

Q-Slope at High Gradients

(*Niobium Cavities*)

*Review about Experiments
and Explanations*

Introduction

- ❖ *Q degradation at high accelerating fields*
 - ❖ *Empirical Cure with cavity baking*
 - ❖ *Problem only push away farther*
 - ❖ *Understand Q-slope origin*

Outline

- Q-Slope : Definition and Features

- Baking Effect

- Experimental Observations (Q slope - R_{BCS} - R_{res})

- Standard Chemistry (BCP) and Electropolishing (EP)

- Consequences on Nb Surface (diffusion process - oxides - B_c)

- Theoretical Models

- Differences between BCP & EP

- Interface Oxide

- Superconducting Parameters Change

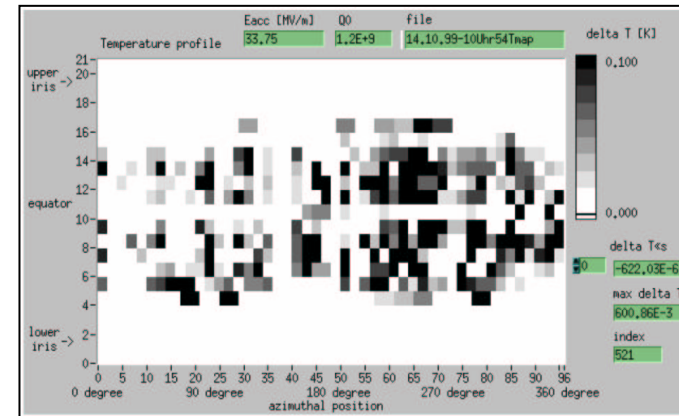
- Experiments ↔ Models

- Conclusion

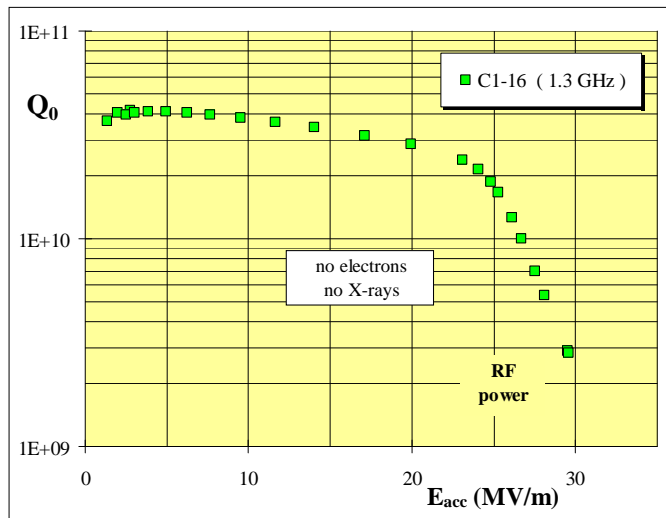
Q-Slope : Definition and Features

quality factor \Rightarrow strong degradation

- $E_{acc} > 20$ MV/m TTF cavities ($B_p > 85$ mT)
- field emission not involved (no e^- , no X rays)
- T map : global heating (B_p max)
- limitation by RF power supply or quench
- seemingly a typical feature of BCP cavities



(L. Lilje *et al.* - SRF '99 - Santa Fe)



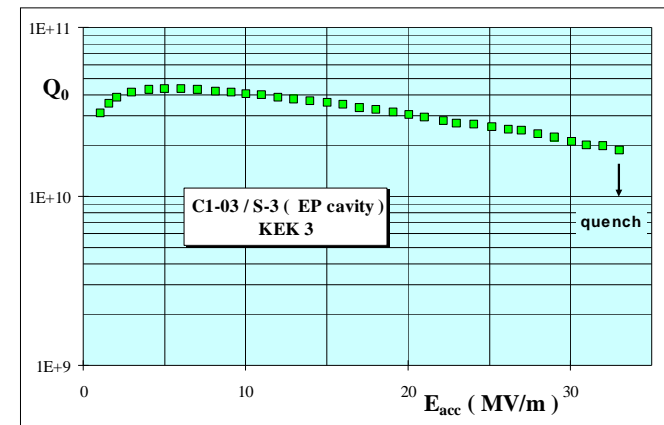
(Nb 1-cell cavity)

"European Headache"
superiority of EP
without Q-slope

K. Saito *et al.*

SRF '97

(Abano Terme)



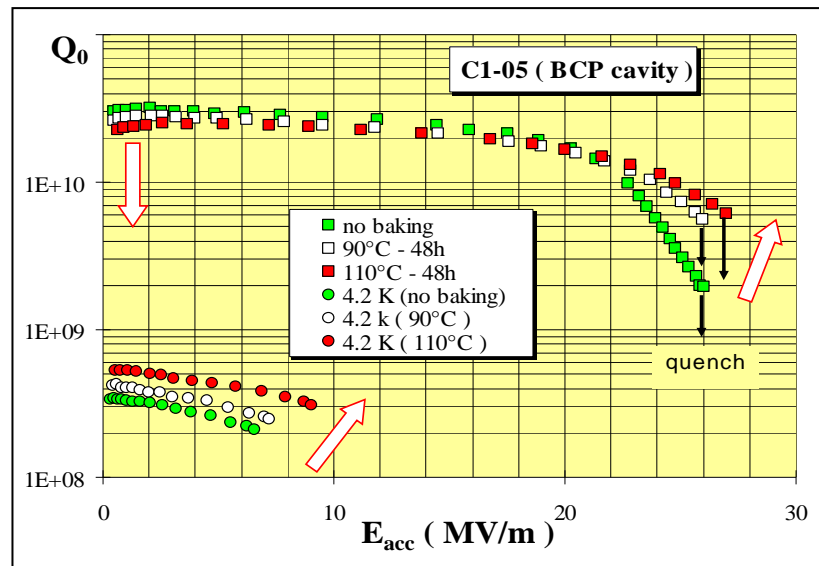
(E. Kako *et al.* - SRF '99 - Santa Fe)

Baking Effect on BCP Cavities

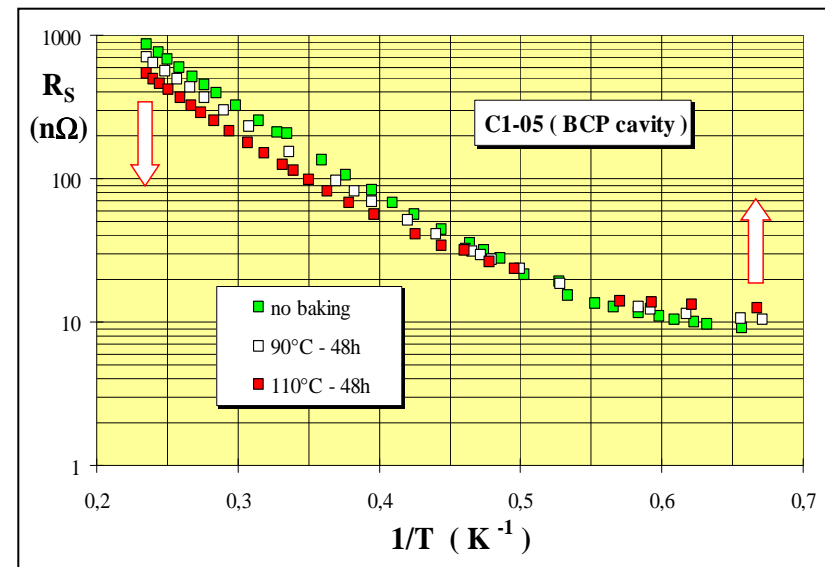
$$(Q_{\text{slope}} - R_{\text{BCS}} - R_{\text{res}})$$

"in-situ" baking discovered on BCP cavity

slope improvement ($90 < T < 120^{\circ}\text{C}$) - degradation ($T > 150^{\circ}\text{C}$)



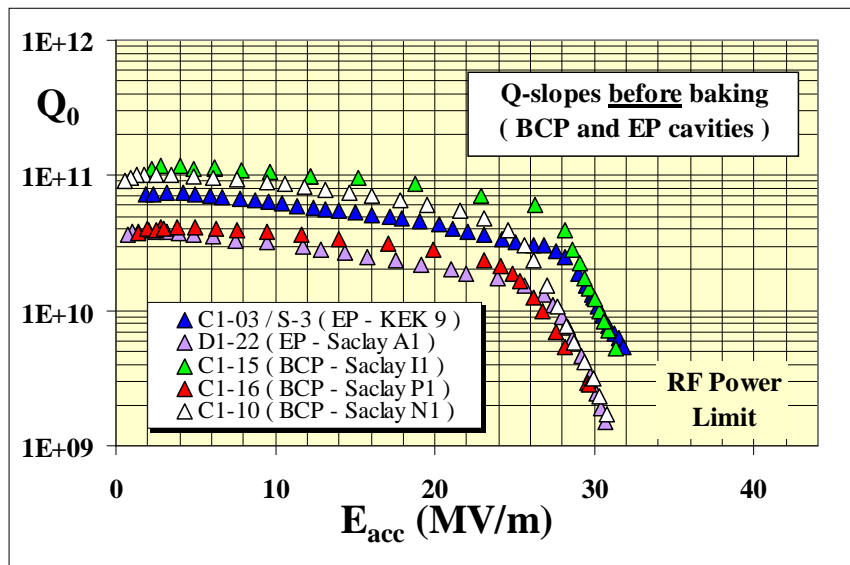
(B. Visentin *et al.* – EPAC '1998 - Stockholm)



Baking Effect on EP Cavities

Same phenomenon on **E.P.** cavities

before baking: Q-slope identical to BCP



SRF ' 99
Santa Fe

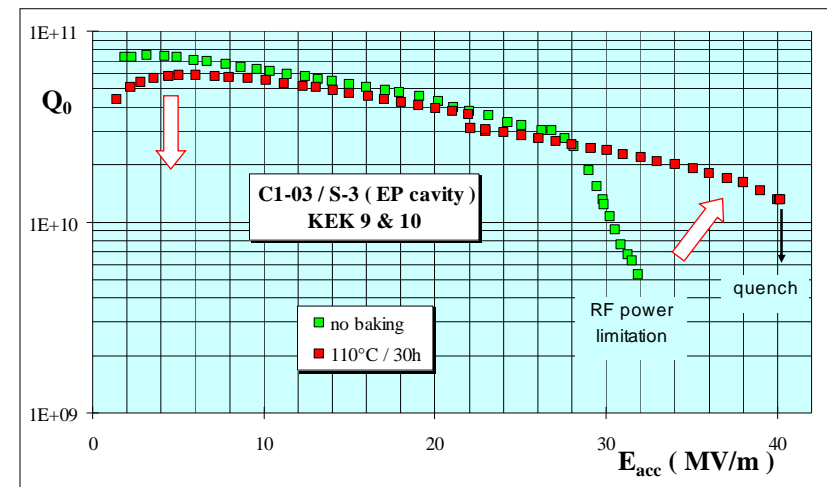
P. Kneisel. – *TUP 044*

K. Saito. – *TUP 031*

L. Lilje *et al.* – *TUA 001*

B. Visentin *et al.* – *TUP 015*

after baking : Q-slope improvement



(Saclay cavity – EP & tested @ KEK)

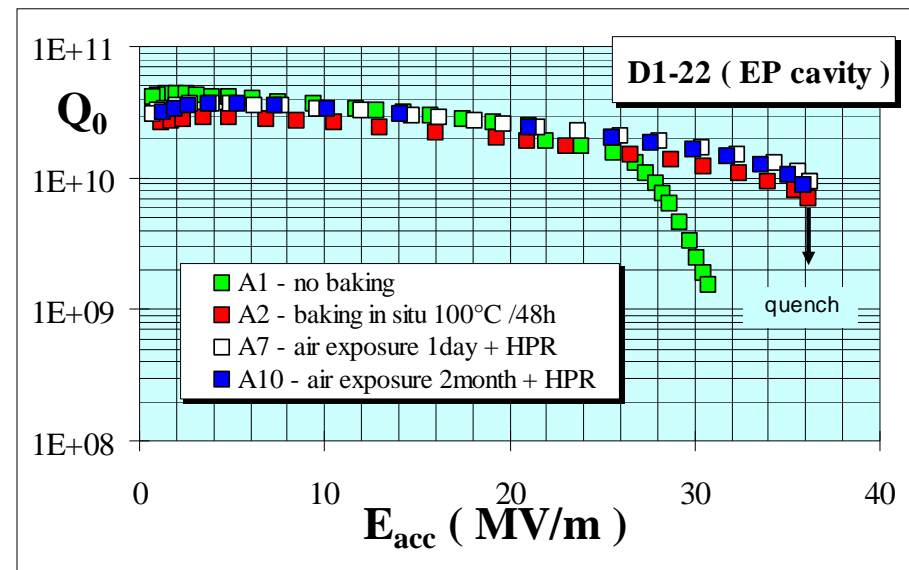
apparent superiority of EP reported before ?
cleaning procedure at KEK
wet cavities (High Pressure Rinse)
directly pumped out and baked at 85°C/20h
to accelerate pumping speed

Surface Re-Oxidation after Baking ?

Unaltered Q-slope

after **Air** exposure (3 hours to 2 months) followed by :
 High Pressure **Water** Rinse + Drying (laminar flow - 3 h)
 - standard conditioning for RF tests -

- 3 hours (cavity closed)
- 8 hours (cavity closed)
RF test bench
- 1 day (cavity closed)
- 9 days (cavity **open**)
(under laminar-flow : class 10)
in clean-room
- 2 months (cavity **open**)
left on the shelf

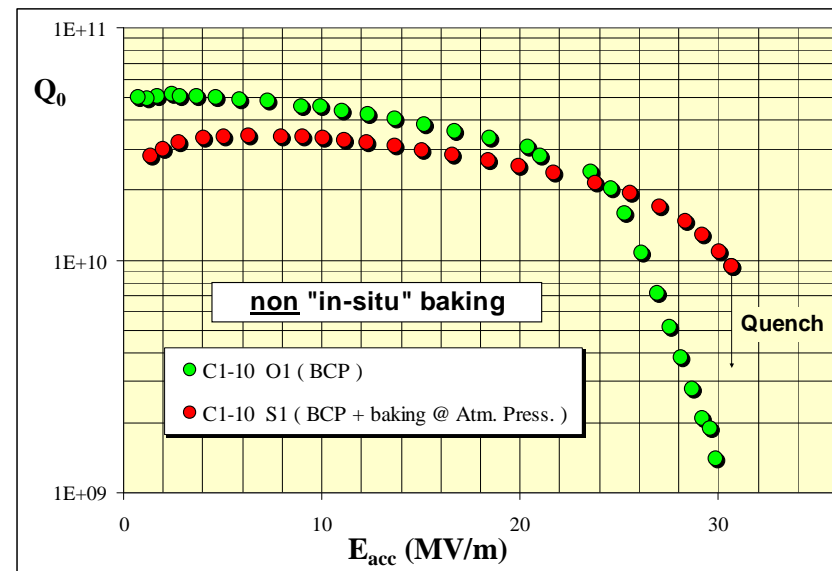


Baking at the Atmospheric Pressure ?

BCP + High Pressure Rinse
 Wet Cavity inside Drying Oven (110°C / 60 h)
 @ Air Atmospheric Pressure - no pumping
 new H.P.R. for RF test

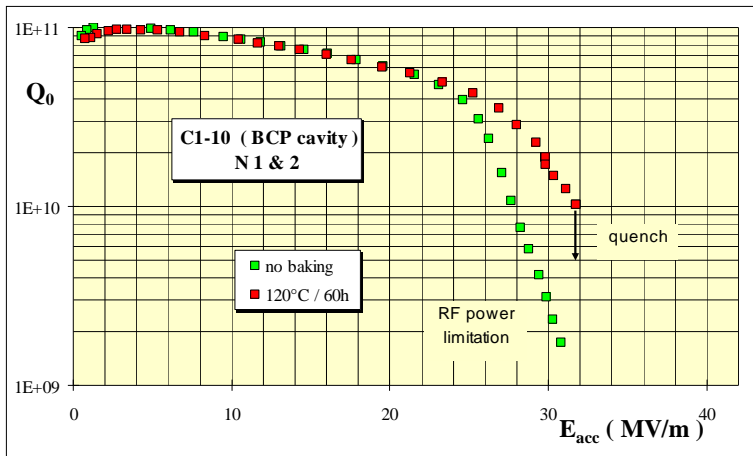
- Cavity (C1-10)
- Q-slope improvement
- Similar to “in-situ baking”

(B. Visentin *et al.* – this workshop)

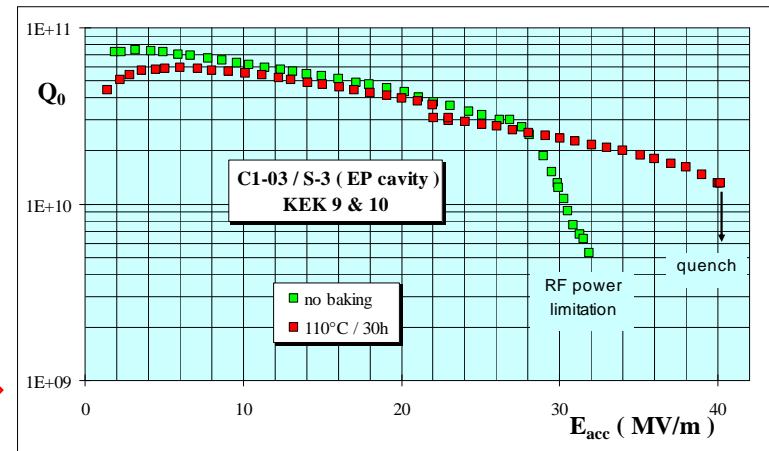


Differences between BCP and EP

- higher efficiency of baking on EP cavities (from 85°C)
- residual slope on BCP cavities even with baking (120°C)

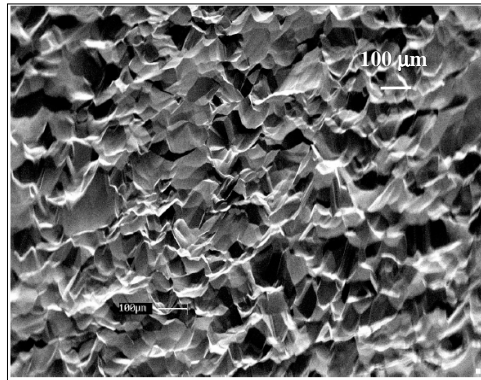


⇔ BCP



EP ⇒

- higher quench field for EP cavities (40 MV/m)



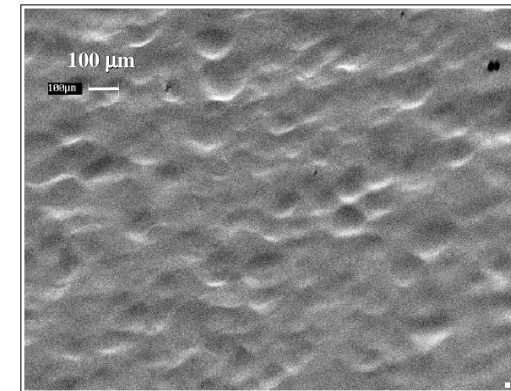
- surface roughness

(R.L. Geng *et al.* - SRF '99 – Santa Fe)

⇔ BCP (117 μm)

EP (90 μm) ⇒

5-9 μm (statistic on step height) 2-5 μm

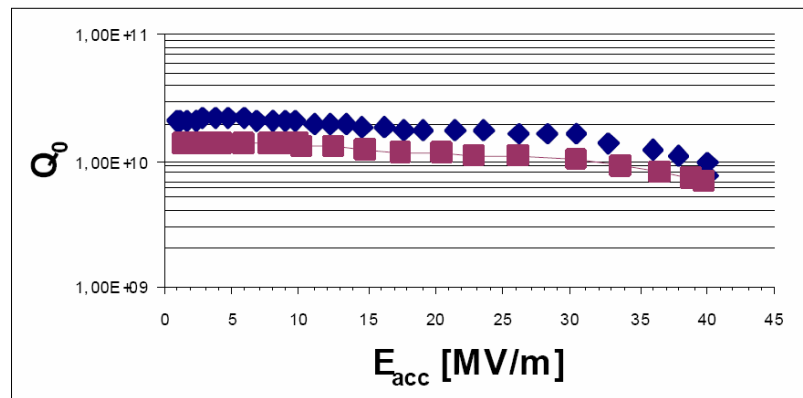


Some Exceptions (?)

Total removal of Q-slope after baking (BCP cavities) (with or without quench fields at 40 MV/m)

Nb cavity “defect free” : (**BCP** but **no baking** specified)

(P. Kneisel *et al.* – SRF ’1995 – Gif/Yvette)



One NbCu clad cavity : 1NC2 (**BCP + 140°C/30h**)

(W. Singer *et al.* – SRF ’2001 - Tsukuba)

Two Nb cavities : C1-15 & C1-16 (**BCP + 120°C/60h**)

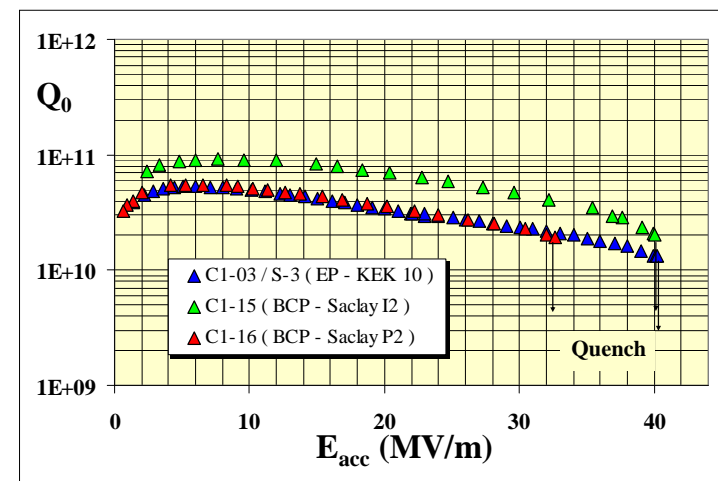
(B. Visentin *et al.* – EPAC’2002 – Paris & this workshop)

C1-15 (inner surface) ⇒

not very smooth

large grains : 2-3 mm²

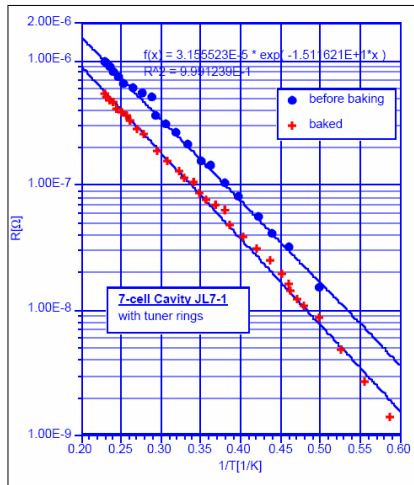
high steps : 4 to 8 μm



Baking Consequences on Nb Surface (1)

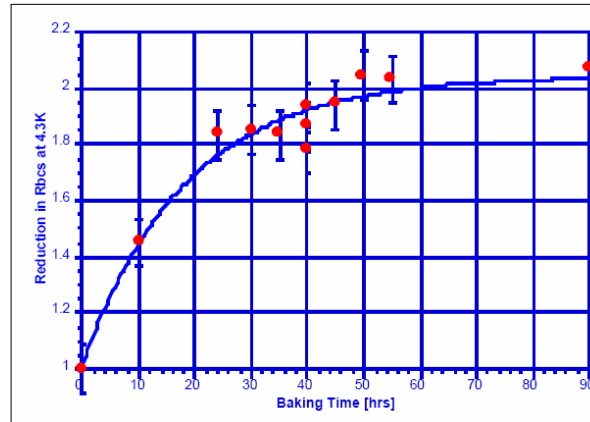
$R_{BCS}(T_{bake}) : \downarrow$

$R_{BCS} \downarrow$ (baking time) \Rightarrow saturation



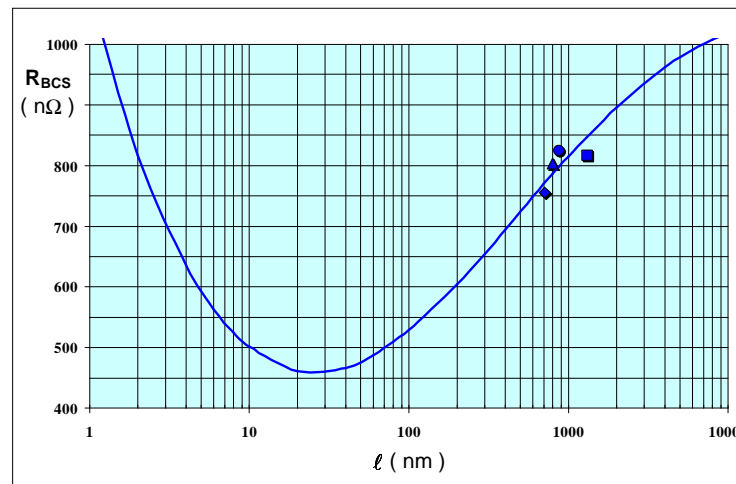
(P. Kneisel - SRF '99 - Santa Fe)

$$R_{BCS} = A(\lambda_L, \xi_F, \ell) \frac{\omega^2}{T} e^{-\Delta/kT}$$



diffusion process
(300 nm)

R_{BCS} @ $T = 4.2$ K
 $T_{bake} = 145^\circ\text{C}$



$\lambda_L = 31$ nm
 $\xi_F = 62$ nm
 $\Delta = 1.46$ meV
 $T = 4.2$ K

Baking Consequences on Nb Surface (2)

(F. Palmer – IEEE Trans. Mag. - 1987)

Influence of oxide layers on Nb surface resistance
(Nb_2O_5 and NbO)

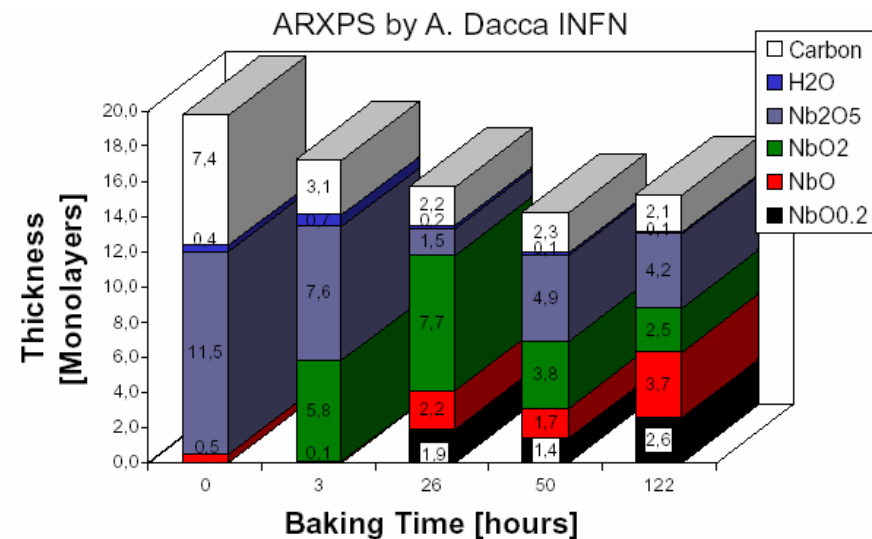
Oxygen can diffuse at low temperature

Change
of the structure oxide
after baking
(Nb_2O_5 ↓ and NbO - $\text{NbO}_{0.2}$ ↑)

A. Dacca *et al.* - Applied Surf. Science - 1998

C. Antoine *et al.* - SRF '99 - Santa Fe

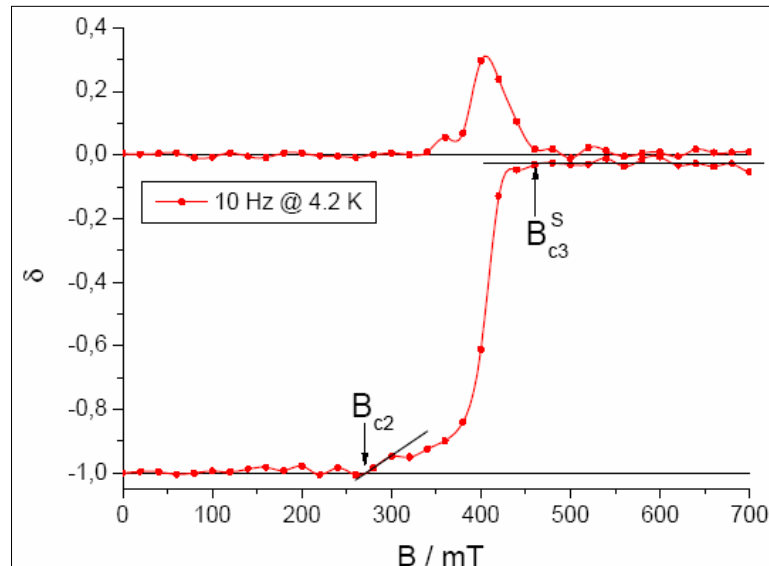
Q. Ma *et al.* - SRF '01 - Tsukuba



(A. Daccà - sample 4 - T=150°C)

Baking Consequences on Nb Surface (3)

Surface Magnetic Field :



	B_{c3S}/mT	B_{c2S}/mT
BCP unbaked	464	273
baked 123C, 48h	560	329
baked 144C, 48h	625	368
150 μ m EP unbaked	520	306
baked 123C, 48h	620	365
40 μ m EP unbaked	480	282
baked 123C, 48h	660	388

(B. Steffen - TTF Meeting - 2003)

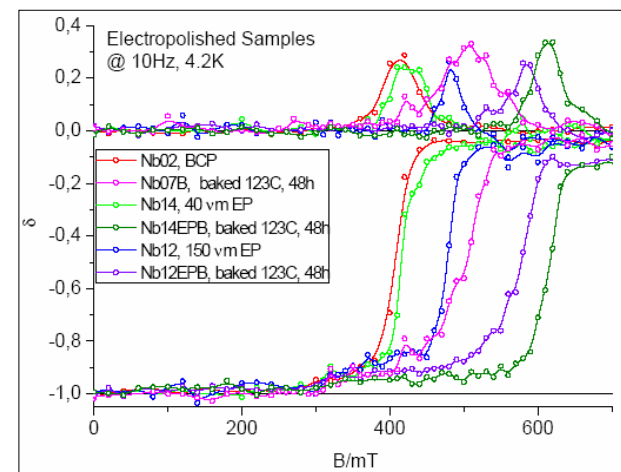
Susceptibility measurements (on sample)

Surface field : $B_{c3}^{surf} = 1.7 B_{c2}^{bulk}$

larger field for EP compare to BCP

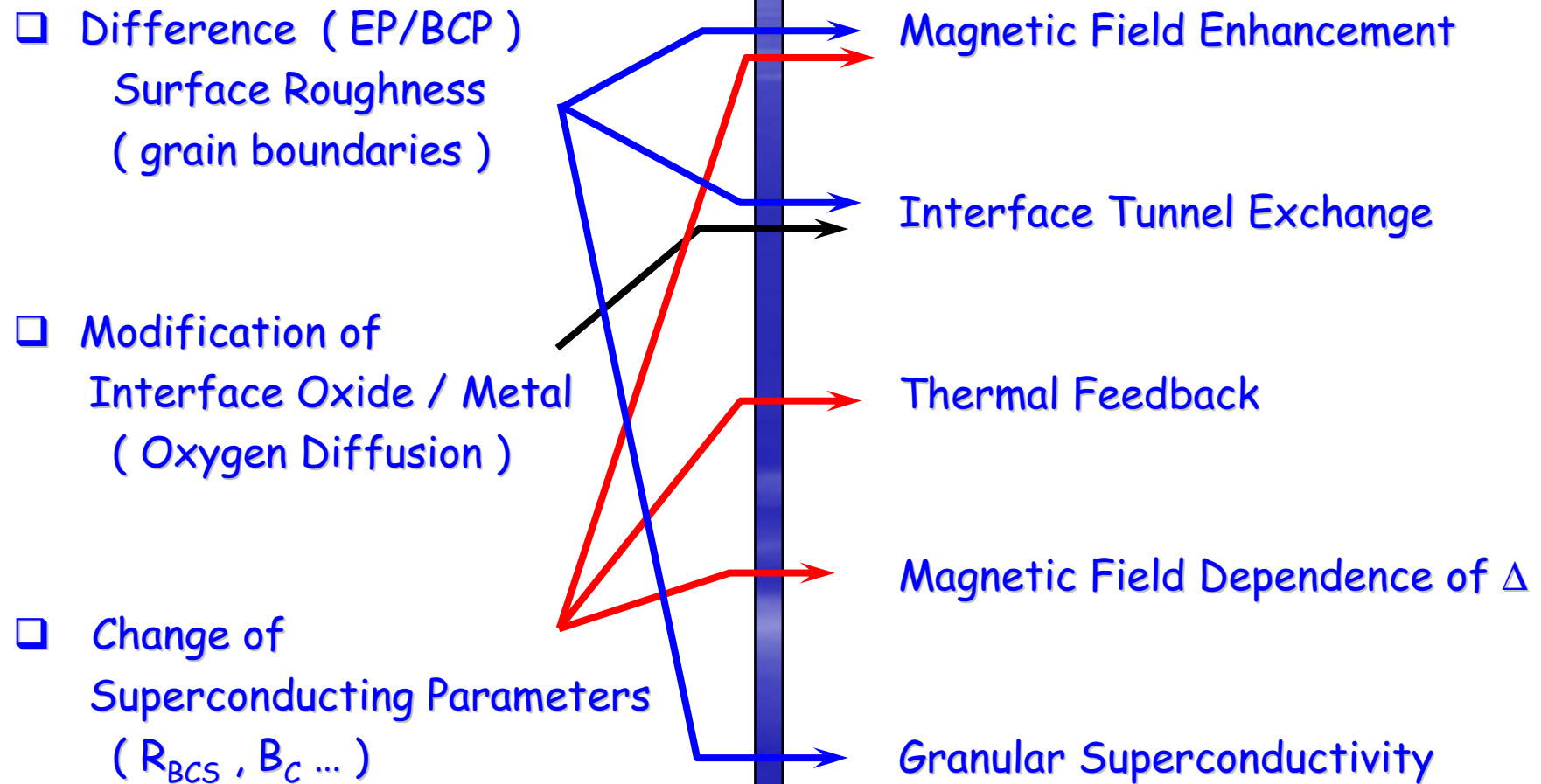
All values are increased by baking

(interpreted by enhancement of impurities : O ?)

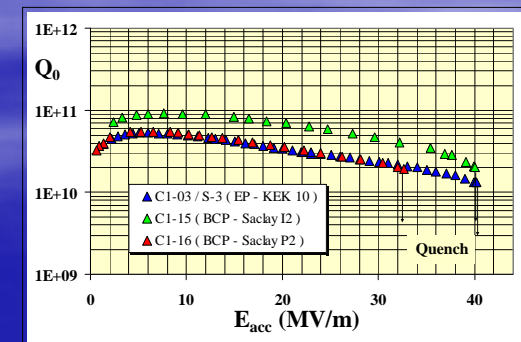
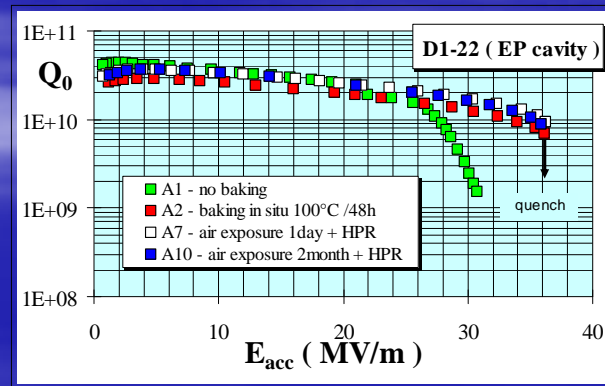
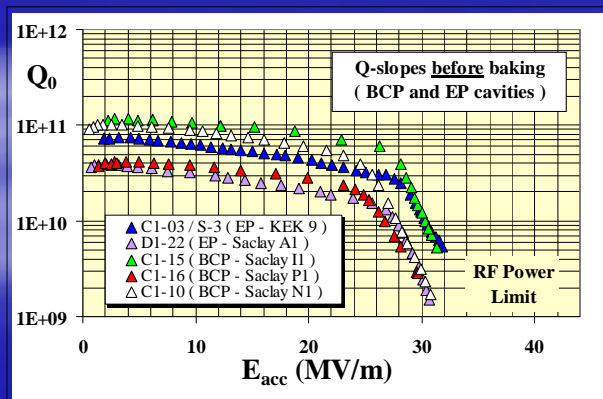
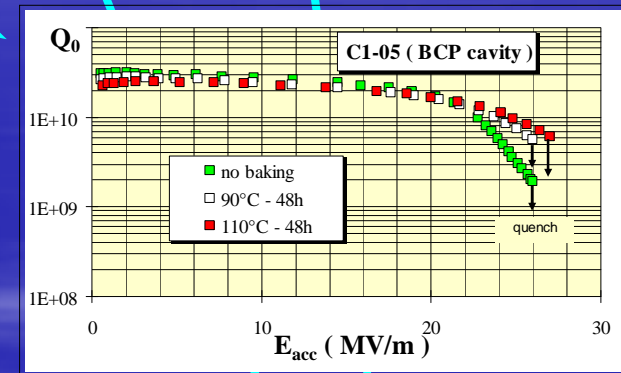
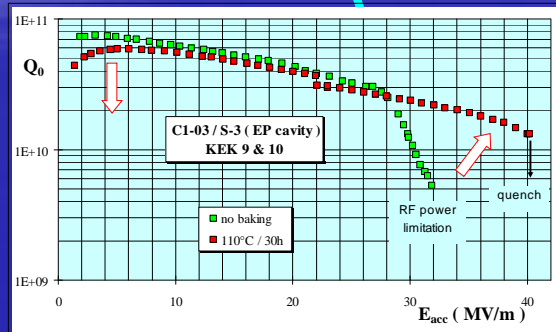
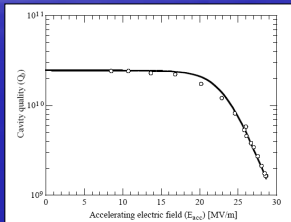
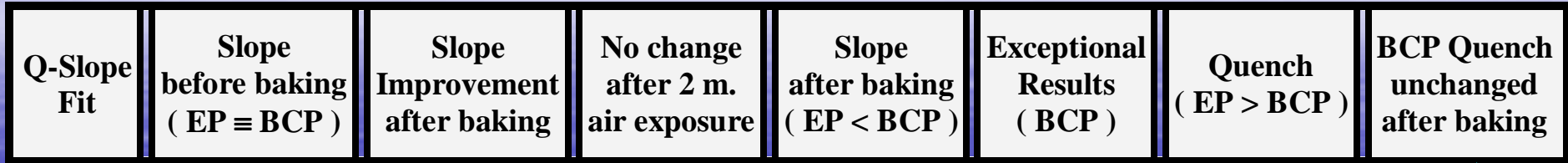


Experiments

Models

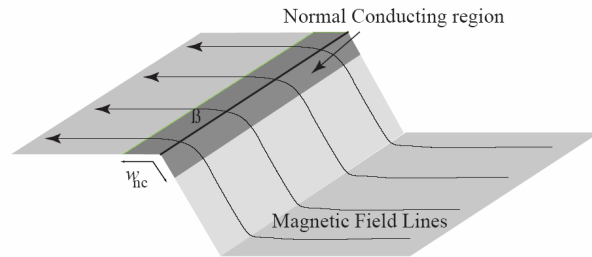


Theoretical Models \Leftrightarrow Experiments



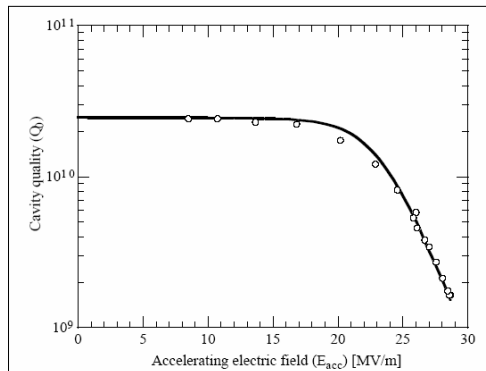
Magnetic Field Enhancement at G.B.

(J. Knobloch - SRF '99 - Santa Fe)



Q-slope origin

the most dissipative G.B. \Rightarrow quench (equator)



electromagnetic code + thermal simulation $\Rightarrow Q_0(E_{acc})$

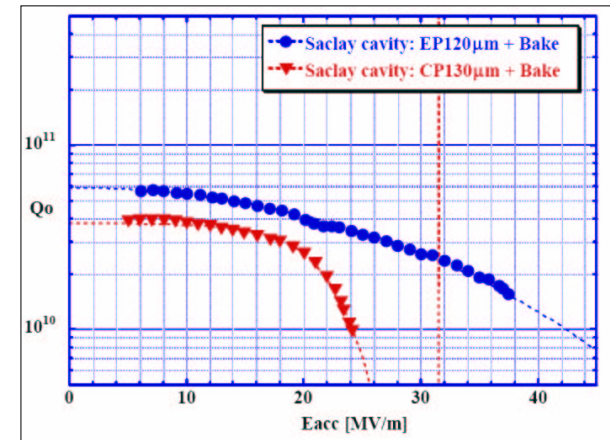
microstructure on RF surface

(surface roughness - step height $10 \mu\text{m}$)

magnetic field enhancement $\beta_m H$

normal conducting region if $\beta_m H > H_C$

factor $1.6 < \beta_m < 2.5$ (BCP)



(K. Saïto - PAC '2003 - Portland)

EP : ($H_C/\beta_m = 223 \text{ mT}$) $\beta_m=1$

BCP : ($H_C/\beta_m = 95 \text{ mT}$) $\beta_m=2.34$

Comments (H - enhancement)

❑ Explanations :

- Q-slope for BCP before baking (good simulation)
- Q-slope improvement after baking ($H_C \uparrow$)
- better slope for EP after baking (β_m lower ~ 1)

❑ Not consistent with :

- slope before baking for EP cavities
(same slope with β_m lower and H_C higher than BCP)
- flat slope (and 40 MV/m) on BCP cavities C1-15 & C1-16
(roughness : 4 to 8 μm $> 2 \mu\text{m} \Rightarrow$ high β_m)
- quench value unchanged for BCP after baking (in spite of $H_C \uparrow$)

Interface Tunnel Exchange

RF field on metallic surface $\left\{ \begin{array}{l} H^{\parallel} \rightarrow Z^H \rightarrow R^H = R^0 \left(1 + \gamma^* H^2 / H_C^2 + \dots \right) \quad (\text{Taylor series}) \\ E^{\perp} \text{ (causes electron emission)} \rightarrow Z^E \text{ (negligible for clean metal)} \end{array} \right.$

Dielectric oxide layer on metal \rightarrow enhancement of Z^E by I.T.E.

(localized states of $\text{Nb}_2\text{O}_{5-y}$ and density of state of Nb)

with electron diffusion at $\text{NbO}_x - \text{Nb}_2\text{O}_{5-y}$ interface

β^* : electric field enhancement factor

conventionally fitted by: $R^E = R^0 (E^{\perp})^8$

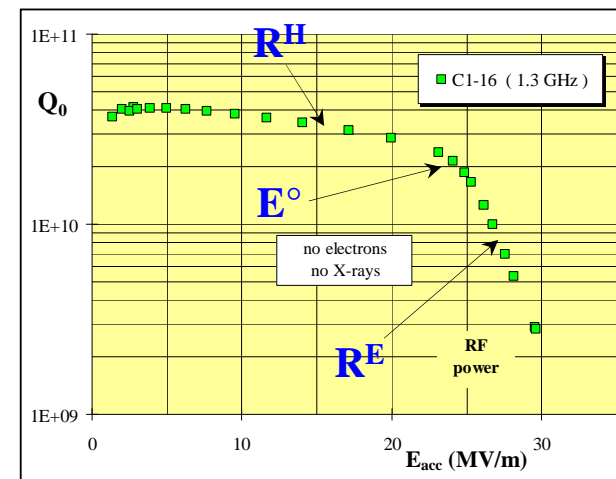
I.T.E. \equiv quantitative description of Q-slope

ITE reduction by :

- smoothening surface (EP)
($\beta^* \downarrow$ and $E^{\circ} \uparrow$)
- baking : $\text{Nb}_2\text{O}_{5-y}$ vanished - better interface
(reduction of localised states)

$$R^E \propto e^{-\frac{C}{\beta^* E^{\perp}}}$$

starting at E° onset value



(J. Halbritter - SRF '01 – Tsukuba)

(IEEE Trans. on Appl. Supercond. 11, 2001)

Comments (I.T.E.)

❑ Explanations :

- Q-slope improvement after baking ($\text{Nb}_2\text{O}_5 \downarrow$)
- better slope for EP after baking (smooth surface - lower β^*)

❑ Not consistent with :

- similar slopes (before baking) for EP and BCP cavities
(surface roughness and β^* are different)
- unaltered slope after a surface re-oxidation ($\text{Nb}_2\text{O}_5 \uparrow$ - 2 months later)
- exceptional flat slopes on BCP cavities C1-15 & C1-16
(in spite of roughness : 4 to 8 μm - higher β^*)

Thermal Feedback

Temperature Dependence of Surface Resistance

(V. Kurakin - EPAC'94 - London)

(E. Haebel - TTF Meeting - 1998)

$$H_S \rightarrow \Delta T = R_{therm} \Delta P_{diss} \propto R_S H_S^2 / 2 \rightarrow R_S(T) = R_S(T_0) + \frac{\partial R_S}{\partial T} \Delta T$$

$$R_S(T) = \frac{R_S(T_0)}{(1 - C \cdot E_{acc}^2)}$$

$$\Rightarrow Q_0 = G/R_S = (a - b \cdot E_{acc}^2)$$

$$R_{BCS} = A(\lambda_L, \xi_F, \ell) \frac{\omega^2}{T} e^{-\Delta/kT}$$

fit parameter :

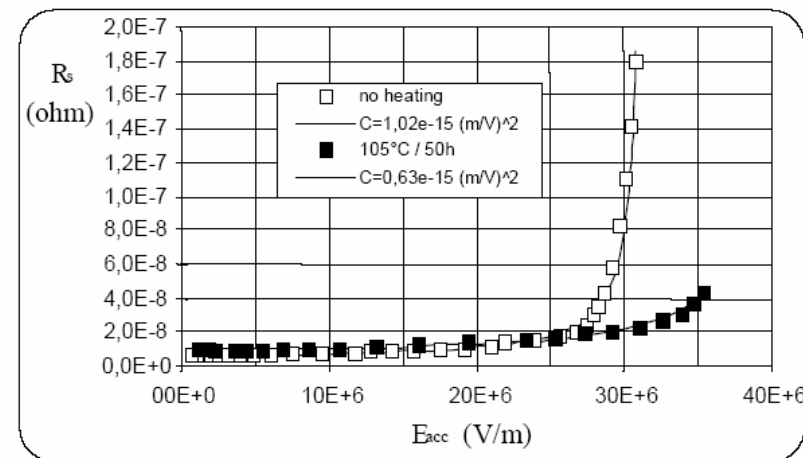
$$\begin{cases} C \approx 1.10^{-15} (V/m)^{-2} \\ C \approx 2.10^8 \frac{\partial R_S}{\partial T} \end{cases}$$

baking effect :

$$\frac{\partial R_S}{\partial T} (A \downarrow)$$

$$C \approx \frac{1}{2} \left(\frac{4 \cdot 10^{-9}}{\mu_0} \right)^2 \left(\frac{e_{Nb}}{\kappa_{Nb}} + \frac{1}{h_K} \right) \frac{\partial R_S}{\partial T}$$

$$\approx 2.10^9$$



(B.V. et al. SRF'99 - Santa Fe)

Energy Gap Dependence

Exponential variation

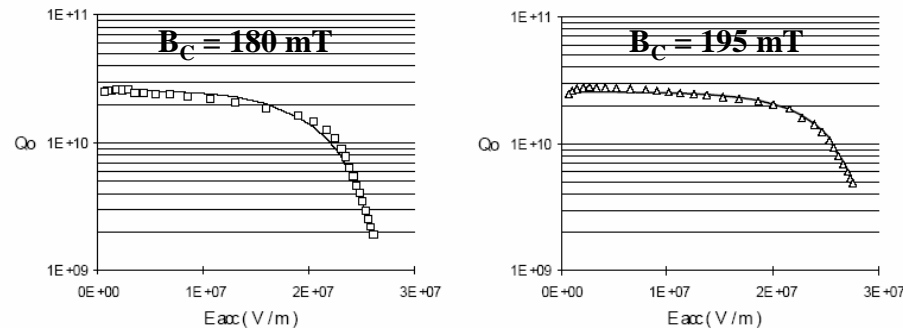
$$R_S = G/Q_0 = R_{res} + A(\lambda_L, \xi_F, \ell) \frac{\omega^2}{T} e^{-\Delta/kT}$$

magnetic field dependence of Δ ?

$$\Delta(H) = \Delta(0) \left(1 - H^2/H_C^2 \right) \quad \text{for } T/T_C < 0.36$$

(A. Didenko - EPAC ' 96 - Sitges)

(V. Mathur *et al.* - Phys. Rev. Let. 9, 374 - 1962)



B_C “fit factor”

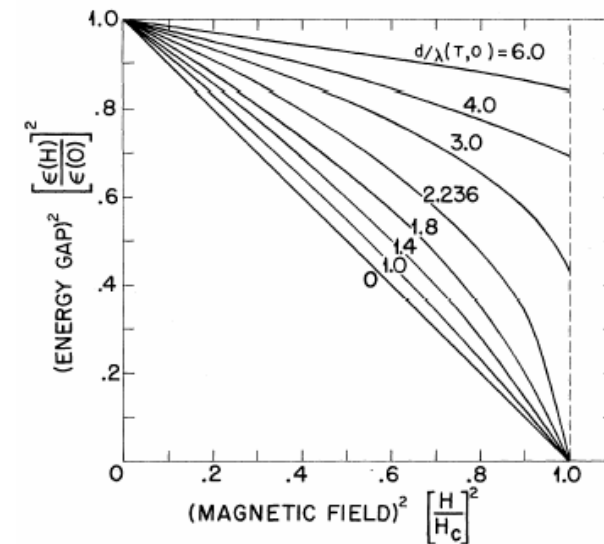
R_{res} , A , $\Delta(0)$ from $R_S(1/T)$ at low field

(B.V. *et al.* - EPAC ' 98 - Stockholm)

only rigorous and experimentally proved

for thin films

(normal state transition 2nd order if $d/\lambda < 5^{1/2}$)



for bulk material ($d \gg \lambda$)

$\Delta(H)$: few % variation

Granular Superconductivity

Grain Boundaries contribution to surface resistance ?

(polycrystalline nature of Nb) - Grain Boundary \equiv weak link (Josephson junction)

(B. Bonin and H. Safa - Superc. Sci. Tech. 4, 1991)

Theory valid for sputtered thin films
effect negligible for bulk niobium (grains $\sim 10 \mu\text{m}$) :
exception :
segregation of impurities located at grain boundaries

Difficulties to apprehend the baking as a way to clean G.B.

(low temperature - diffusion O)

Experiment on Grain Boundaries Specific Resistance

(H. Safa *et al.* - SRF '99 - Santa Fe)



Similarities EP / BCP

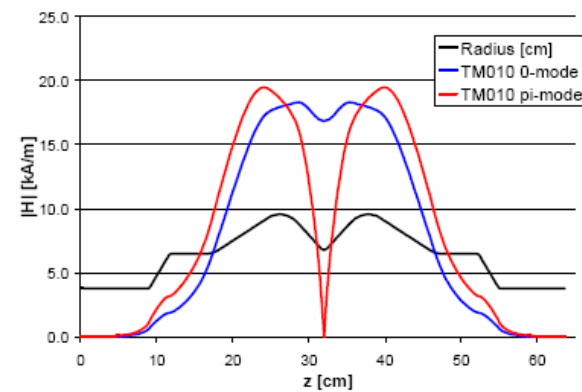
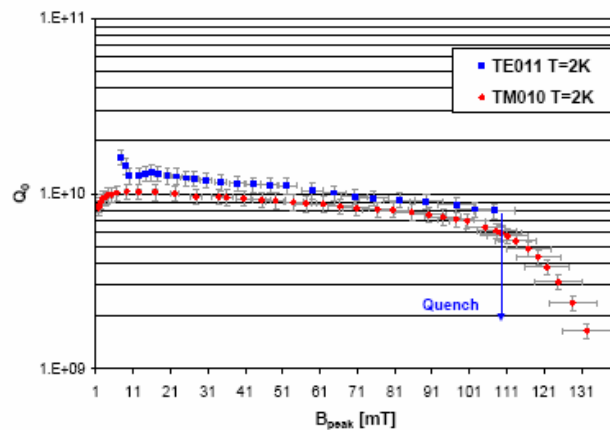
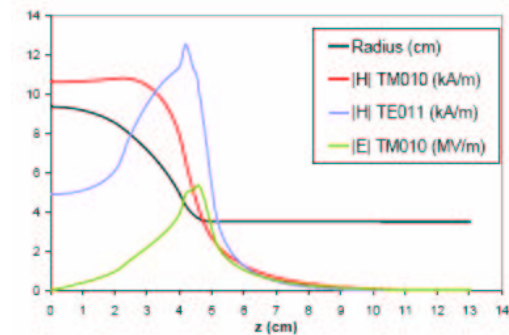
	Q-Slope Fit	Slope before baking (EP \equiv BCP)	Slope Improvement after baking	Slope after baking (EP < BCP)	No change after 2 m. air exposure	Exceptional Results (BCP)	Quench (EP > BCP)	BCP Quench unchanged after baking	Validity
Magnetic Field Enhancement	Y	N (β_m et $H_C \neq$)	Y ($H_C \uparrow$)	Y ($\beta_m < ; H_C >$)	-	N (high β_m)	Y ($\beta_m < ; H_C >$)	N ($H_C \uparrow$)	Y
Interface Tunnel Exchange	Y (E^8)	N ($\beta^* \neq$)	Y ($Nb_2O_{5-y} \downarrow$)	Y (low β^*)	N ($Nb_2O_{5-y} \uparrow$)	N (high β^*)	-	-	Y
Thermal Feedback	Y (parab.)	Y	Y ($R_{BCS} \downarrow R_{res} \uparrow$)	N	-	N	-	-	N (coeff. C)
Magnetic Field Dependence of Δ	Y (expon.)	N ($H_C \neq$)	Y ($H_C \uparrow$)	Y ($H_C >$)	-	N	-	-	N (thin film)
Segregation of Impurities	-	N (\neq segreg.)	N (only O)	-	-	Y (cleaning)	-	-	Y

Experimental Program at J.Lab.

" H enhancement at grain bound. " \Leftrightarrow " Interface Tunnel Exchange "

- Single cell cavity (BCP - EP - w/o Baking) excited in modes TM_{010} or TM_{011} : H_S
- Two cell cavity TM_{010} ($0-\pi$ mode) : scan the surface (E, H)
- Q_0 (B_{peak}) - (BCP - EP - w/o Baking)
- Preliminary results :

(G. Ciovati *et al.* - PAC ' 2003 - Portland)

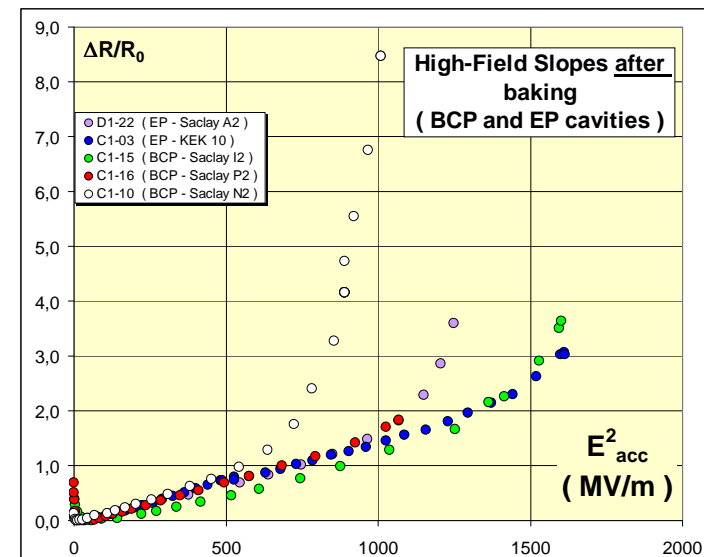
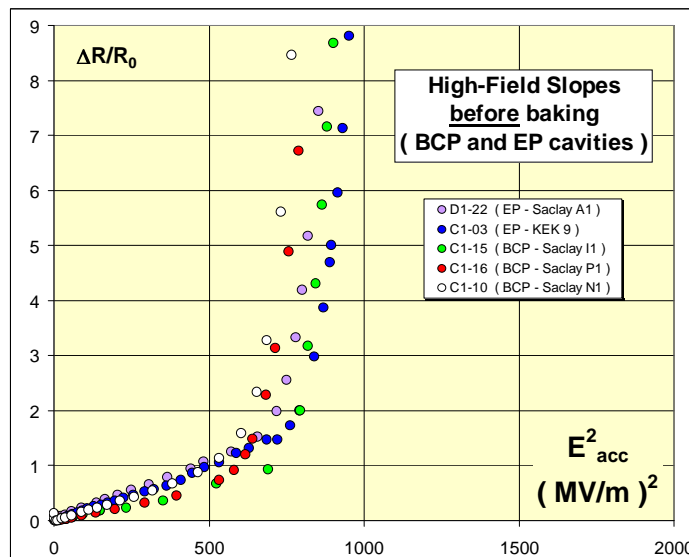


Conclusion

Experiments and Models have to progress in the future
to understand the phenomenon and to cure it

all the more so since

Q-slope is not totally removed by baking (even for EP cavities)



if quench improvement possible (> 40 MV/m) \Rightarrow Q-slope appearance

Papers at this Workshop

□ Q-slope

A Review of High-Field Q-Slope Studies at Cornell - H. Padamsee *et al.* (**Mo-P14**)

Why does the Q-slope of a Nb cavity change...? - I.V. Bazarov *et al.* (**Th-P02**)

Q-slope analysis of niobium SC RF cavities - K Saito (**Th-P19**)

Q-Slope : Comparison BCP and EP - Modification by Plasma... - B. Visentin *et al.* (**Mo-P19**)

□ Baking

A Pleasant Surprise: Mild Baking Gives Large Improvement... - G. Ereemeev *et al.* (**Mo-P18**)

Low temperature heat treatment effect on high-field EP cavities - J. Hao *et al.* (**Mo-P16**)

Effect of low temperature baking on niobium cavities - G. Ciovati *et al.* (**We-O14**)

□ Surface Analysis

In situ XPS investigation of the baking effect ... - K Kowalski *et al.* (**Th-P09**)

Near-Surface Composition of Electropolished Niobium ... - A.M. Valente *et al.* (**Mo-P15**)

Grain boundary specific resistance and RRR measurements... - S. Berry *et al.* (**Th-P03**)