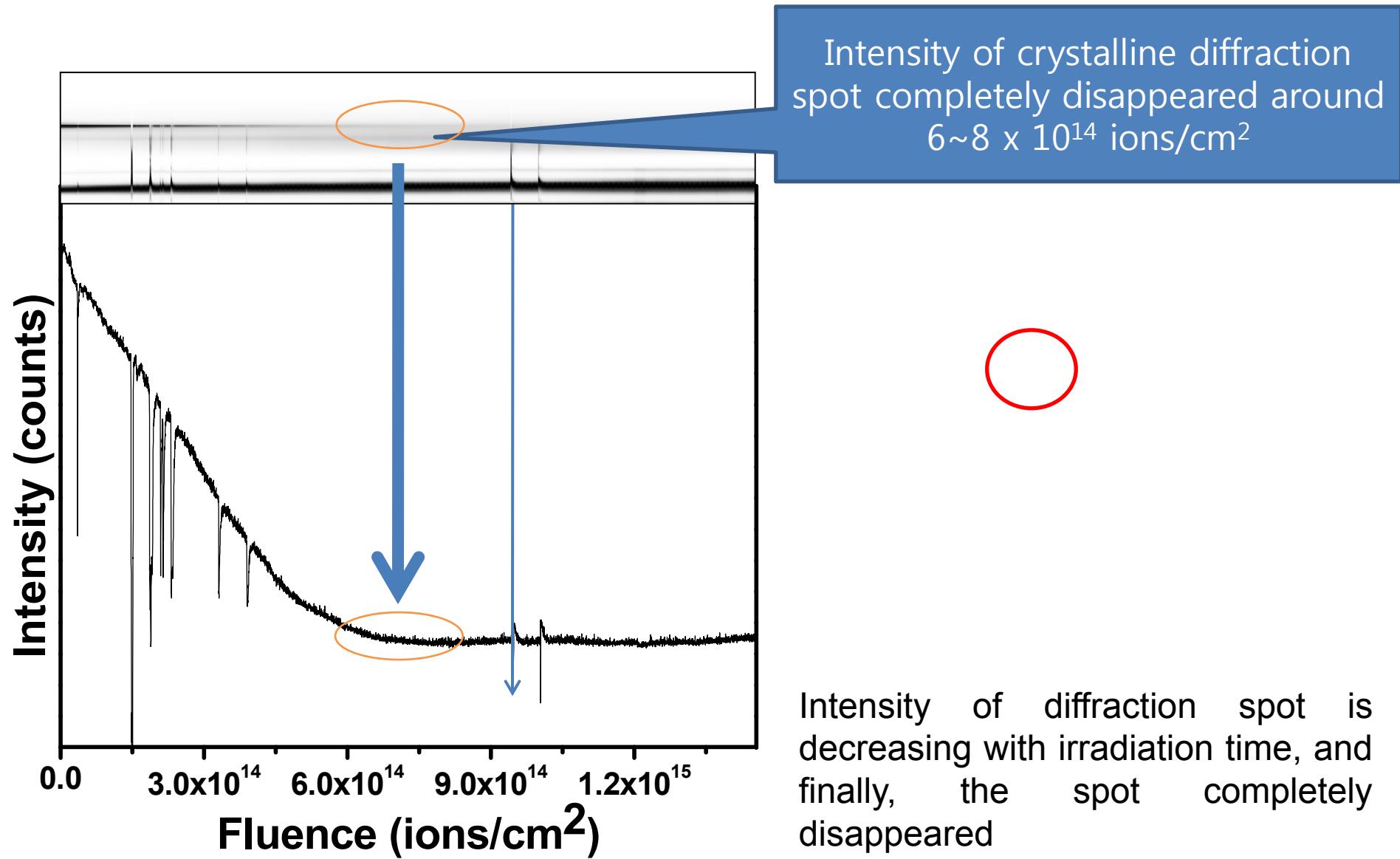


(d)

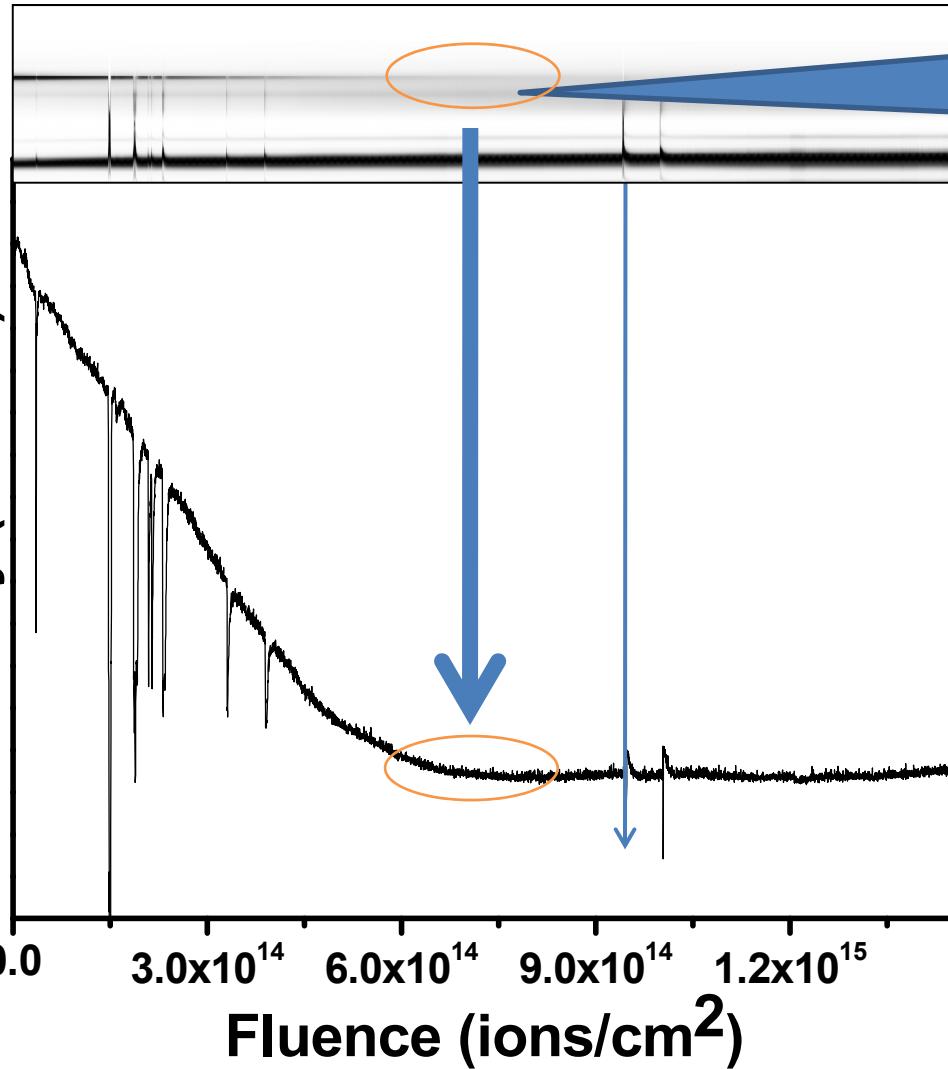
Intensity

Reciprocal distance (1/ $\text{\AA}$ )

# In-situ diffraction spot intensity evolution of LaAlO<sub>3</sub> at cryo temp.



# In-situ diffraction spot intensity evolution of LaAlO<sub>3</sub> at cryo temp.

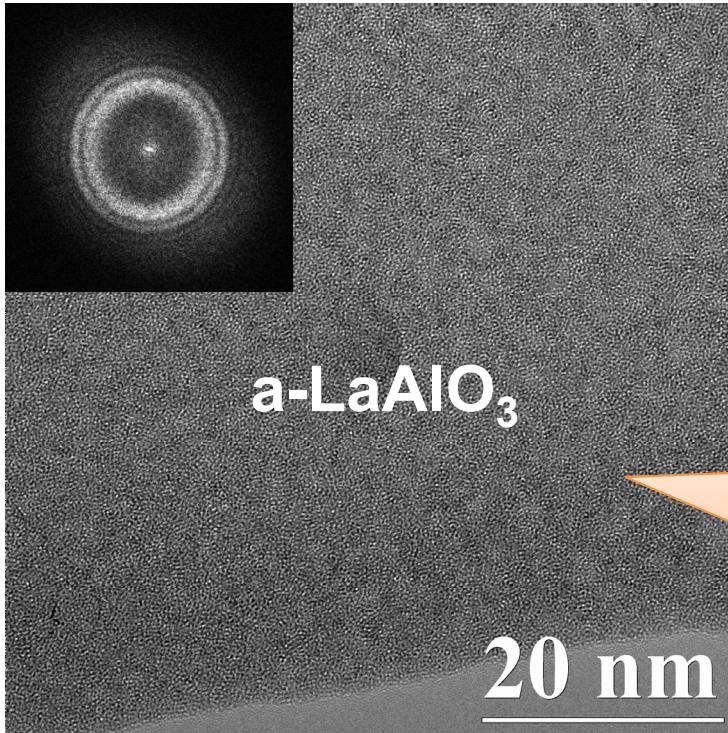
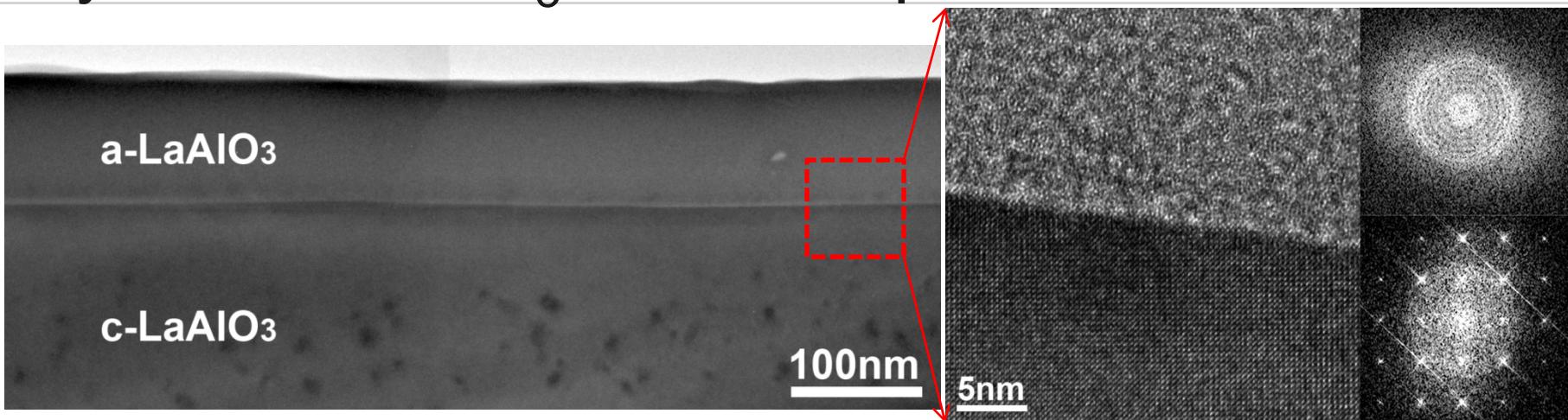


Intensity of crystalline diffraction spot completely disappeared around  $6\sim 8 \times 10^{14}$  ions/cm<sup>2</sup>



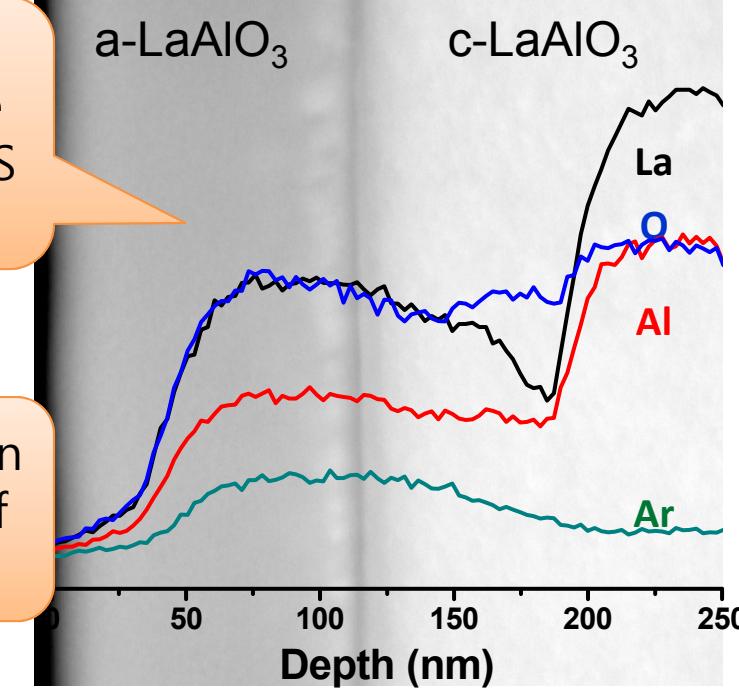
Intensity of diffraction spot is decreasing with irradiation time, and finally, the spot completely disappeared

# TEM/STEM-EDS observations for crystalline LaAlO<sub>3</sub> at low temp.



HAADF image  
with STEM-EDS

High resolution  
TEM image of  
a-LaAlO<sub>3</sub>



# Summary

1. Crystalline-to-amorphous (c-a) phase transformation of  $\text{LaAlO}_3$  was investigated by **combining 100 keV  $\text{Ar}^{++}$  ion irradiation with in-situ RHEED system**.
  - RT: around 100 nm-thick amorphous layer formed by irradiation
  - Cryogenic temp.: more than 100 nm-thick amorphous layer was formed by irradiation (~ identical to damage depth from SRIM2009 calculation)
2. Measured c-to-a phase transformation time (fluence) is dependent on the incident angle of the electron-beam. In the case of an incident angle of  $3^\circ$  at cryogenic temp., in-situ RHEED observations show that the phase transformation fluence is around  $6\sim8 \times 10^{14}$  ions/cm<sup>2</sup>



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# Thank You

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