

Advances in SRF Material Science aimed at High Q cavities



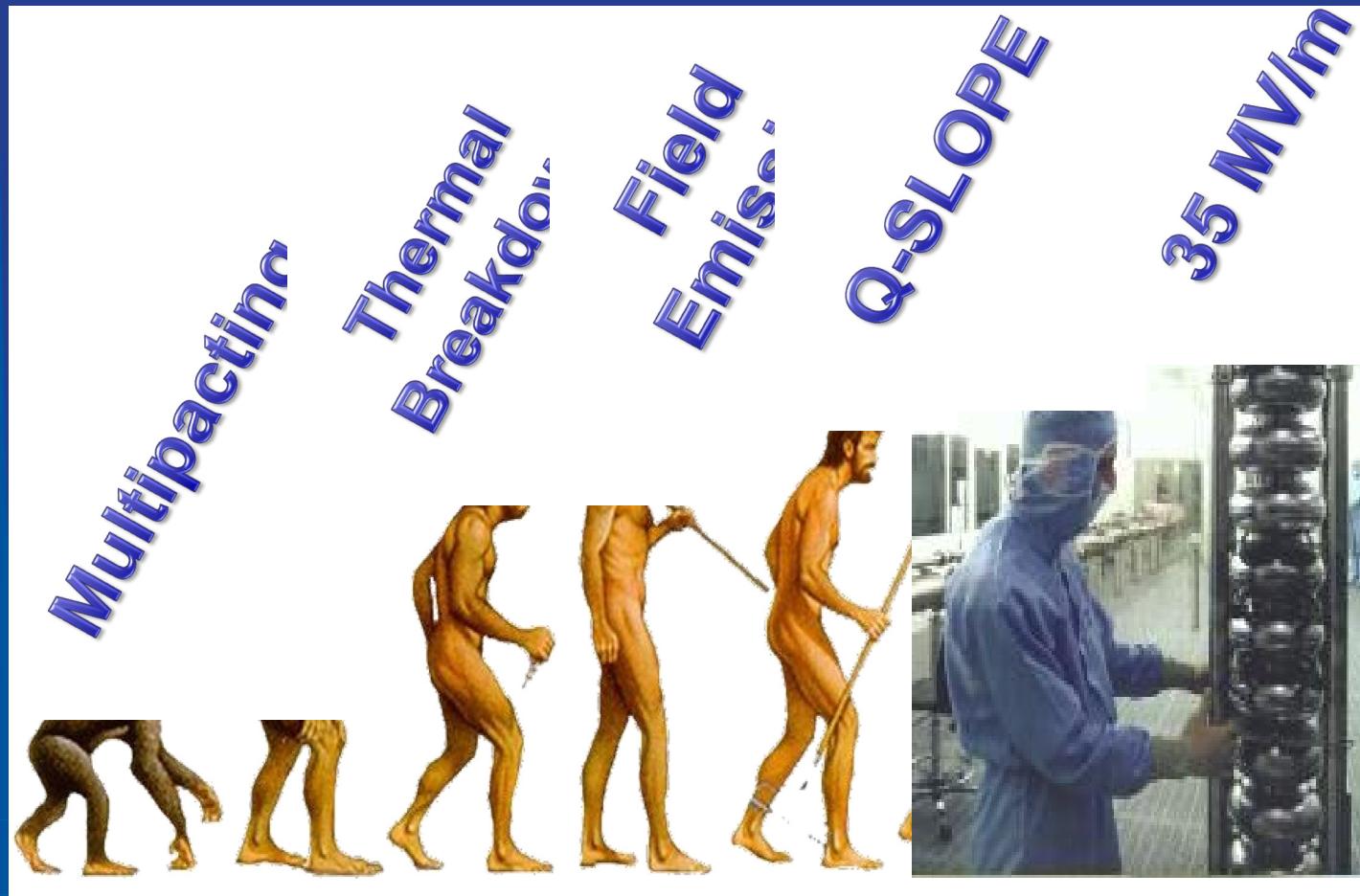
Anna Grassellino
SRF Development Department, TD
Fermilab

NA-PAC 2013
Pasadena, California

Outline

- Motivation for pushing for higher quality factors
- New insights on understanding of origin and causes of RF losses
 - *Hydrides formation*
 - *Flux trapping*
 - *Surface resistance decomposition*
- New surface processing techniques for high Q cavities
 - *Doping with small amount of interstitial impurities*
- Q ‘preservation’ in a real machine
- Outlook and conclusions

SRF evolution (courtesy of H. Padamsee)



*Steady progress due to basic understanding of limiting phenomena
and invention of effective cures*

SRF core technology for accelerators worldwide

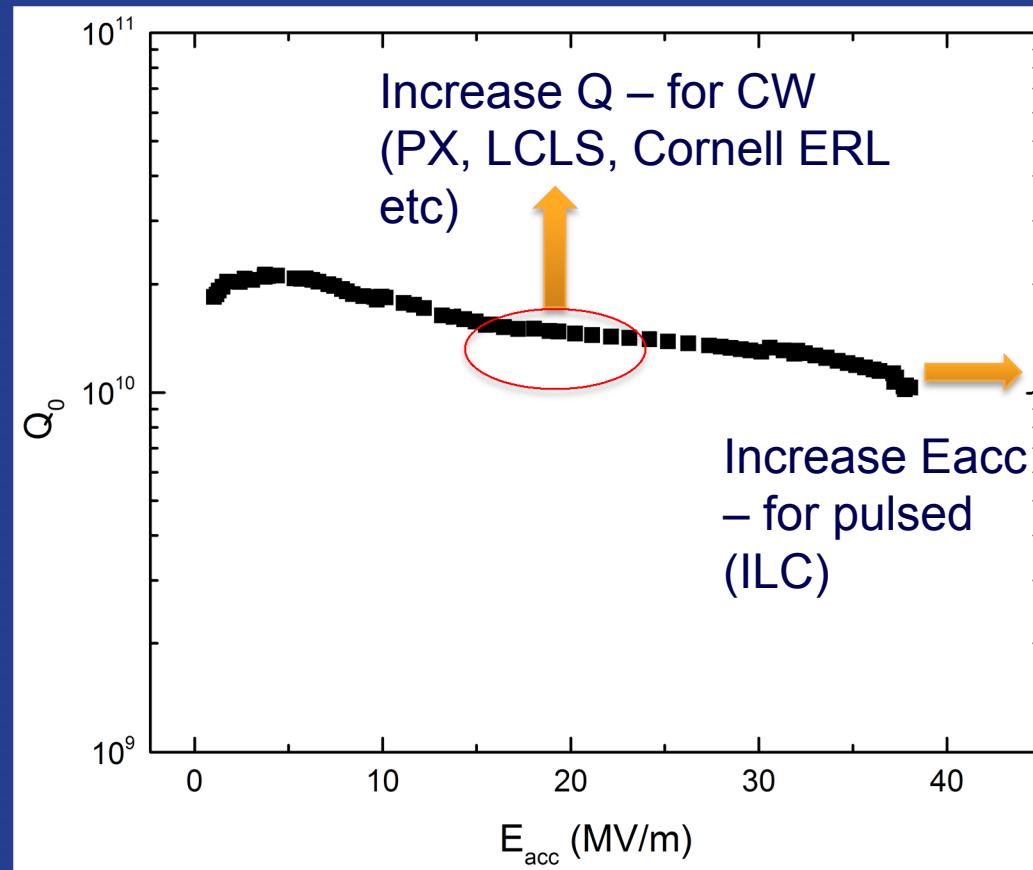
- Low energy nuclear physics, for nuclear shape, spin, vibration... – Heavy ion linacs
- Medium energy nuclear physics, structure of nucleus, quark-gluon physics
 - Recirculating linac
- Nuclear astrophysics, for understanding the creation of elements
 - Facility for rare isotope beams (FRIB)
- X-Ray Light Sources for life science, materials science & engineering
 - Storage rings, free electron lasers, energy recovery linacs
- Spallation neutron source for materials science and engineering, life science, biotechnology, condensed matter physics, chemistry
 - High intensity proton linac
- Future High Intensity Proton Sources for
 - Nuclear waste transmutation, energy amplifier, power generation from Thorium
- High energy physics for fundamental nature of matter, space-time
 - Electron-positron storage ring colliders, linear collider, proton linacs for neutrinos

Need for higher Q

$$Q_0 = \frac{\omega_0 \mu_0 \int_V |\mathbf{H}|^2 dv}{R_s \int_S |\mathbf{H}|^2 ds}.$$

$$Q_0 = \frac{G}{R_s},$$

$$G = \frac{\omega_0 \mu_0 \int_V |\mathbf{H}|^2 dv}{\int_S |\mathbf{H}|^2 ds}$$



- Increasing Q of cavities – very important for high duty factor accelerators – virtually all planned superconducting machines (e.g. Project X, LCLS-II, ERLs...)
- Capital and operational costs scale with dissipated power $\sim 1/Q$

*How to approach the problem?
What to look for?*

The ‘macroscopic’ does not always matter

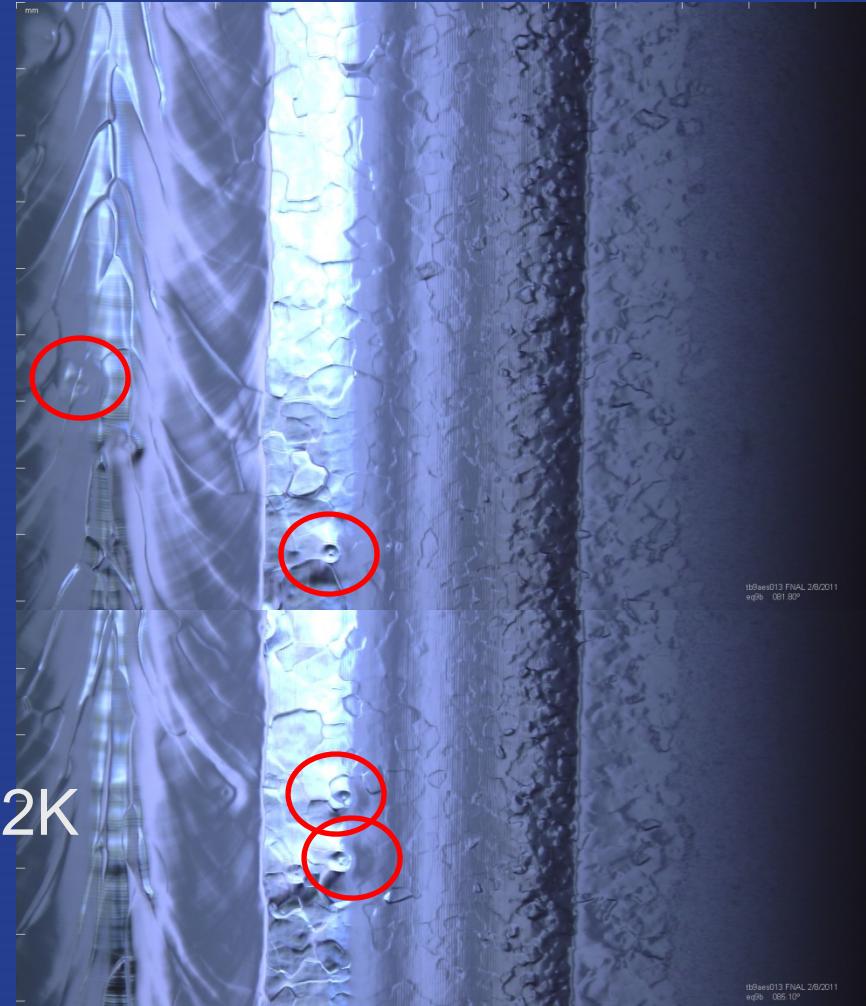
Typical surface processing sequence:

100-200 µm EP/BCP + 800°C + light EP/BCP + 120°C

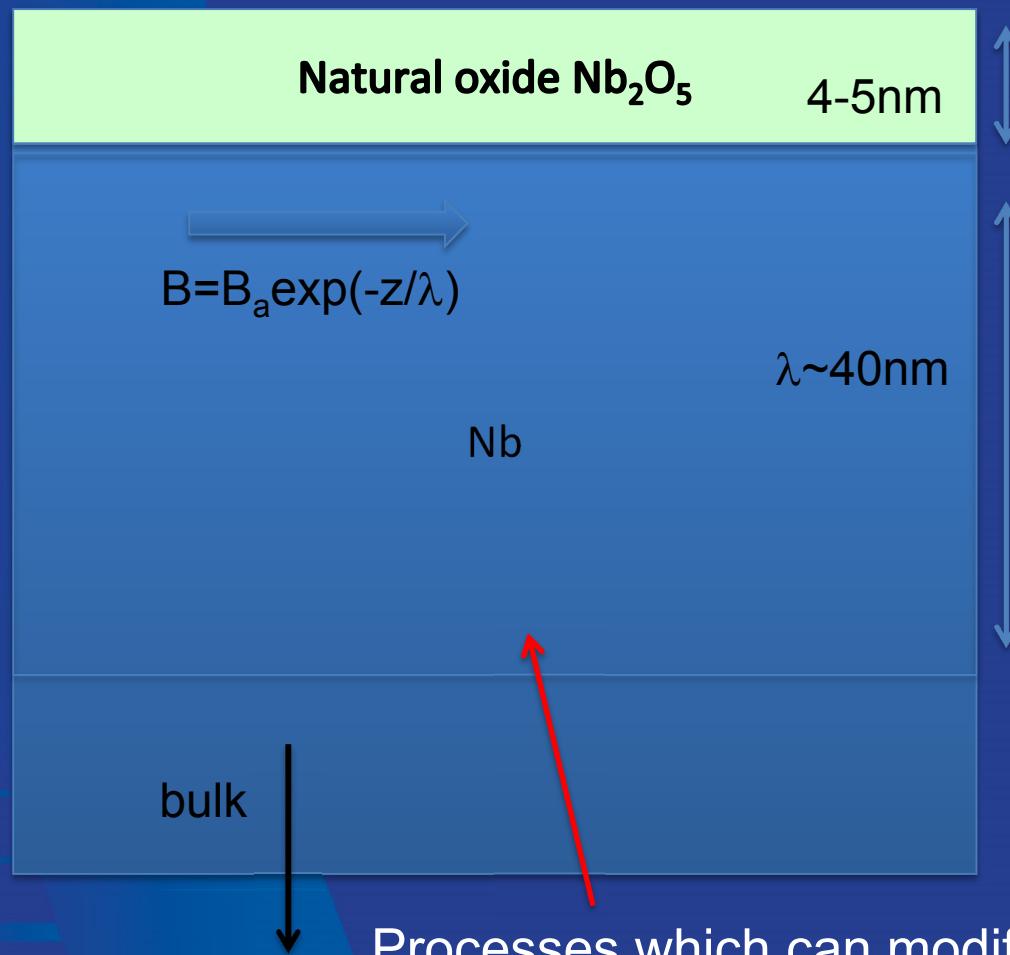


Bad looking surfaces, normal performance:

- Swirl cavity, Q above $2e10$ @ 2K
- Pits cavity, 38 MV/m



RF surface resistance of a superconductor: the microscopic is what mostly matters

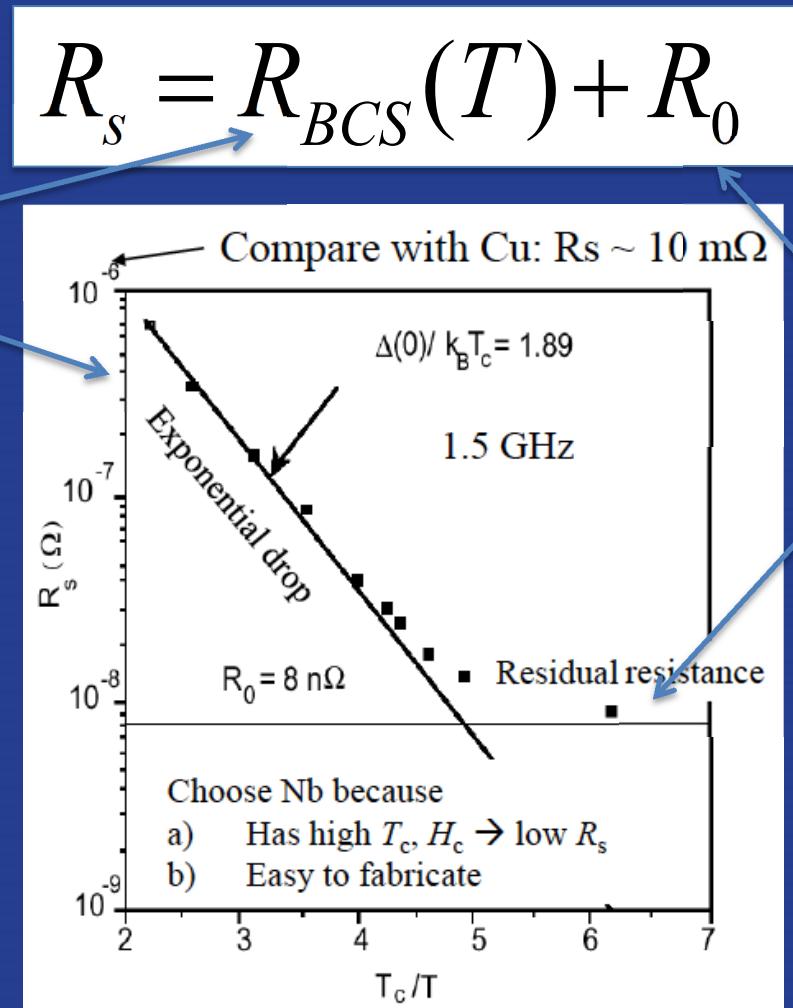


- Supercurrents flow in the thin London layer to screen the SC bulk from magnetic field
- In DC case currents flow with zero resistance but in RF case - due to the inertia of the Cooper pairs – electric field is present in the London layer causing losses
- Recent studies (Romanenko et al – [Phys. Rev. ST Accel. Beams 16, 012001 \(2013\)](#)) show that the first few nanometers strongly affect the Q value!

Processes which can modify this layer in a controlled manner are of special interest

Nb microwave surface resistance

BCS resistance:
'unpaired' electrons
at a finite T:
• Scales as f^2
• Depends on
mean free path,
gap, critical T,
several
microscopic
parameters
(London depth,
coherence
length)



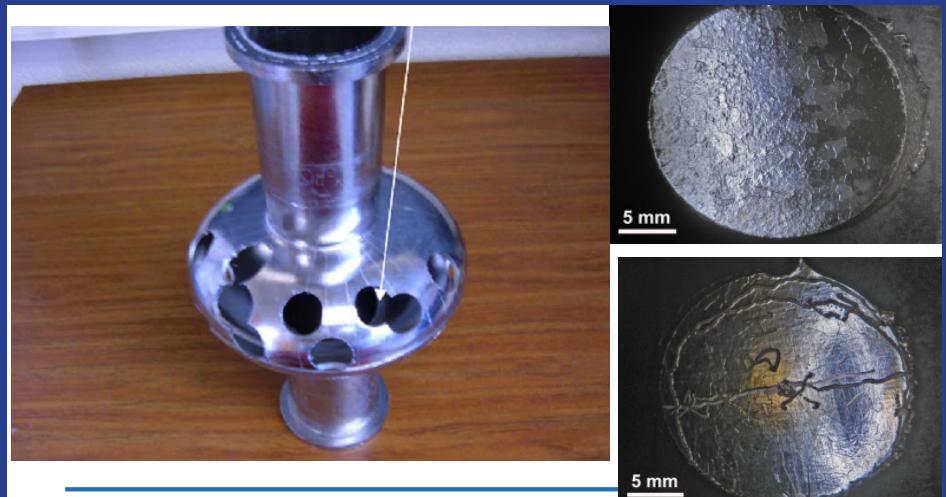
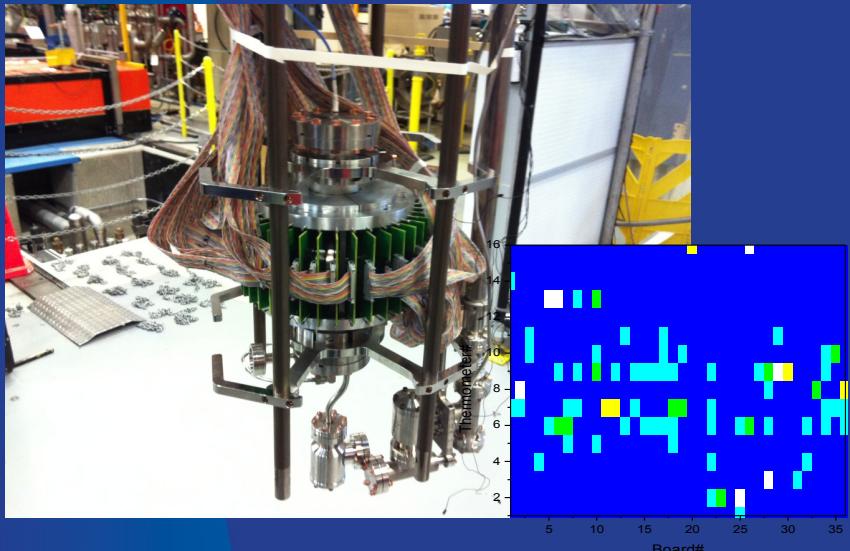
Residual
resistance:
T independent
component, known
and unknown
contributions:

- Scales as \sqrt{f}
- Trapped flux
- Hydrides

New insights on understanding of origin and causes of RF losses

Techniques employed to gain understanding of origin of RF losses

- Experiments on cavities – eg $R_s(B)$ deconvolution as a function of surface treatments, T-mapping...
- Elemental and microstructural investigations: SIMS, XPS, XRD, PAS, ERD, **cryogenic laser confocal microscopy**...
- Superconducting properties studies: magnetization, muon spin rotation, **Bitter decoration**, magneto optical imaging, point contact tunneling...



Causes of RF losses: hydrides

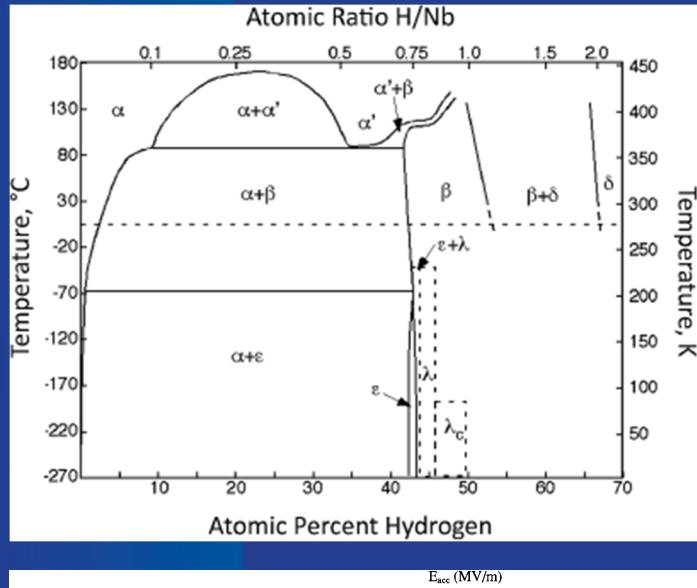


FIGURE 5. Q_0 curves of a cavity, once measured after a fast cooldown and once after a slow cooldown[11].

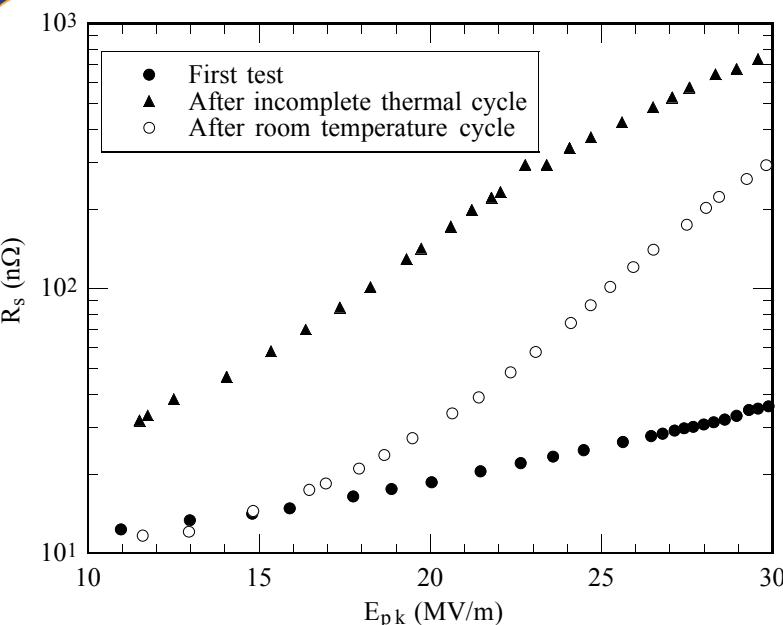
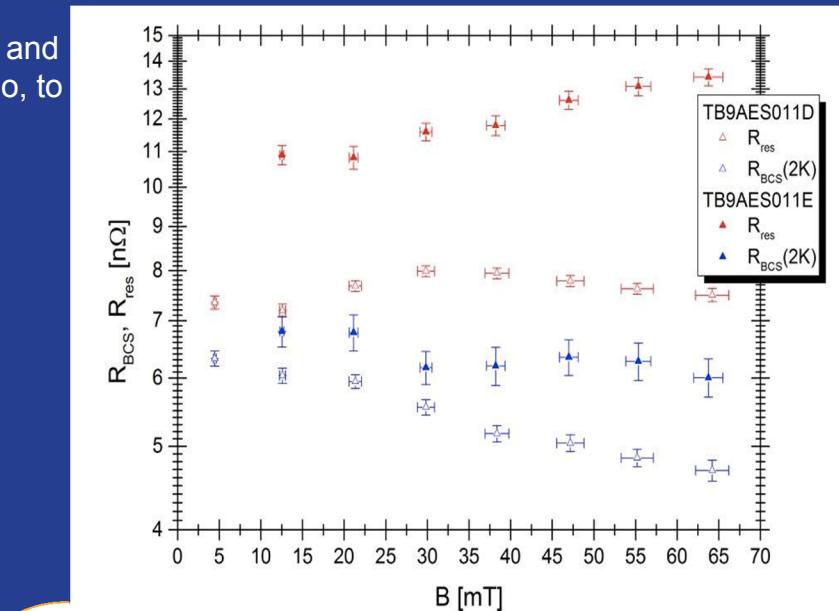
1980's, reactor-grade material ($\text{RRR} = 40$) had been used. Then there was a strong push to improve the quality of as-delivered niobium and to post-purify cavities during a 1400°C bakeout combined with solid-state gettering[8, 9, 10]. In that manner RRR values were increased up to 500 and, as predicted, achievable gradients were also improved substantially[2]. However, the use of high-purity niobium also lead to the discovery of a new anomalous loss mechanism—the so-called "Q disease."

THE Q DISEASE

Discovery

A significant advantage of a vertical test stand as in Figure 2 lies in the ability to

M. Checchin and
A. Grassellino, to
be published



Knobloch and Padamsee, 8th
Workshop on RF
Superconductivity, Padova,
Italy. SRF 981012-12

Causes of RF losses: hydrides

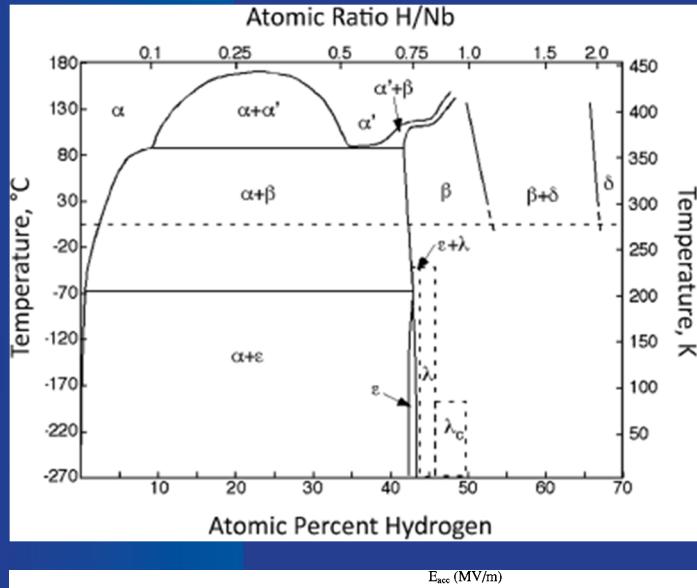


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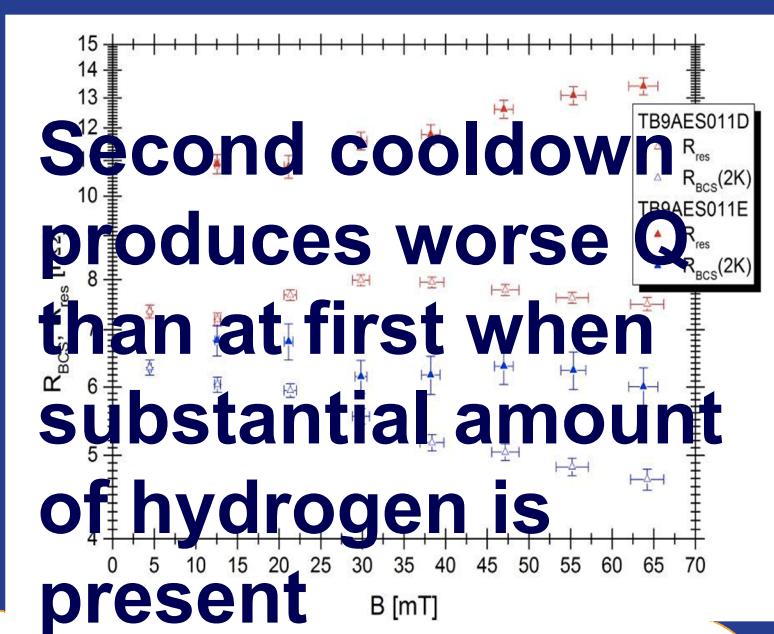
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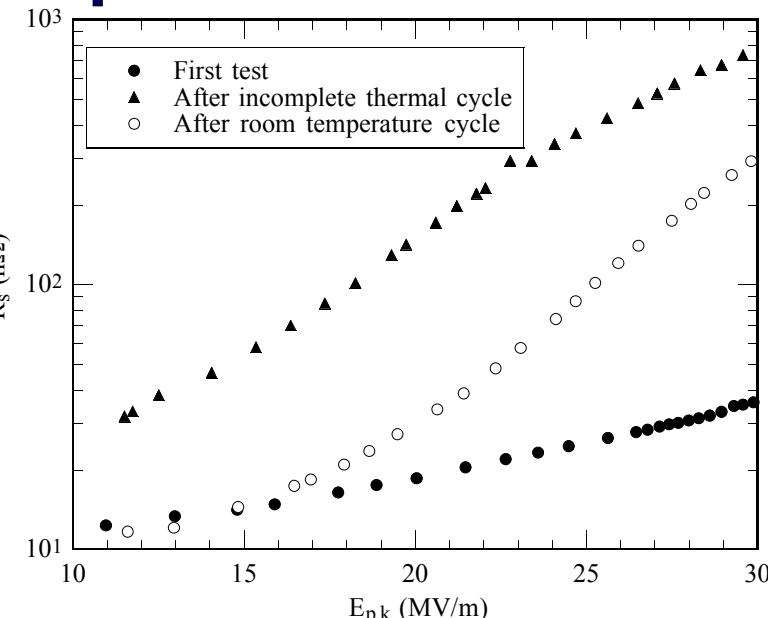
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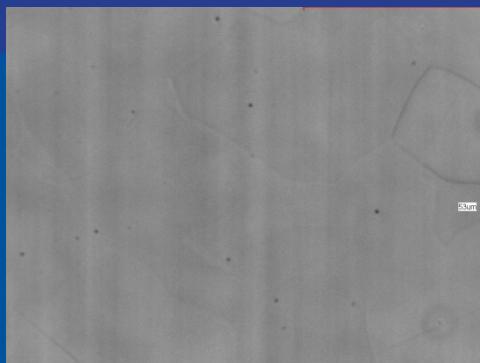
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Second cooldown produces worse Q than at first when substantial amount of hydrogen is present



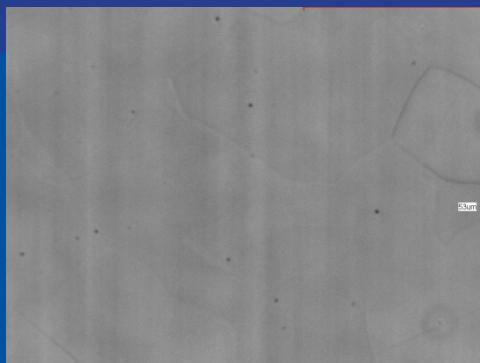
Cryogenic Laser Confocal Microscopy by A. Romanenko et al: First cooldown



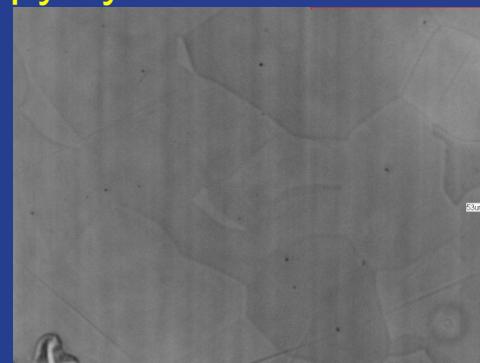
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F. Barkov, A. Romanenko, and A. Grassellino,
Phys. Rev. ST Accel. Beams, 15 122001 (2012)
F. Barkov, A. Romanenko, A. Grassellino,
Proceedings of SRF'2013, TUP014

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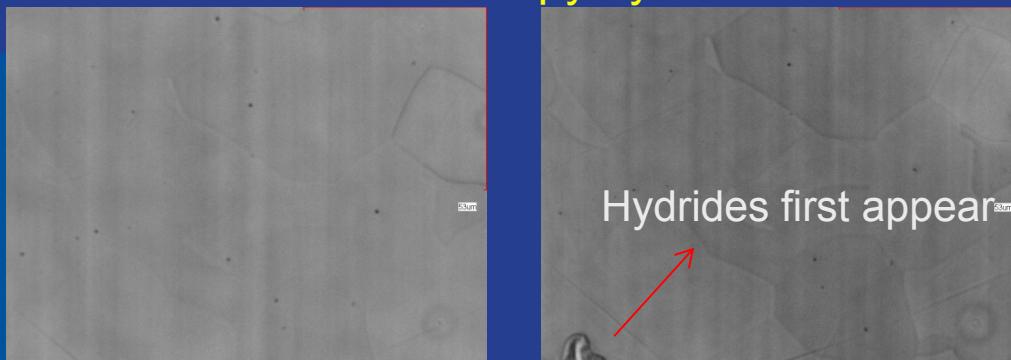
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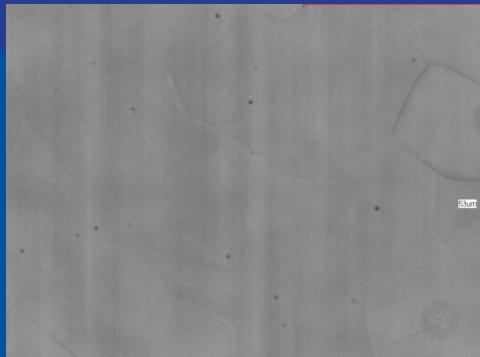
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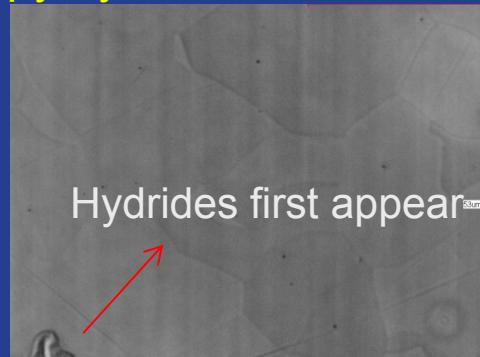


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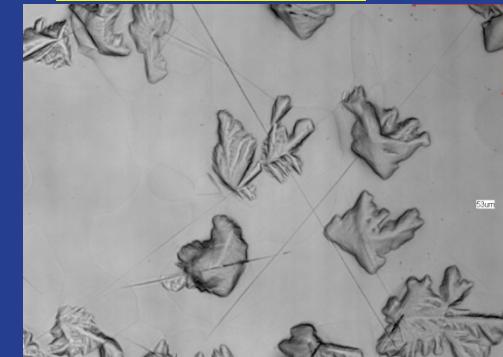
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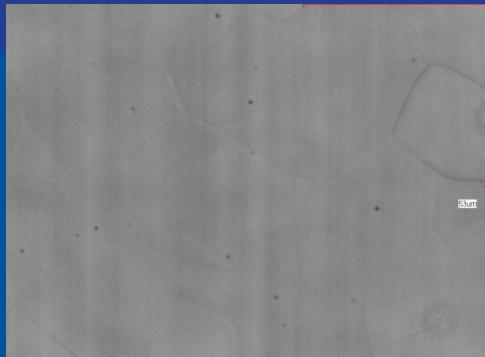


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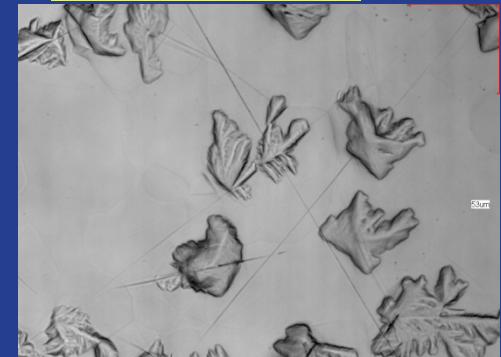
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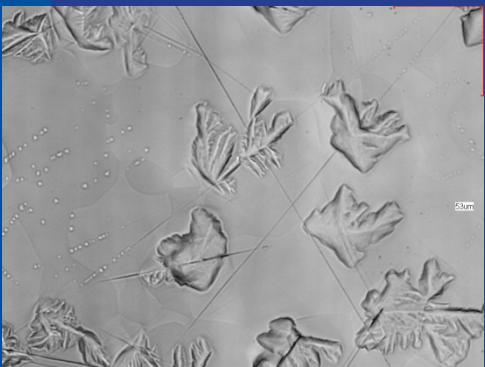
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T=160K



T=140K

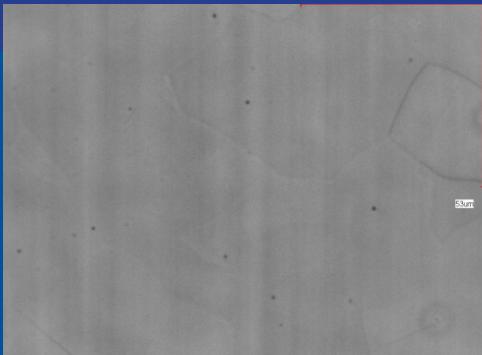


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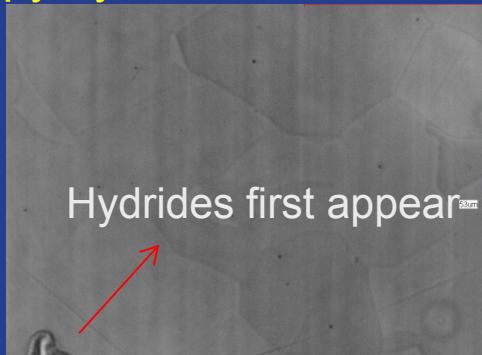
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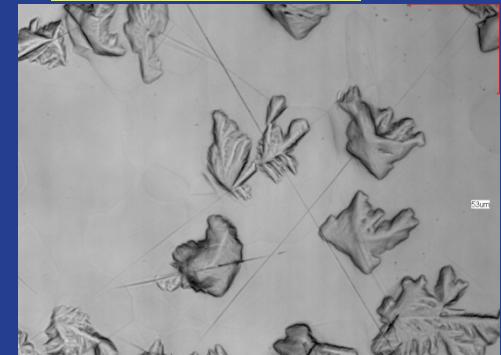
50 μm



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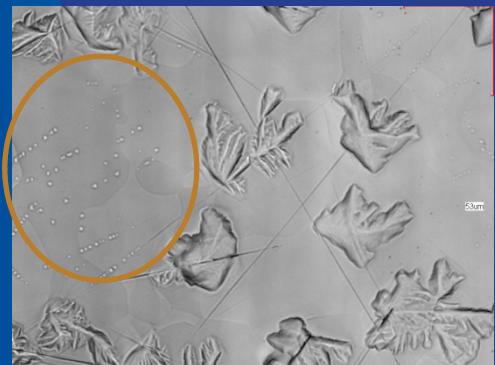


T=160K



T=140K

Second
(smaller)
phase of
hydride
forms

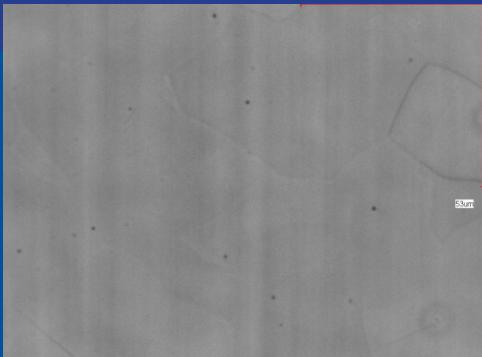


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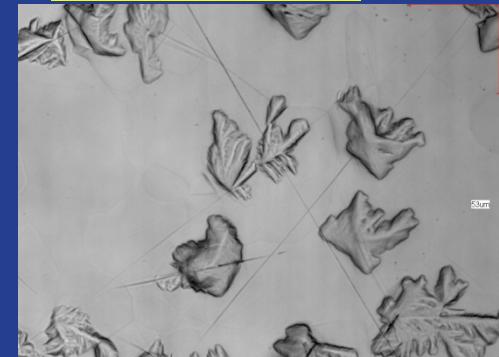
50 μm



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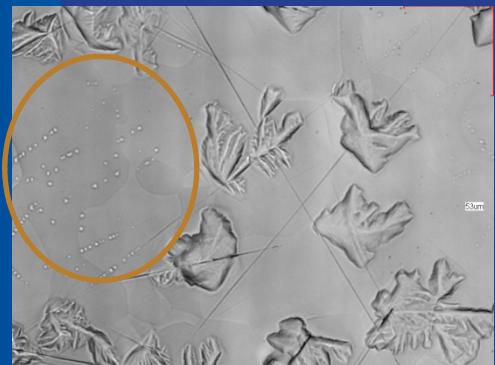


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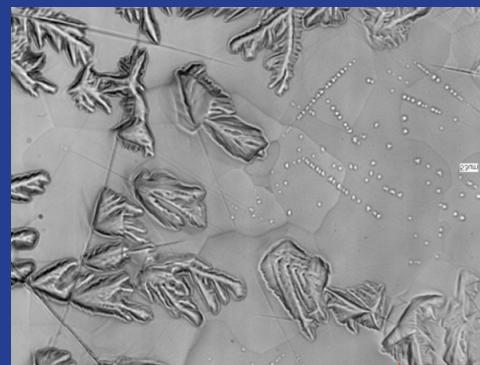


T=140K

Second
(smaller)
phase of
hydride
forms



T=100K

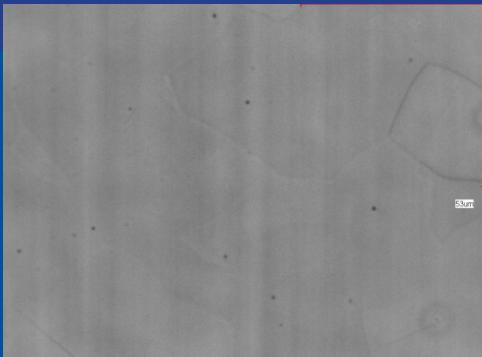


T=6K

F. Barkov, A. Romanenko, and A. Grassellino,
Phys. Rev. ST Accel. Beams, 15 122001 (2012)
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Cryogenic Laser Confocal Microscopy by A. Romanenko et al: First cooldown

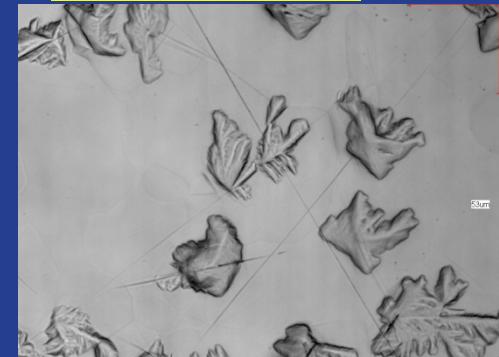
50 μm



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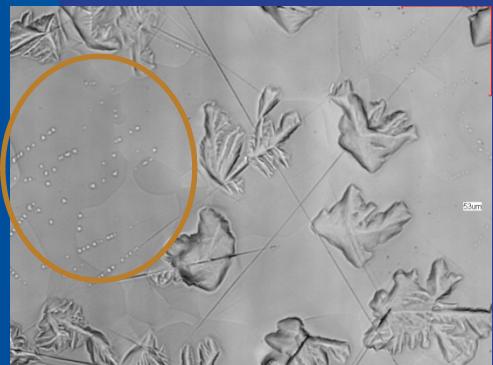


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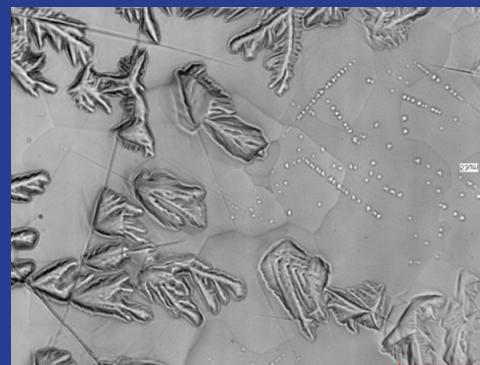


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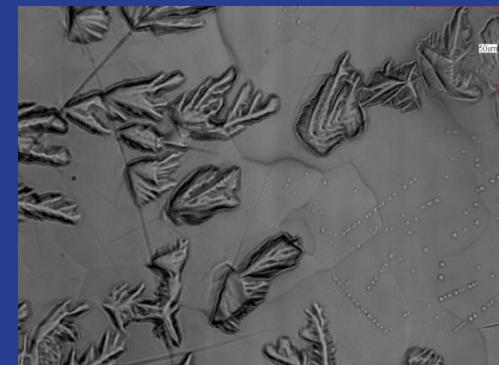
Second
(smaller)
phase of
hydride
forms



T=100K



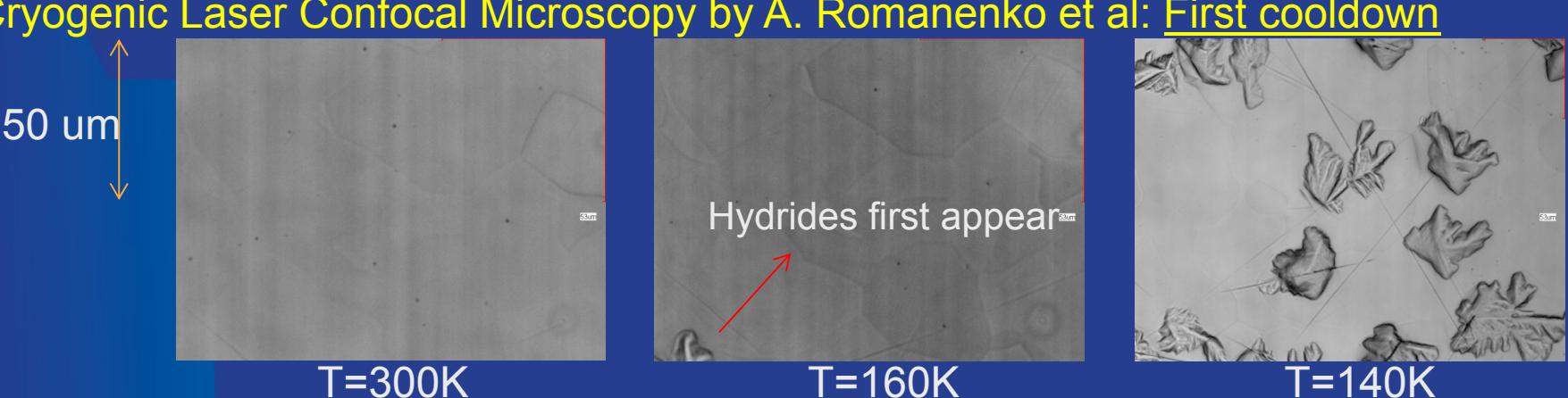
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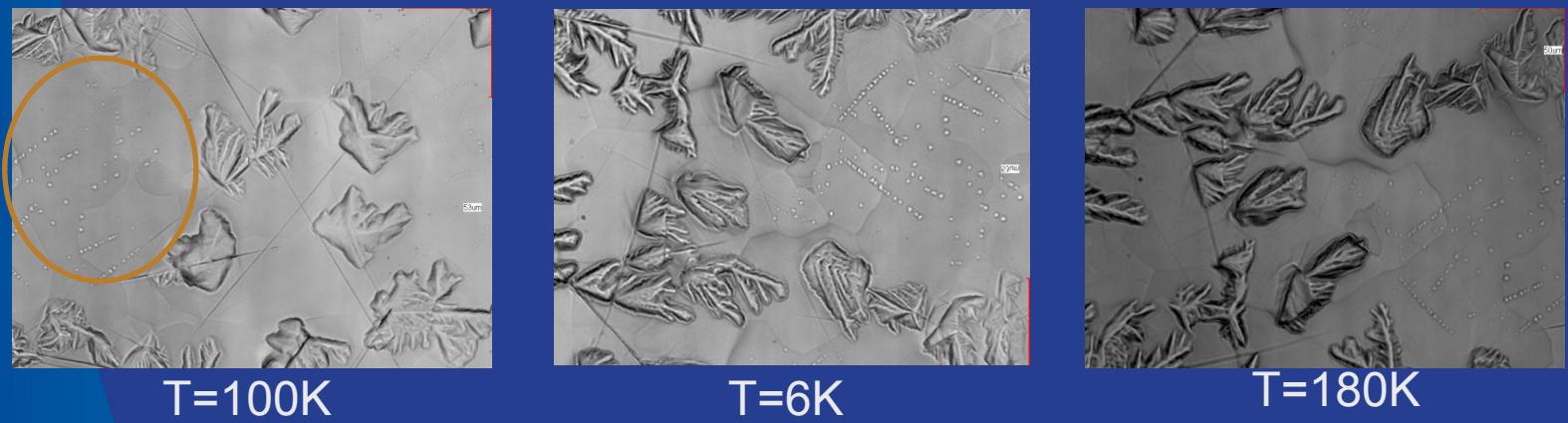
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Cryogenic Laser Confocal Microscopy by A. Romanenko et al: First cooldown

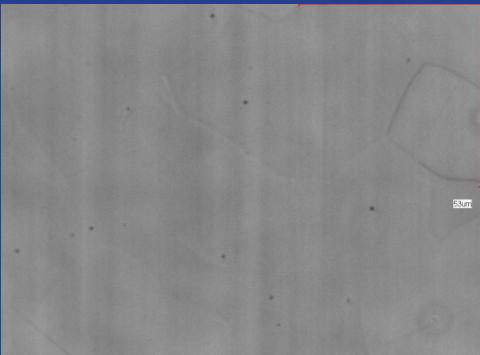


Second
(smaller)
phase of
hydride
forms



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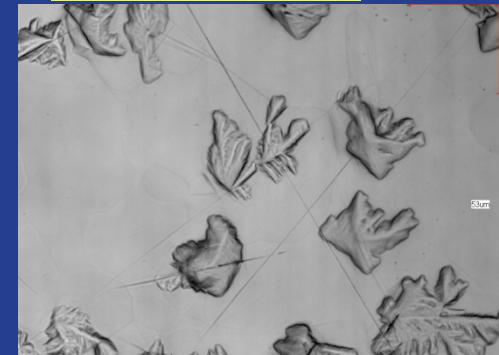
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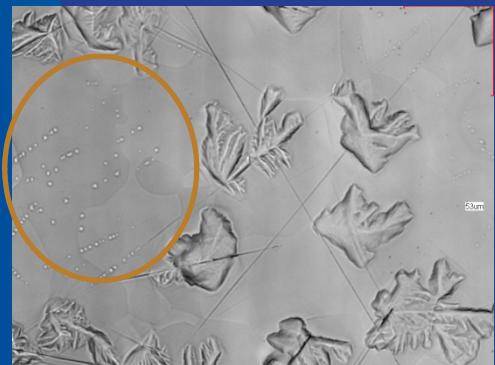


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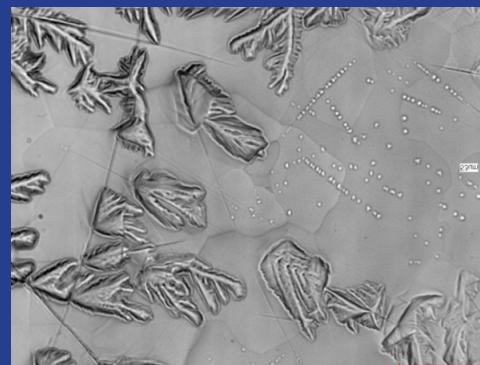


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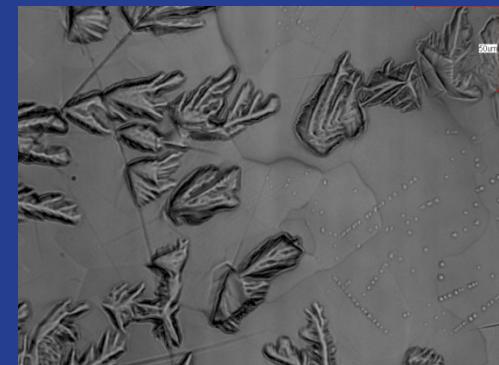
Second
(smaller)
phase of
hydride
forms



T=100K

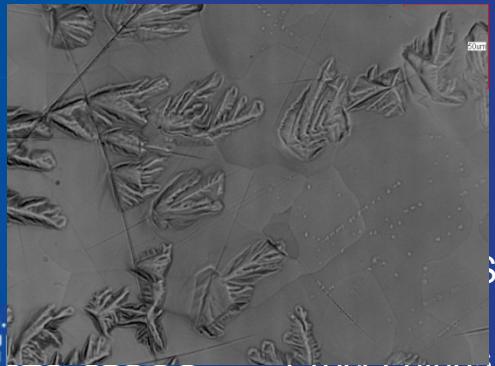


T=6K



T=180K

Large
phase
starts to
dissolve

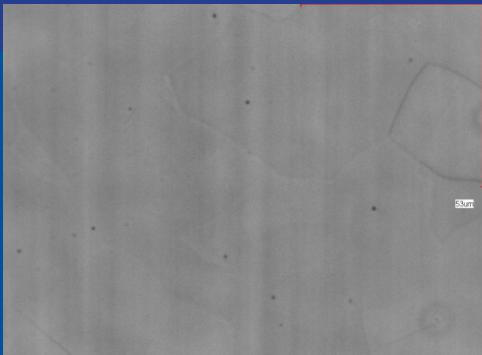


T=210K

Romanenko, and A. Grassellino,
NST Accel. Beams, 15 122001 (2012)
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Cryogenic Laser Confocal Microscopy by A. Romanenko et al: First cooldown

50 μm

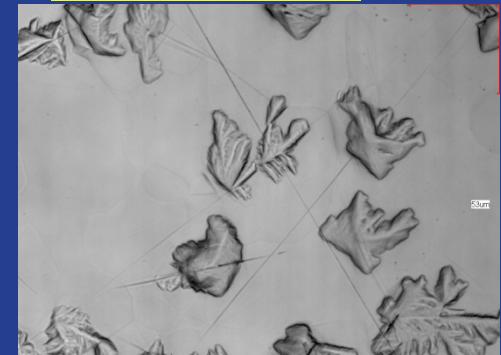


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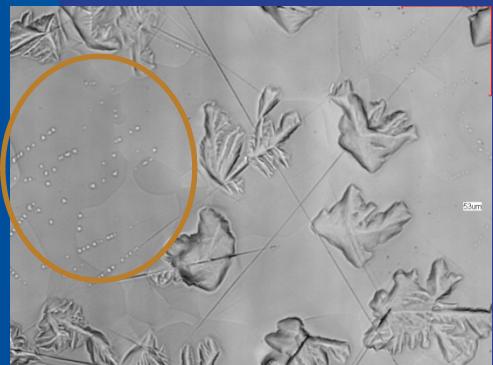
Hydrides first appear

T=160K

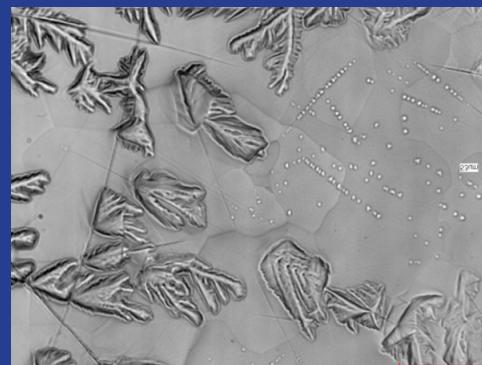


T=140K

Second
(smaller)
phase of
hydride
forms



T=100K

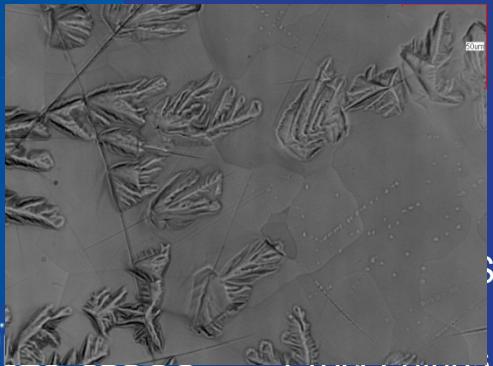


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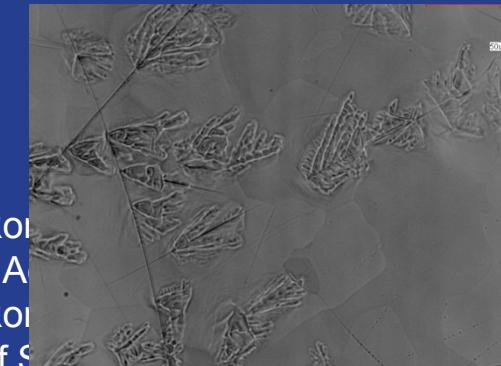


T=180K

Large
phase
starts to
dissolve



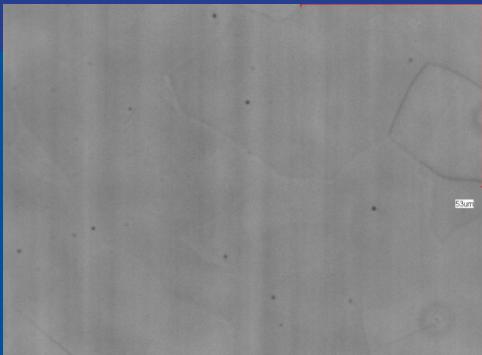
T=210K



T=260K

Cryogenic Laser Confocal Microscopy by A. Romanenko et al: First cooldown

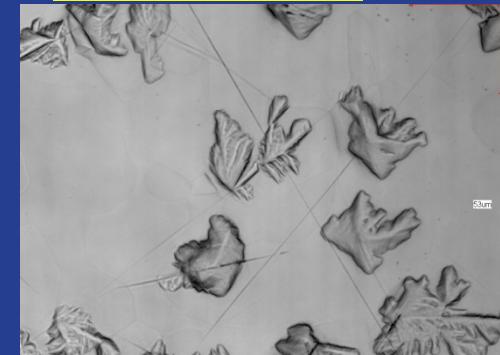
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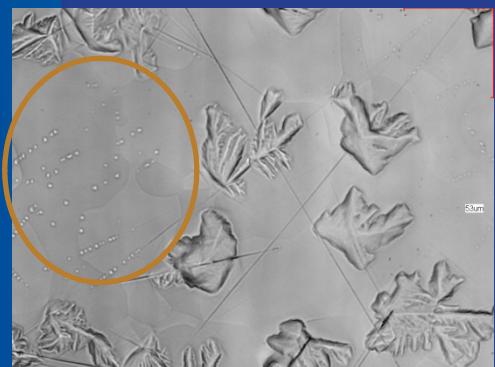
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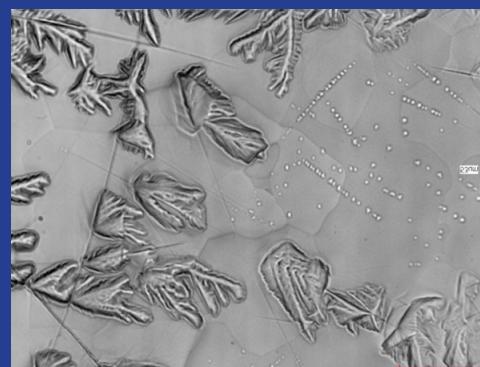
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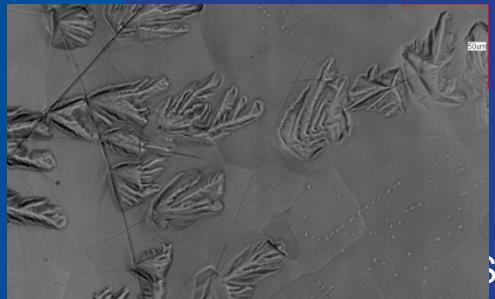
Second
(smaller)
phase of
hydride
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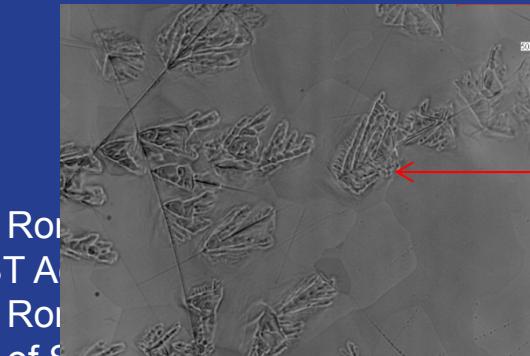
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Large
phase
starts to
dissolve



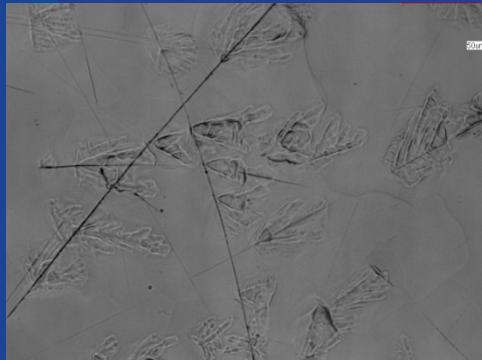
T=210K



T=260K

Hydrides gone,
dislocation skeleton
(deformation) remains
on the surface

Cryogenic Laser Confocal Microscopy by A. Romanenko et al : Second cooldown

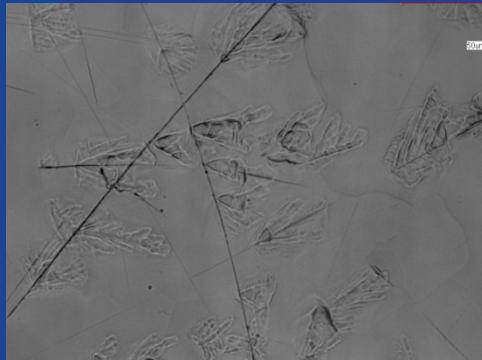


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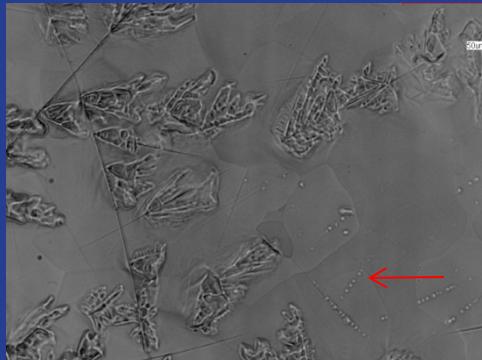


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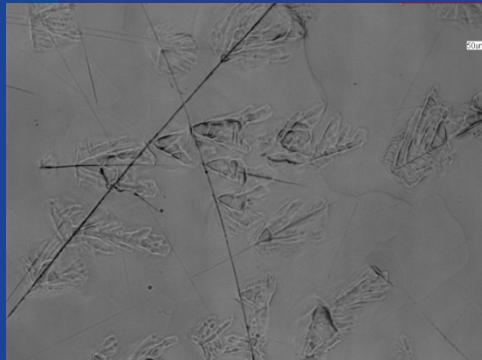
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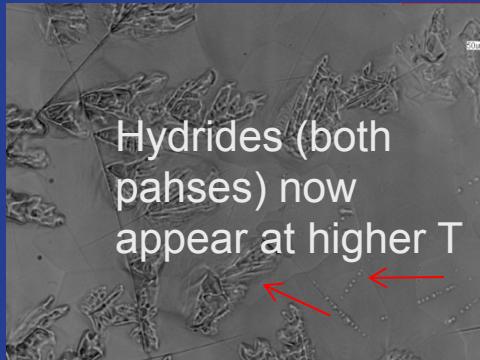
190K

F. Barkov, A. Romanenko, and A. Grassellino,
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Cryogenic Laser Confocal Microscopy by A. Romanenko et al : Second cooldown



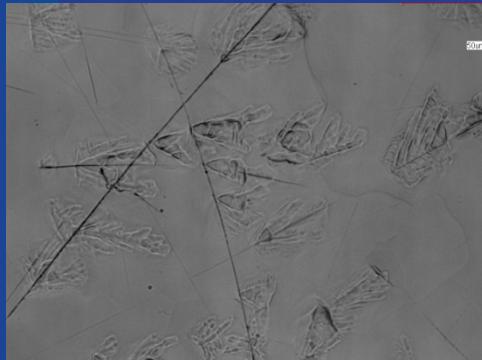
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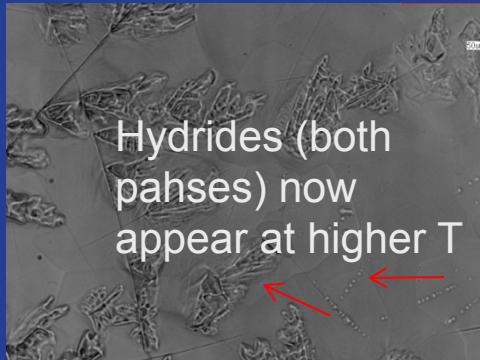
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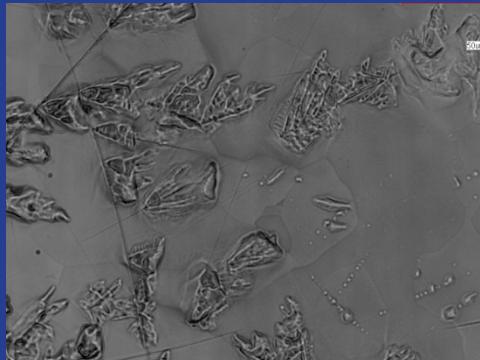
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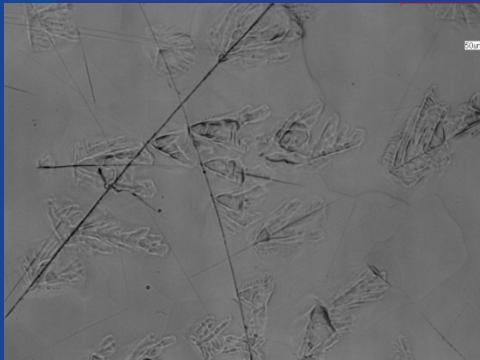
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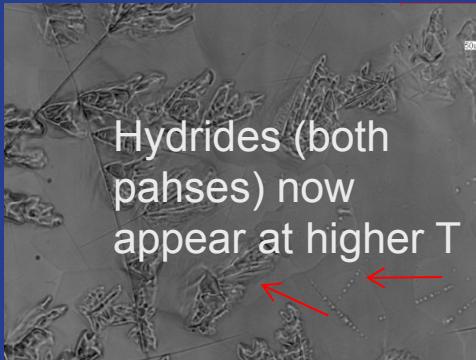
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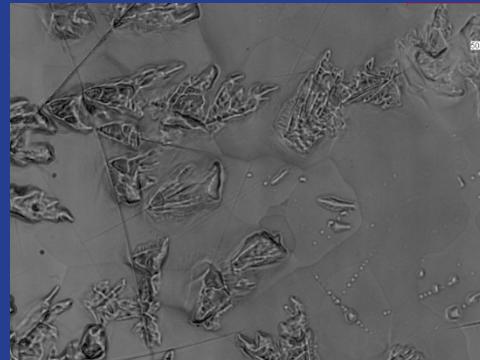
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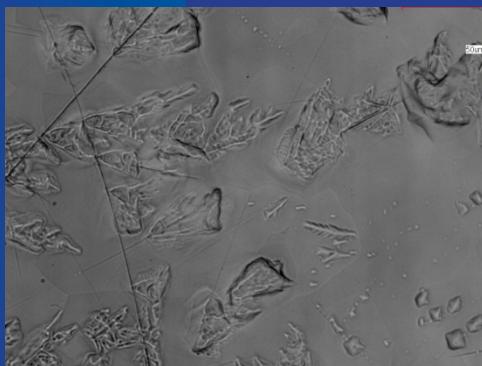
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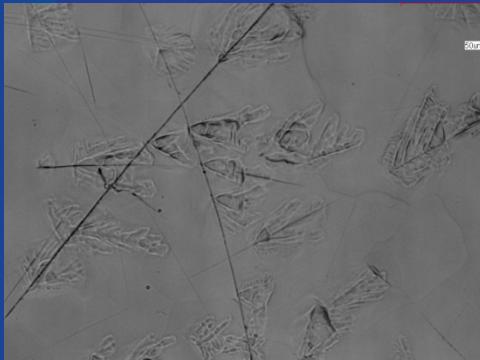
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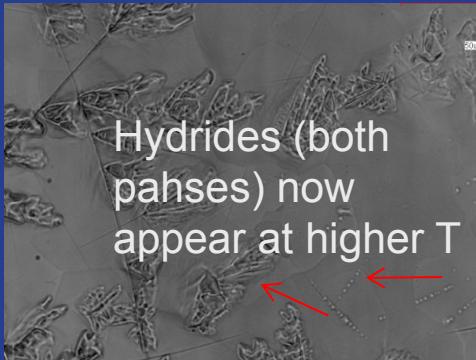
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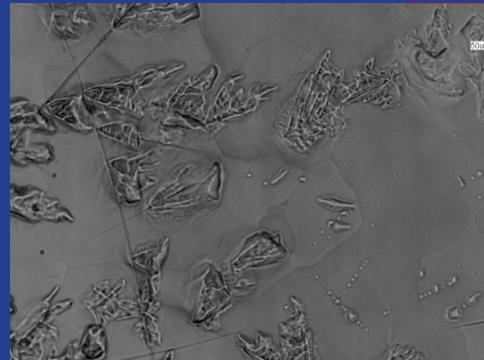
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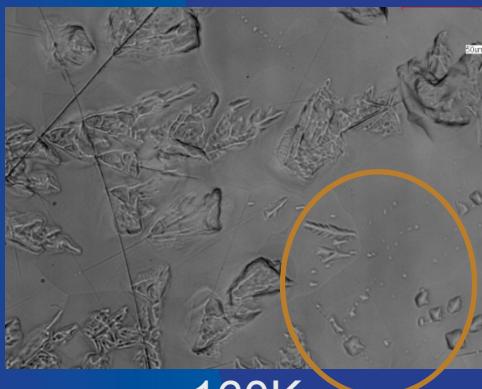
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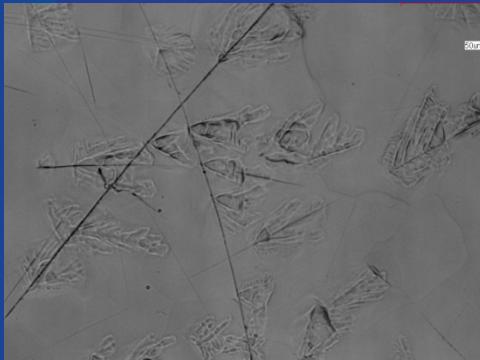
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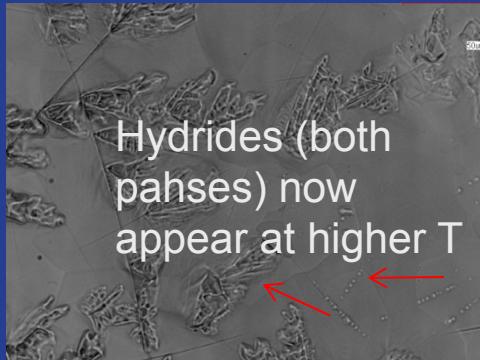
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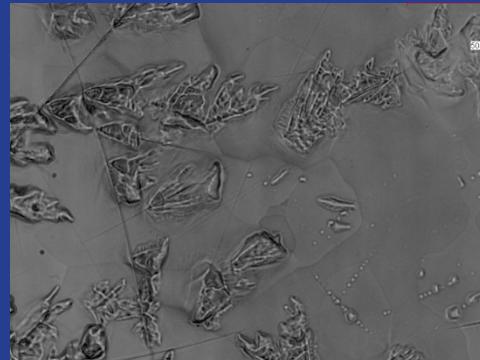
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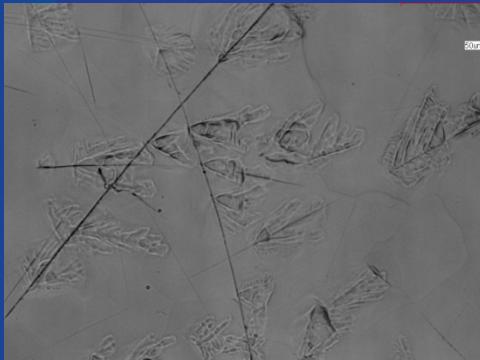
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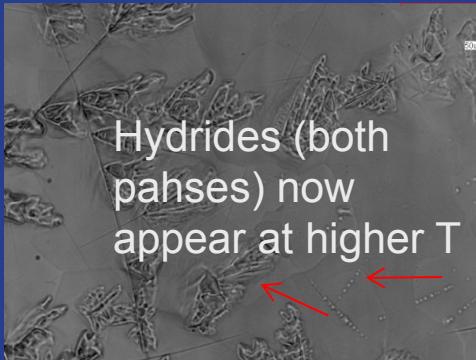
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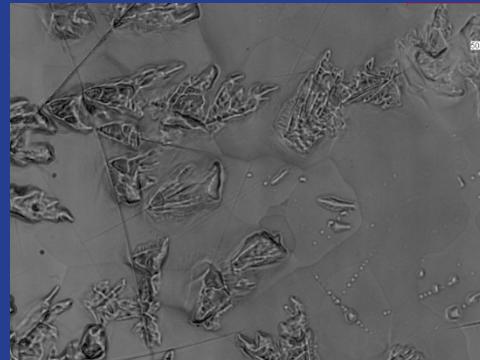
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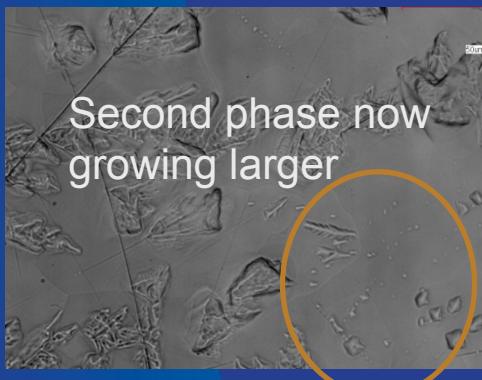
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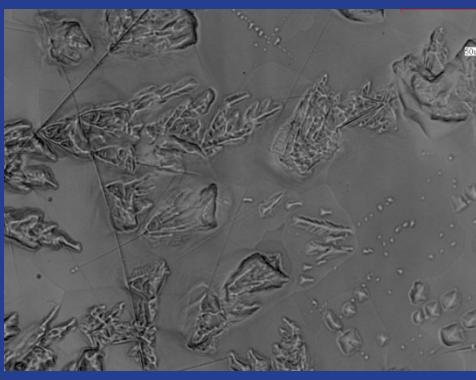
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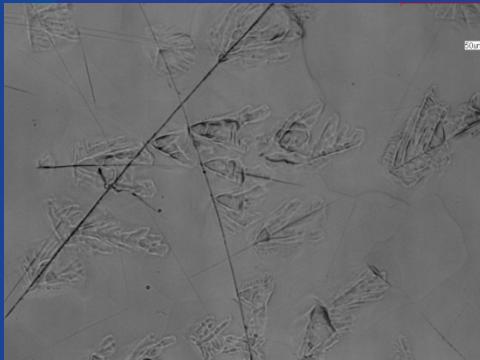
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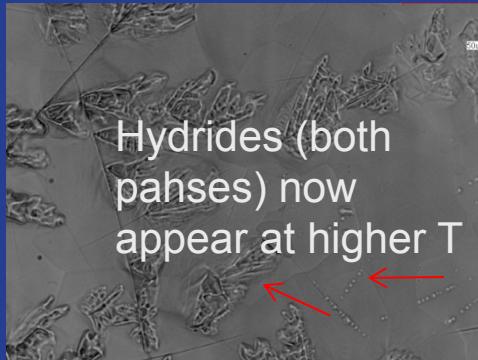
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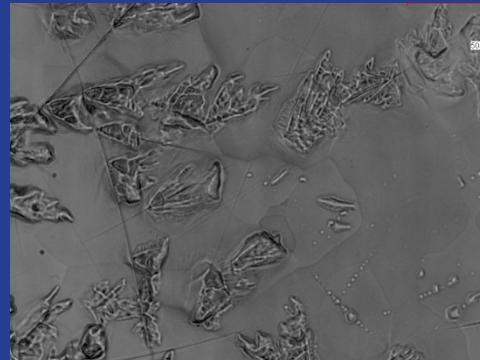
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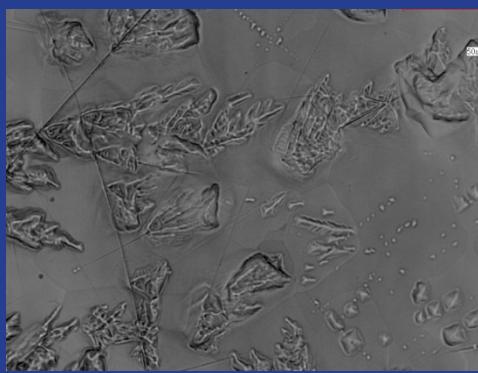
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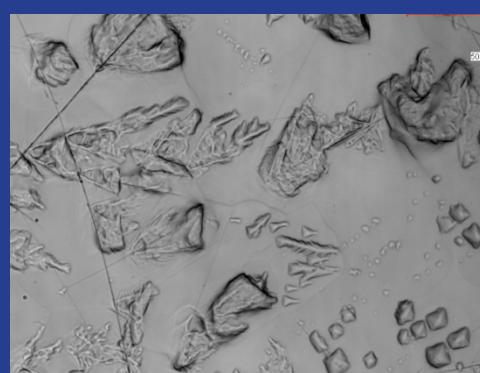
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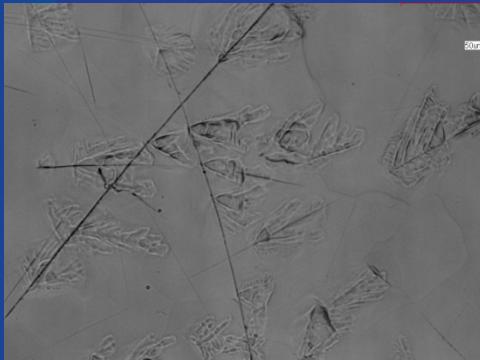
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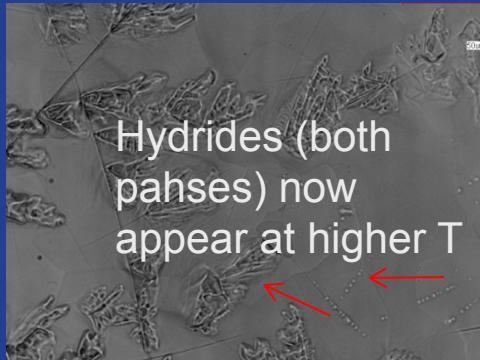
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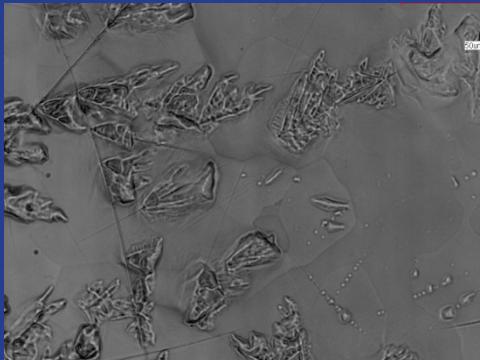
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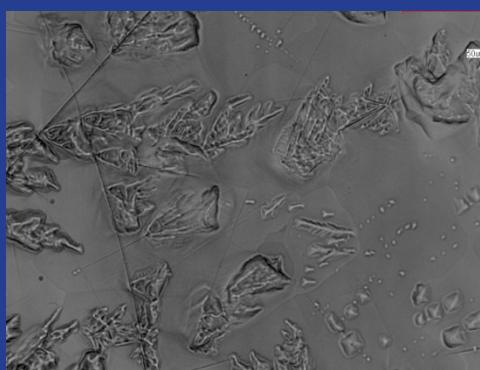


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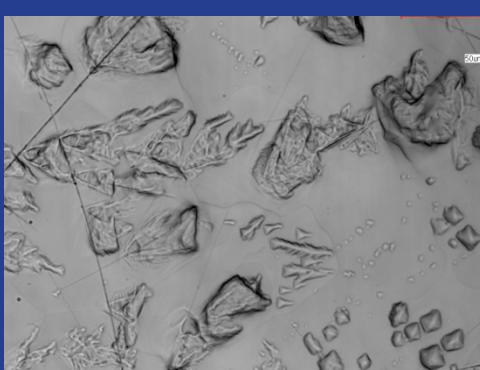


Second phase now
growing larger

160K



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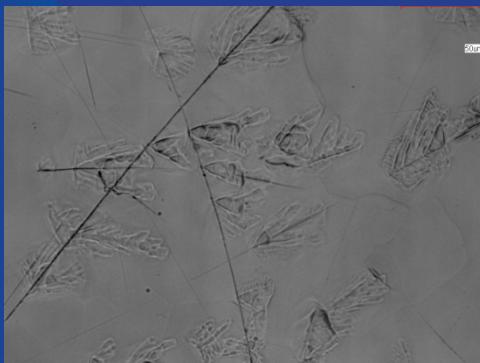


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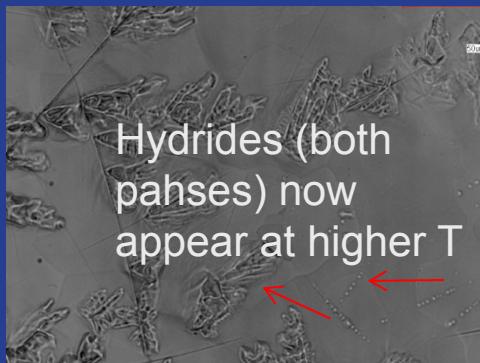


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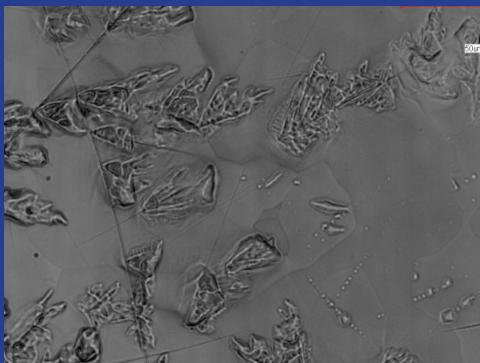
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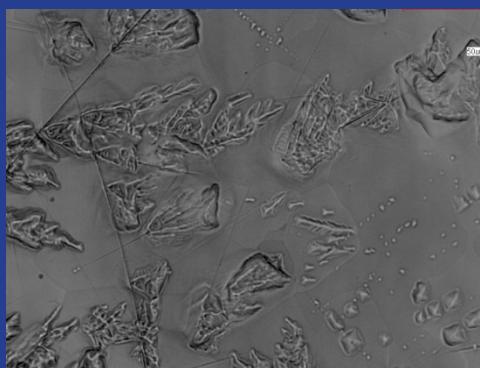
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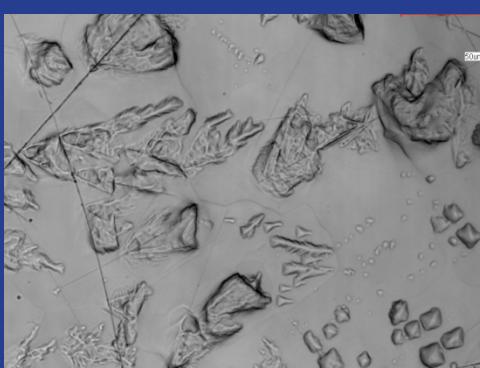
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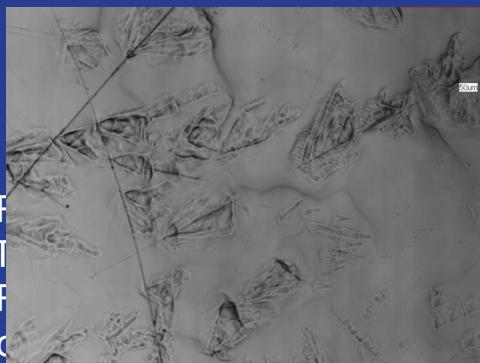
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ENERGY
5K

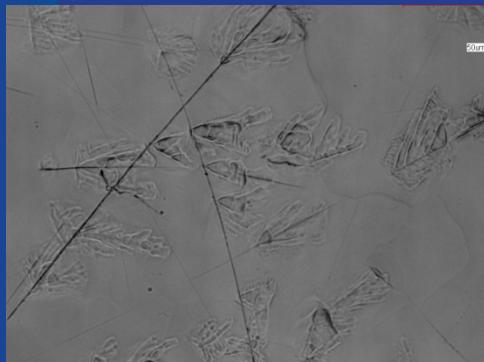


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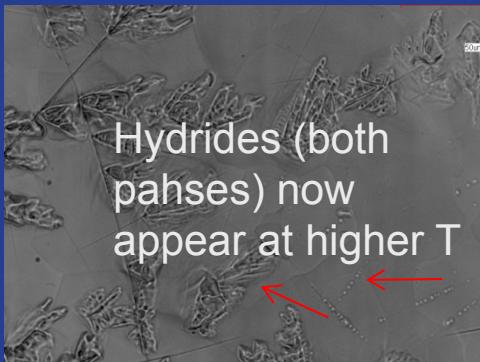


Fermilab

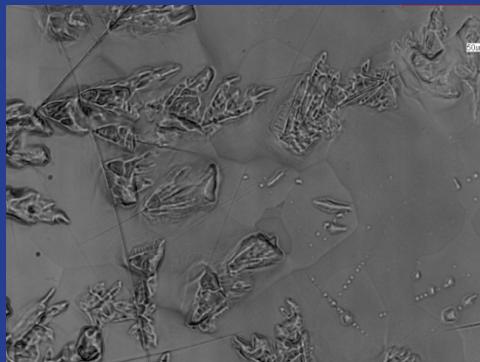
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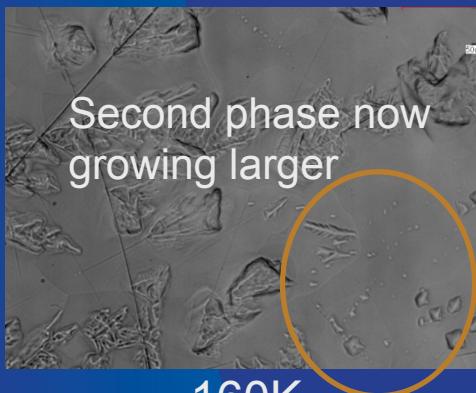
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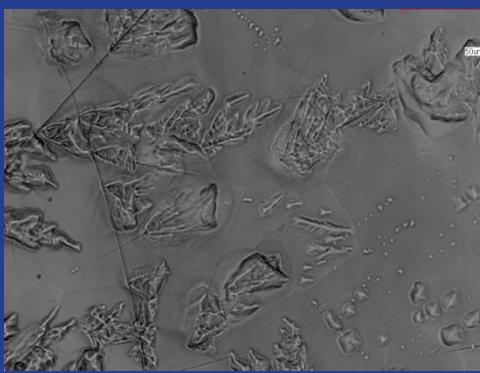
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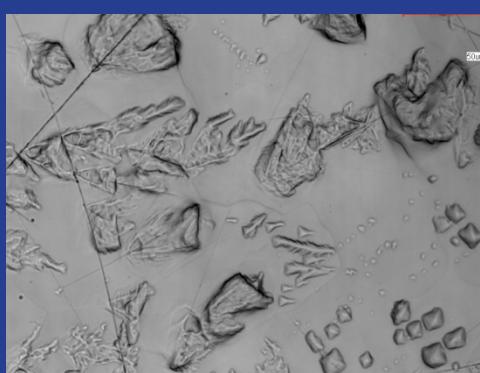
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ENERGY
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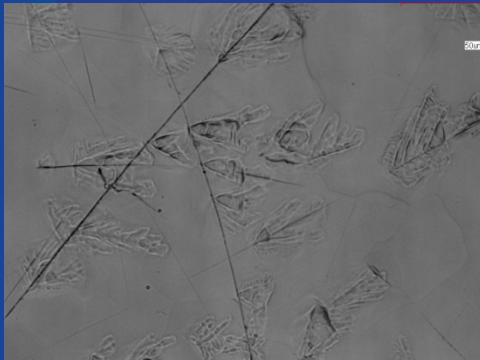


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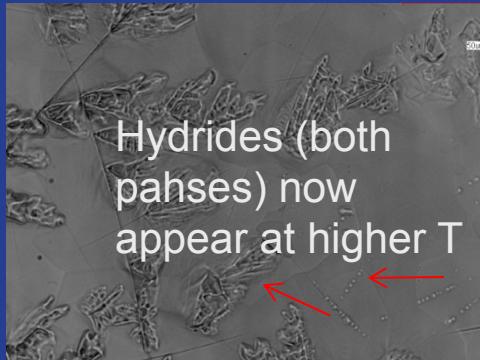
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 Fermilab

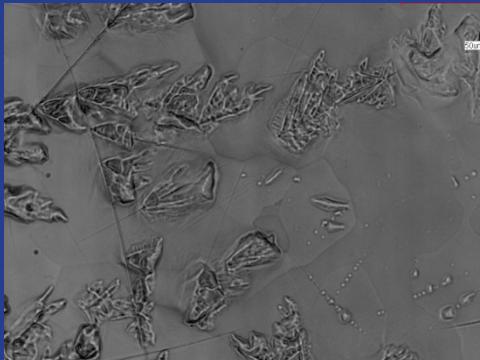
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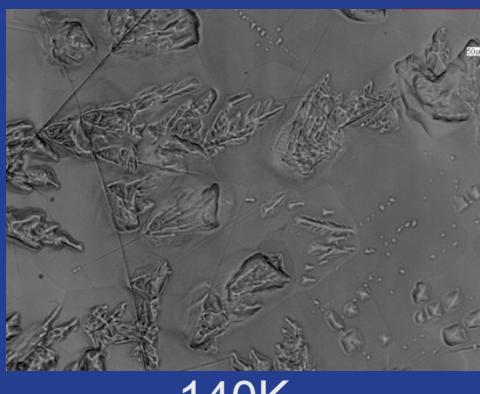
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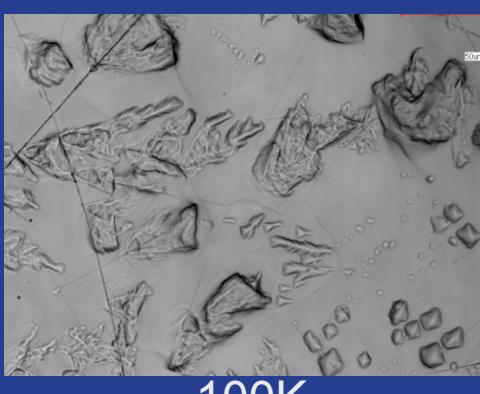
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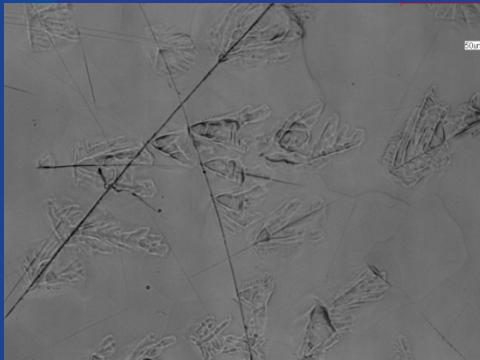
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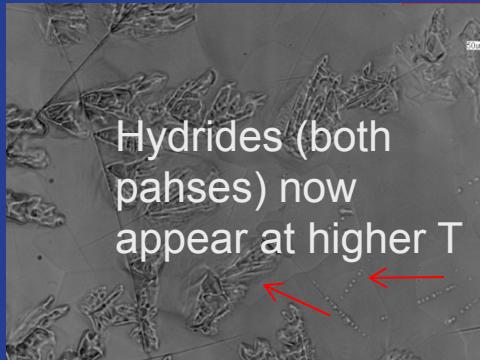
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Hydrides form much easier and at a higher temperature at both the "skeleton" left behind in the first cryocycle and at the locations of the second phase => cannot avoid Q-disease

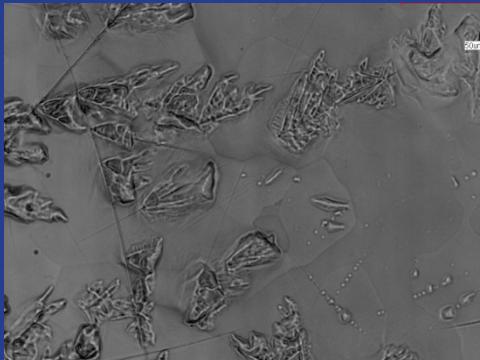
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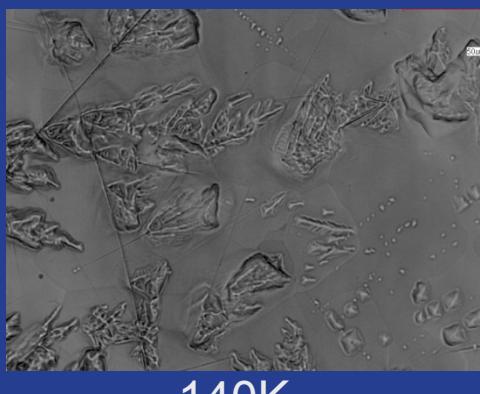
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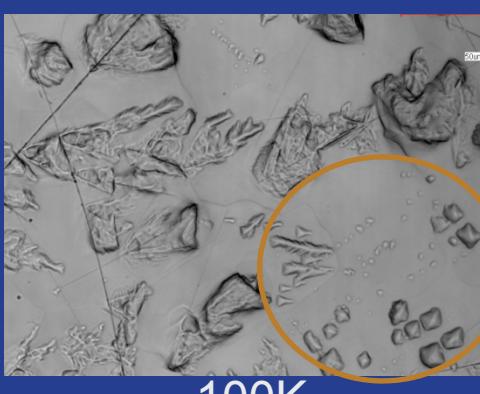
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ENERGY
5K



300K

Hydrides form much easier and at a higher temperature at both the "skeleton" left behind in the first cryocycle and at the locations of the second phase => cannot avoid Q-disease

Causes of RF losses: Trapped flux

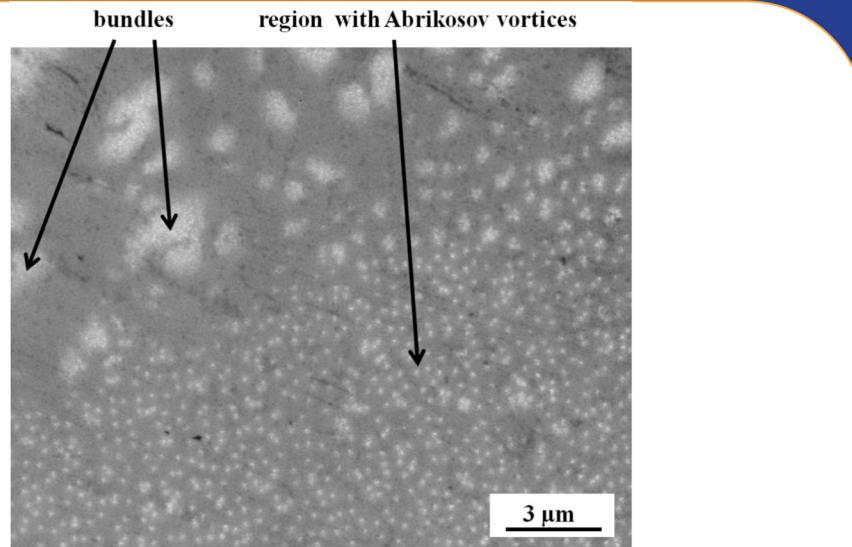
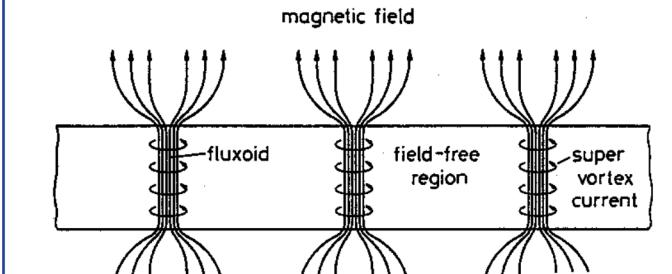
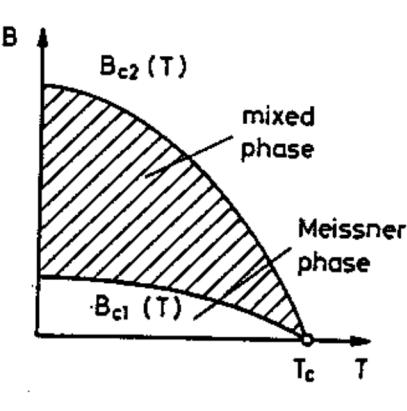


Figure 4: Decoration pattern for a flat sample at the magnetic field 8 mT reveals coexistence of regions with bundles and Abrikosov vortices.

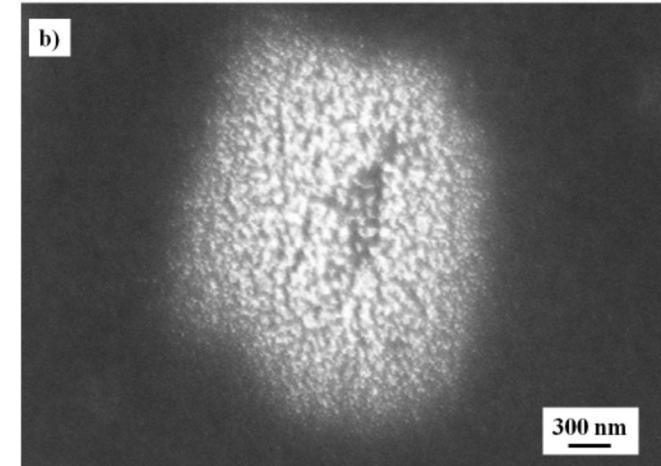
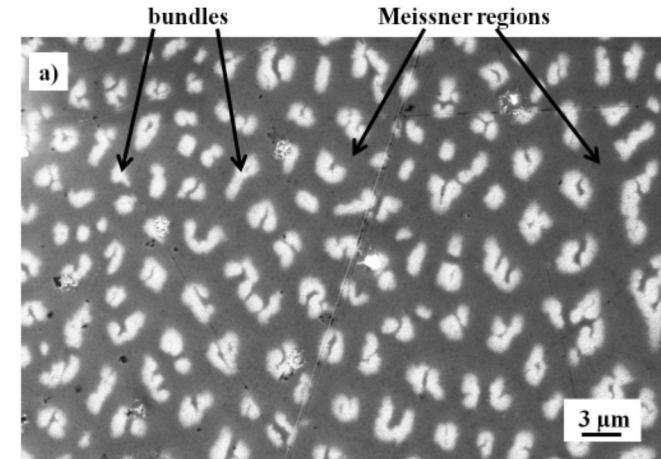
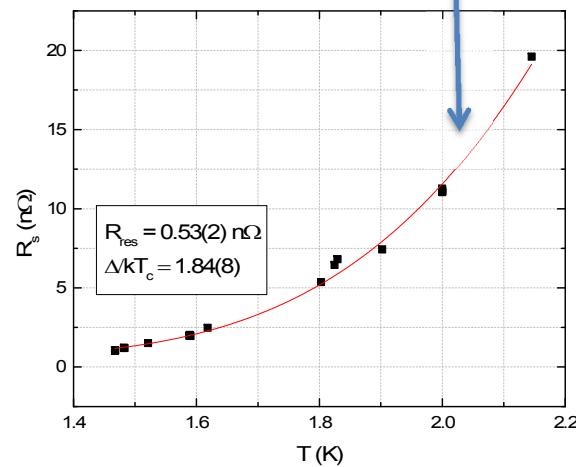
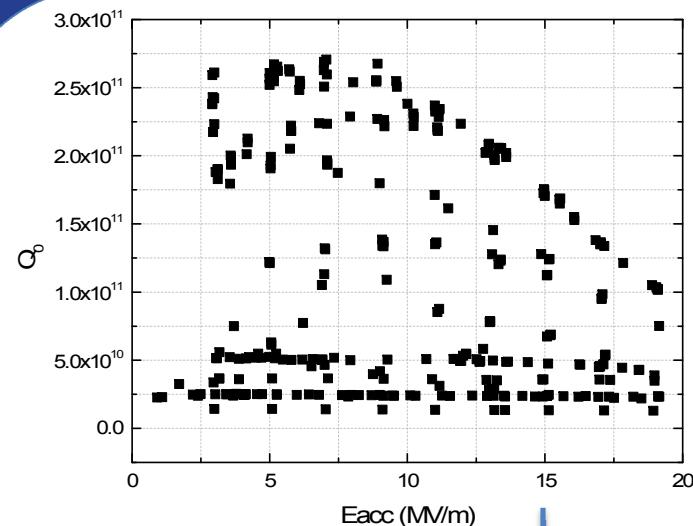


Figure 1: a) Decoration pattern at the magnetic field 10 mT reveals coexistence of “bundles” and Meissner regions. Each bundle carries 80 magnetic flux quanta in average; b) zoom-in of a single bundle.

Origin of RF losses: Field Dependence of Surface Resistance for typical treatments

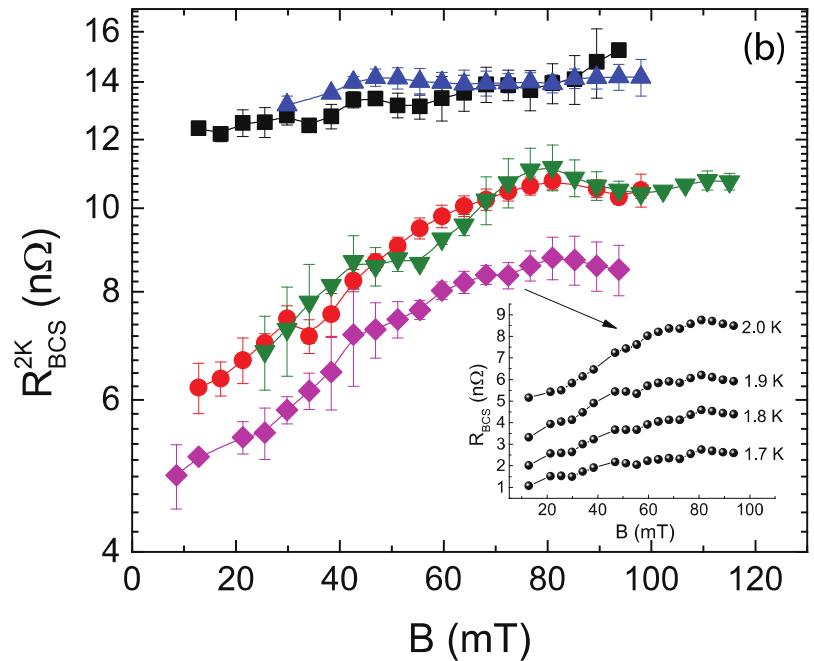
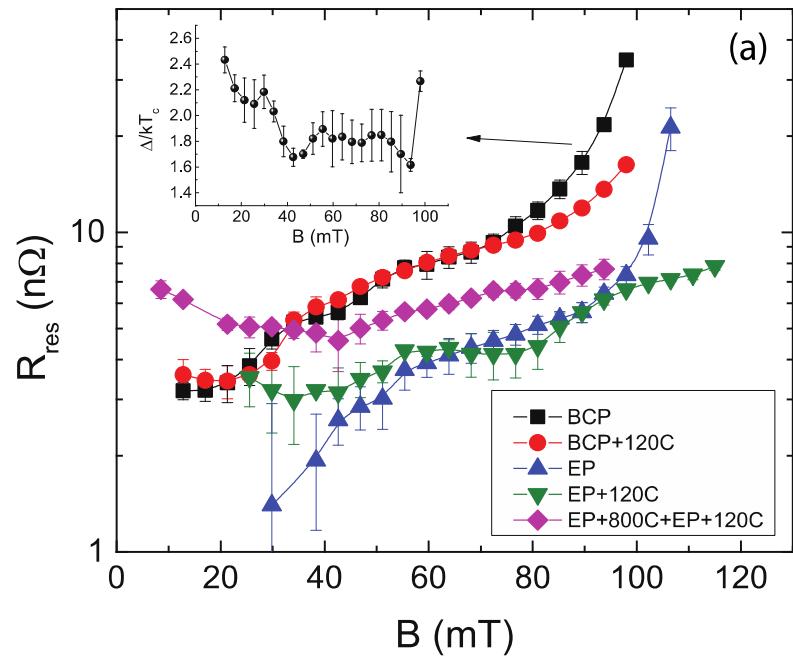
- Crucial question – how does *medium field Q-slope* emerge from its components $R_{BCS}(B)$ and $R_0(B)$?
- Answering allows:
 - Obtain $R_s(B,T)$ *predictions for any standard treatment* (EP, BCP, mild bake, anneal...) to design accelerators -> missing input for optimization
 - *Baseline for comparison* with new, innovative treatments
 - *Fundamental understanding* of “Q-slopes”

- Obtain as many $Q(B,T)$ measurements as practical at *ALL fields* (not only at a single low field as is customary)
- At each fixed field fit corresponding $Q(T)$ to extract R_{res}
 - Also gives $R_{\text{bcs}}(T) = R_s(T) - R_{\text{res}}$



Bath temperature

Results (1.3 GHz)



- Temperature dependent component uniquely determined by surface processing
- Both R_{BCS} and R_0 field dependence is determined by the surface processing
- R_{BCS} decreases but becomes *strongly field dependent after 120C*
- Medium field Q slope is *NOT due to thermal feedback*, stronger $R_0(B)$ for *BCP vs EP*

New surface processing techniques for high Q

Attack strategy based on fundamental understanding of causes of RF losses

Ideas to minimize BCS resistance:

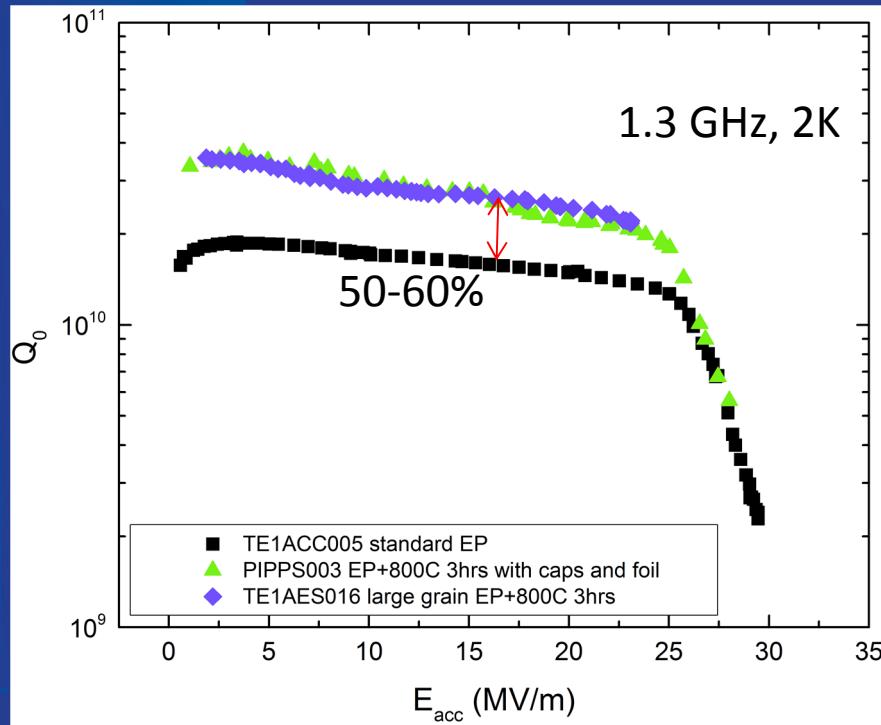
- Modify mfp, penetration depth → *doping* with interstitials
- Modify gap → eliminate sources of gap suppression (precipitates), create *higher T_c* compound (NbN, Nb₃Sn...)

Ideas to minimize residual resistance:

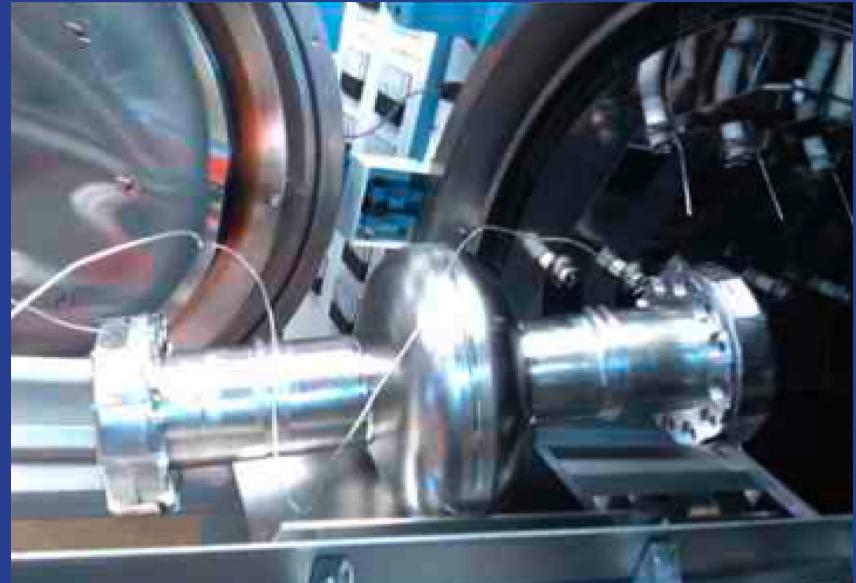
- Minimize trapped flux → eliminate *trapping centers*
- Avoid precipitates (e.g. hydrides) → minimize *nucleation centers*, minimize *hydrogen reabsorbed*, neutralize hydrogen via doping

See talk by Sam Posen!

Annealing as a last processing step

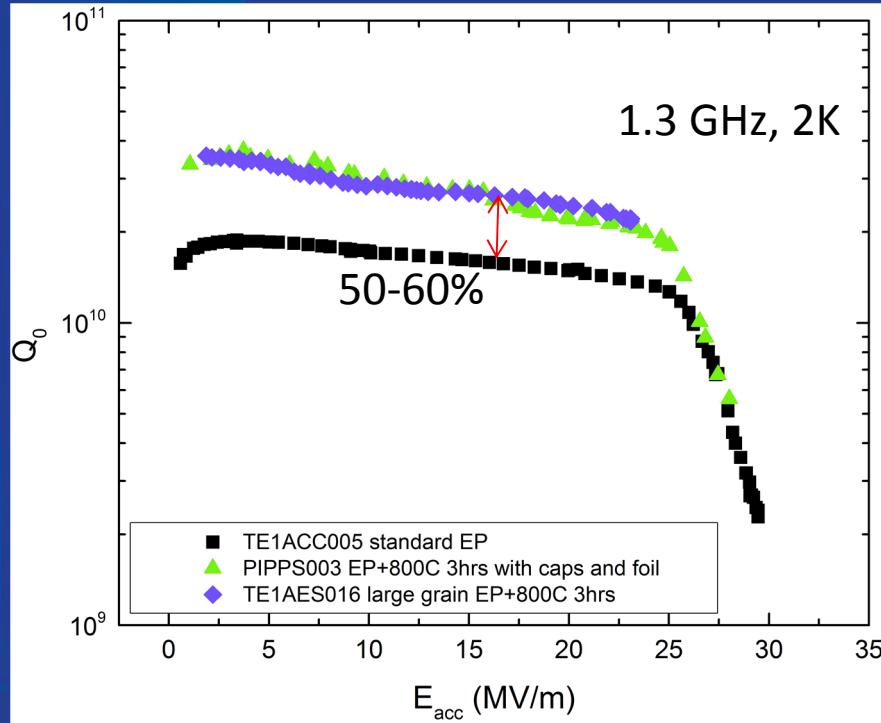


A.Grassellino et al, <http://arxiv.org/abs/1305.2182>

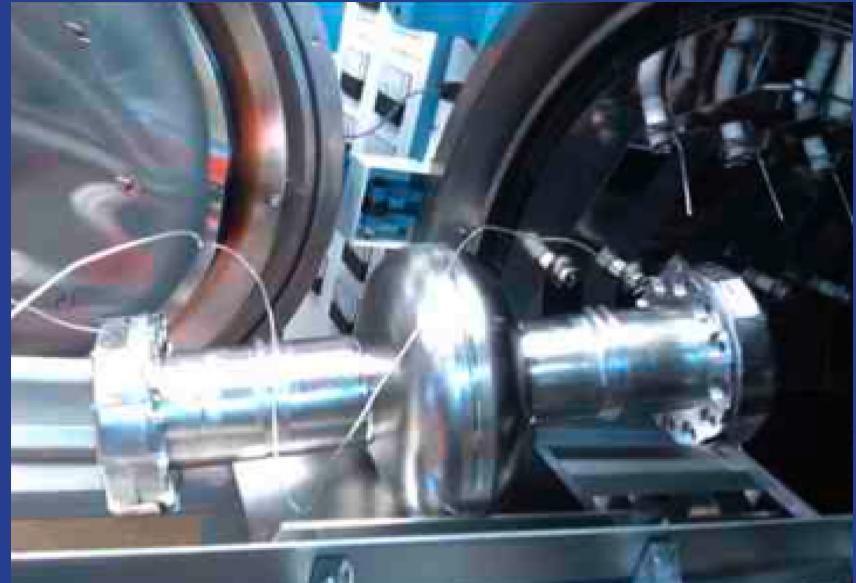


- *EP + 800C 2 hrs + 20-40 micron EP + 120C*
- *Systematically low $R_0 \sim 1n\Omega$*
- *Extra cost savings from skipping the post furnace chemical processing*

Annealing as a last processing step

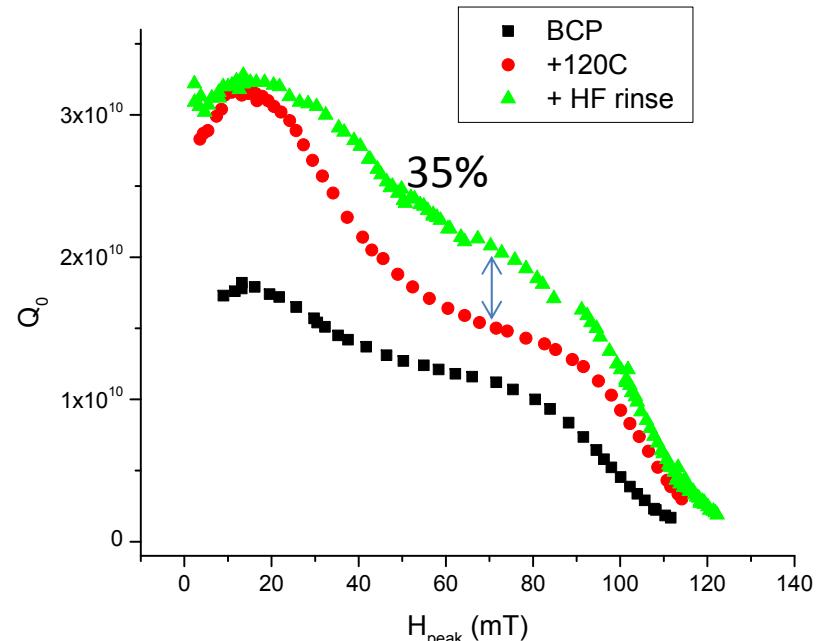
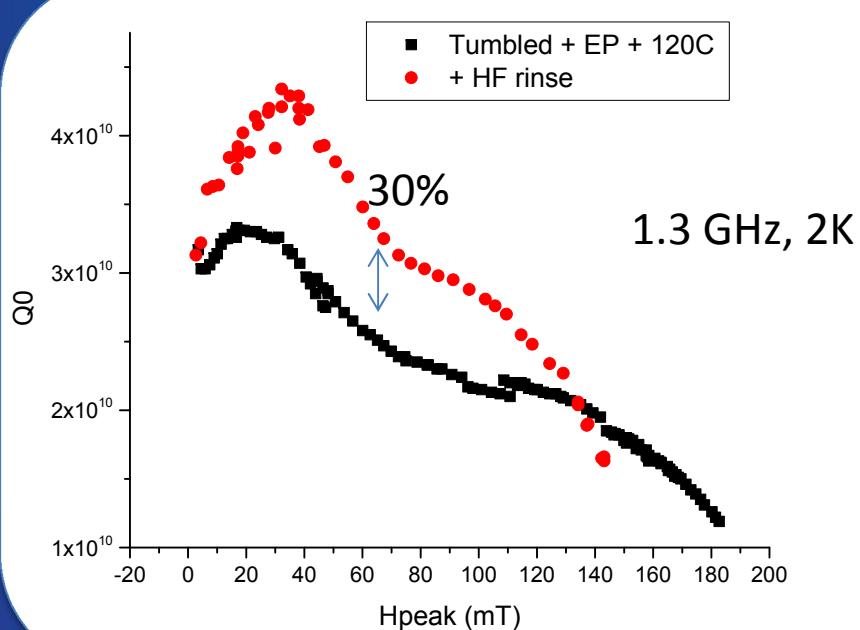


A.Grassellino et al, <http://arxiv.org/abs/1305.2182>



- $EP + 800C$ 2 hrs + 20 ~~micron~~ EP + ~~800C~~ \rightarrow higher Q
- Systematically low $R_0 \sim 1n\Omega$
- Extra cost savings from skipping the post furnace chemical processing

Simple higher Q_0 recipe: 120C bake +1 HF rinse

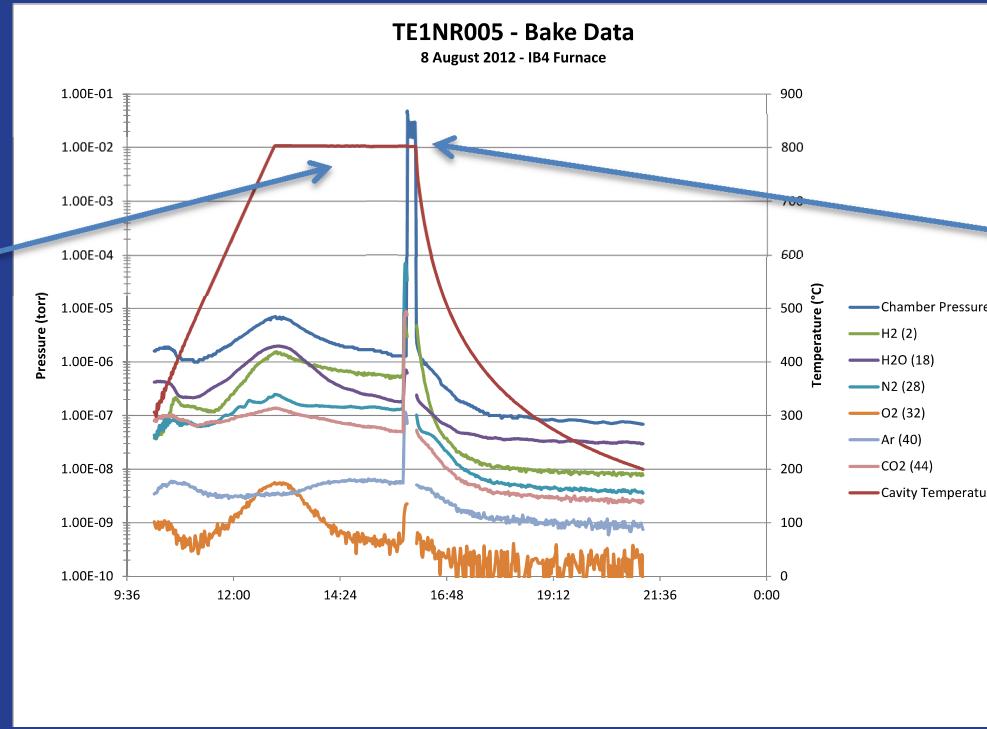


- Single HF rinse (5 min) followed by water rinse is beneficial for the medium field Q value – gains of up to 35% measured at 70 mT, 2K, up to 100% at 1.8K
- Cornell already implemented successfully on several multicell cavities

A. Romanenko et al, Phys.Rev.ST Accel.Beam 16 (2013) 012001

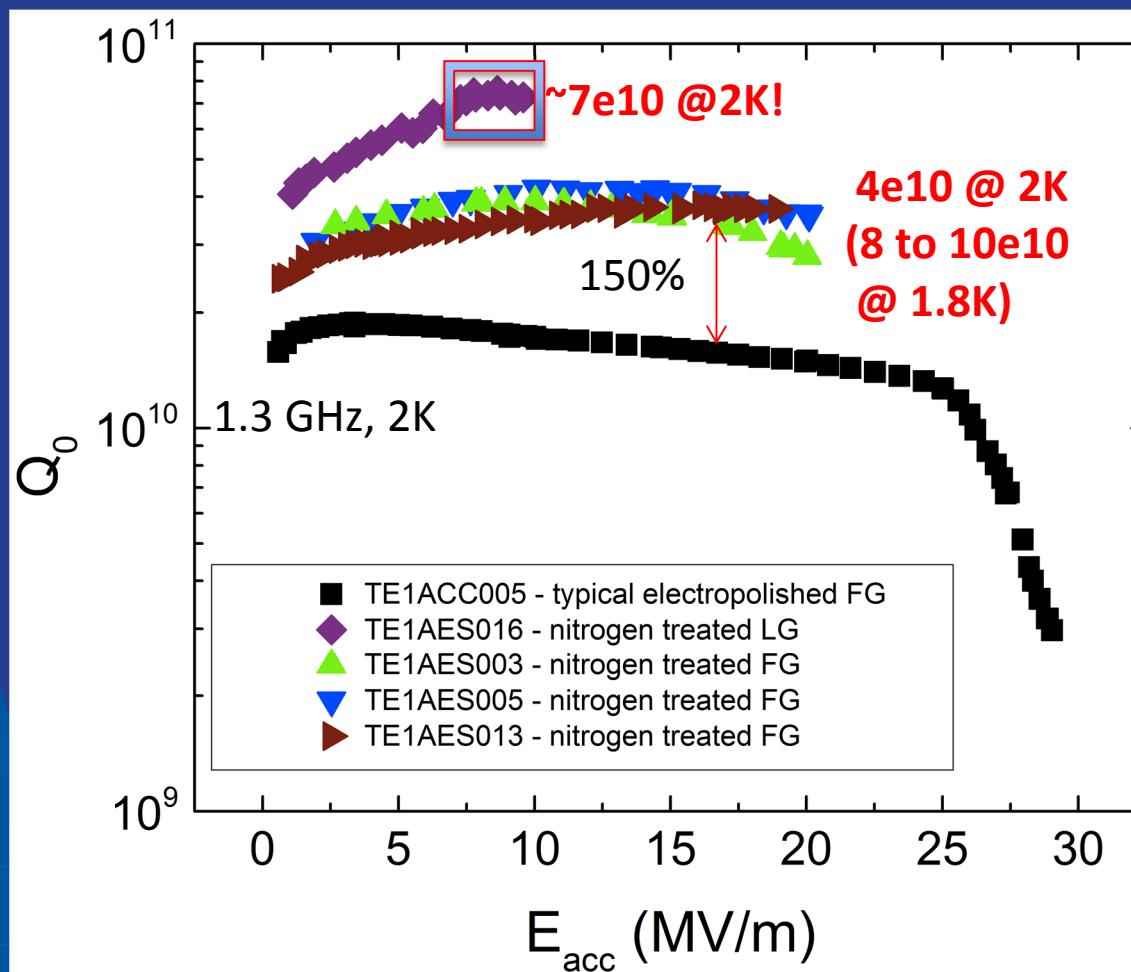
NEW- FNAL: High T bake in nitrogen gas

Standard 800C degassing cycle



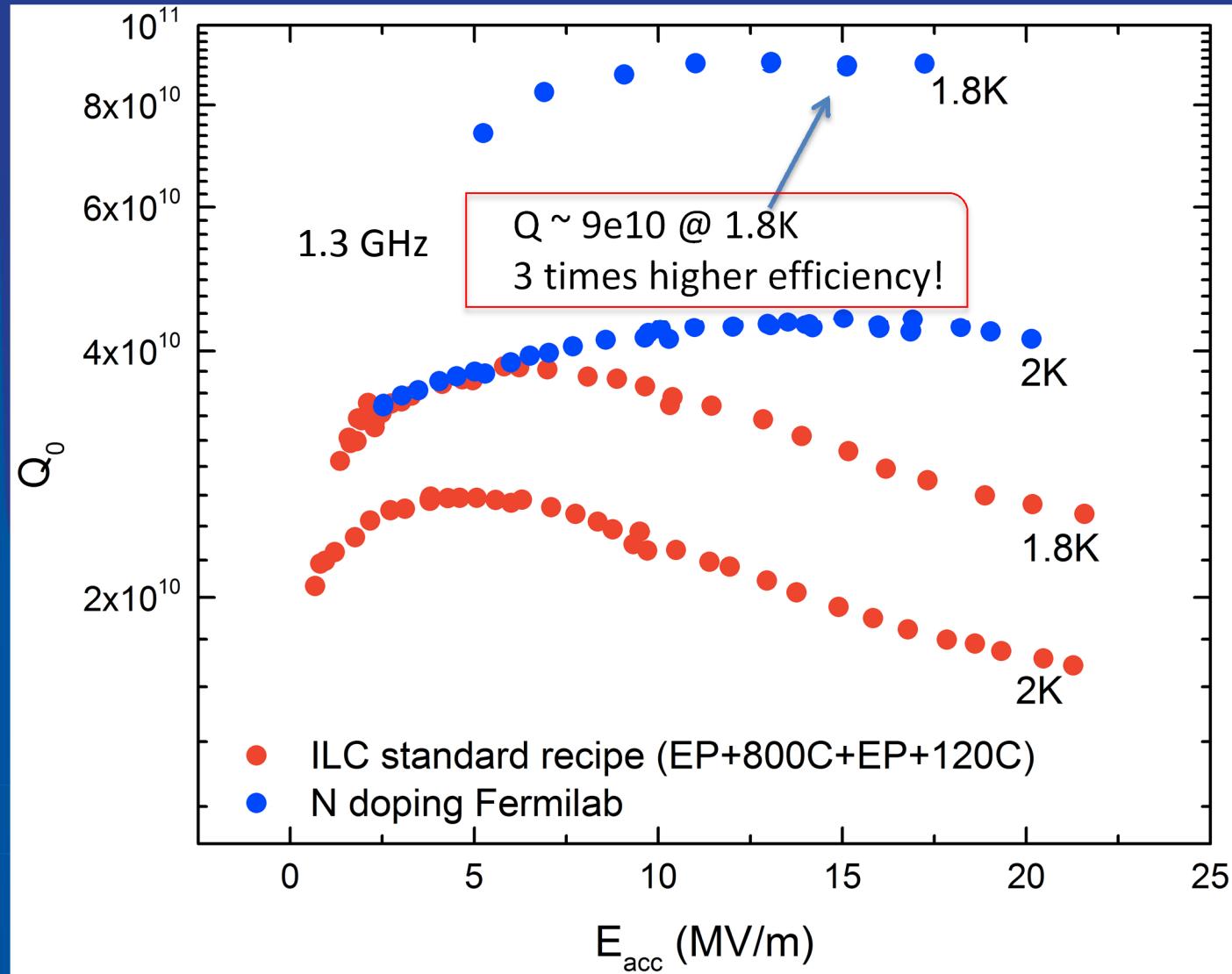
- Several cavities treated with nitrogen at different T: 600C, 800C and 1000C for different duration
- Q all extremely poor after treatment $\sim 10^7$ - 10^9
- Then, we removed a certain amount of material via electropolishing

Record Quality Factors obtained via impurity doping - beyond the expected BCS limit for Nb

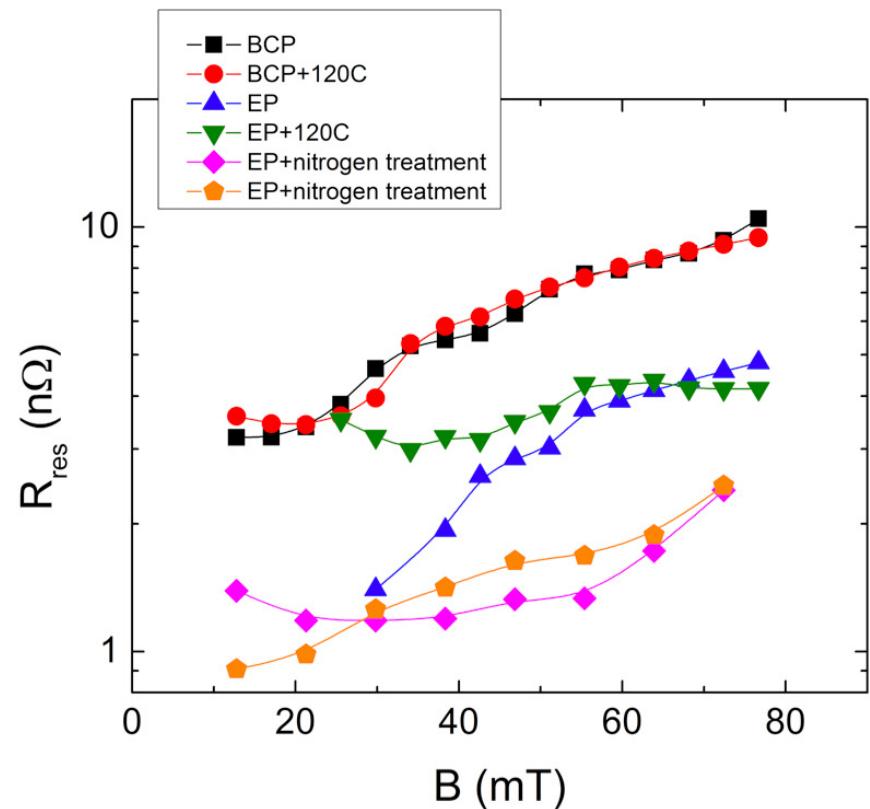
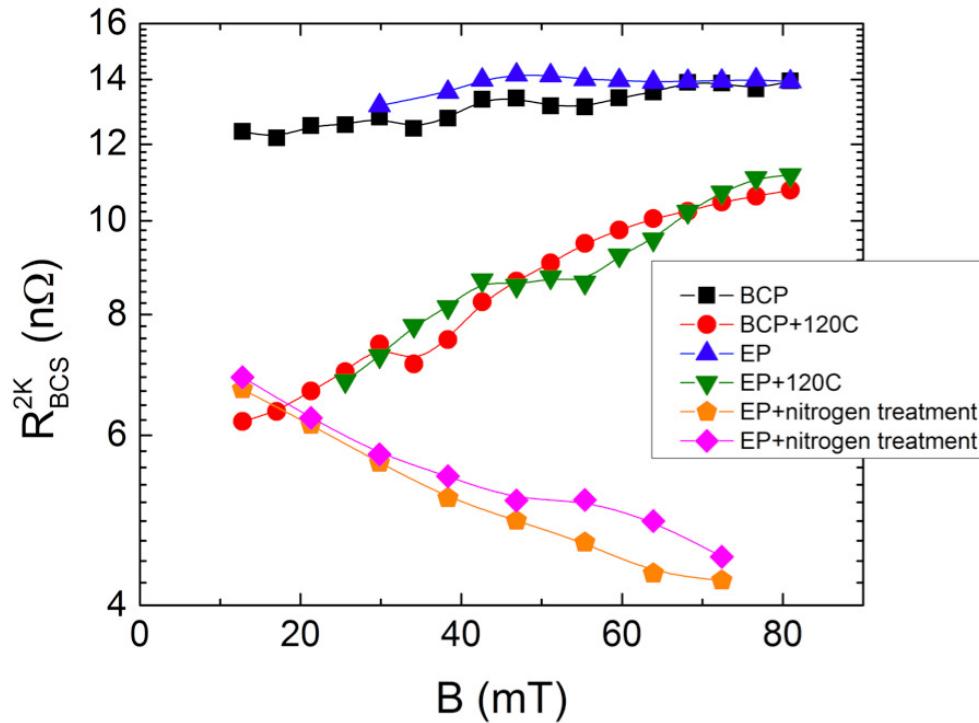


A.Grassellino et al, 2013 *Supercond. Sci. Technol.* 26 102001

Comparing nitrogen treated to standard ILC processing at 2 and 1.8K



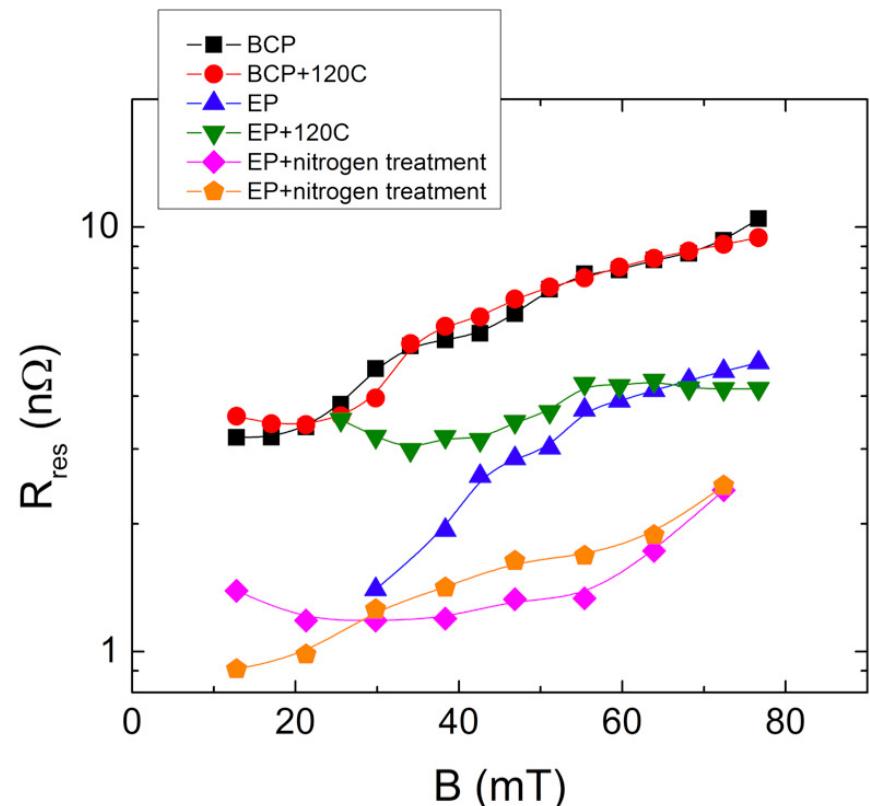
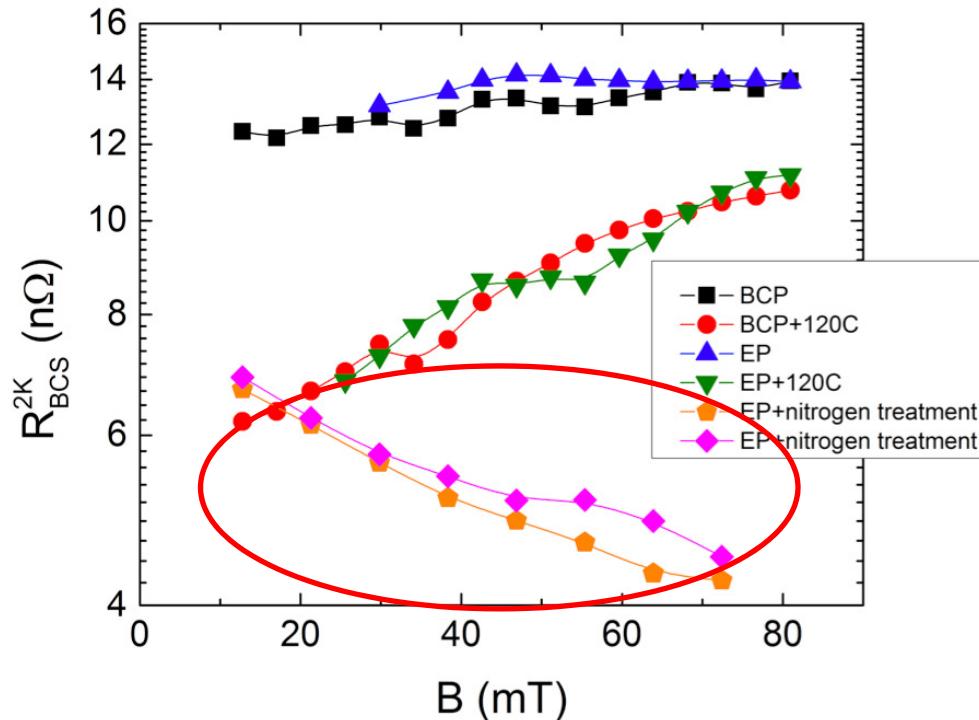
Where does the improvement originate? The reversal of $R_{BCS}(B)$



Previously unseen phenomenon!

A.Grassellino et al, 2013 *Supercond. Sci. Technol.* 26 102001

Where does the improvement originate? The reversal of $R_{BCS}(B)$



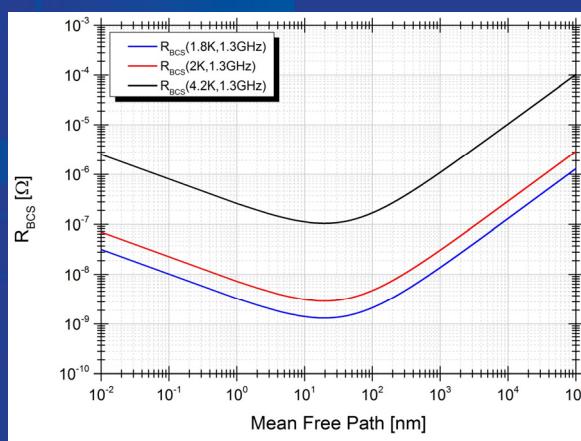
Previously unseen phenomenon!

A.Grassellino et al, 2013 *Supercond. Sci. Technol.* 26 102001

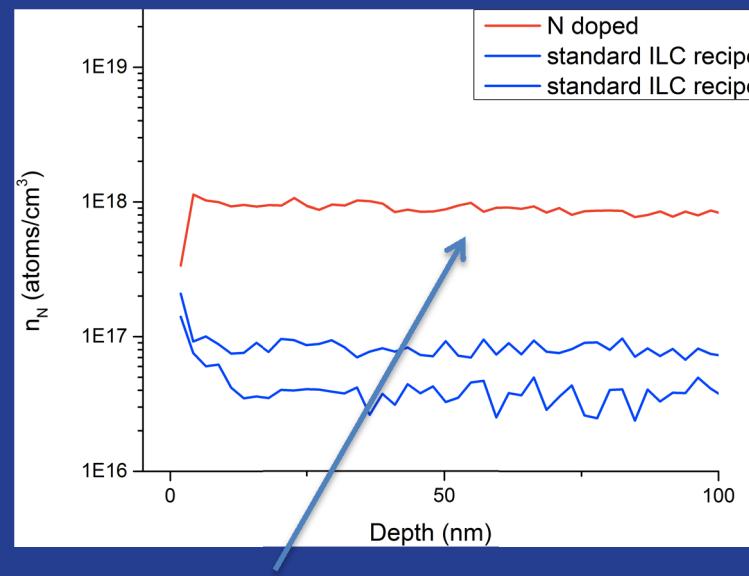
So far, our understanding of it

-BCS is lowered with lower mean free path:

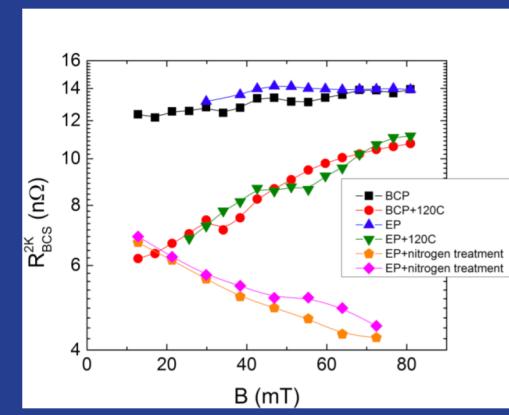
- EP, BCP → clean limit, high BCS, little to no field dependence
- In the 120C bake case mfp near surface ~ 2nm (→ lower BCS at low field, BUT dirty limit, field dependence of the gap causes slope)
- Nitrogen/Argon treated: intermediate purity! Near surface ~ 40 nm → minimum of BCS at low field, but reverse field dependence unclear, maybe intrinsic?



M. Checchin, calculated in two fluid model approx



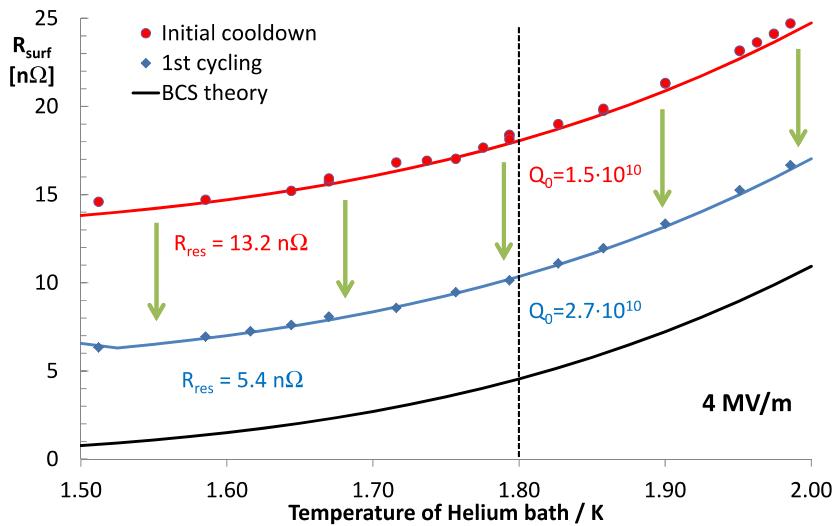
SIMS results showing 10 times higher than typical nitrogen concentration



Q ‘preservation’ in a real machine

HZB studies of trapped flux due to thermocurrents

Influence of thermal cycling on R_{surf}

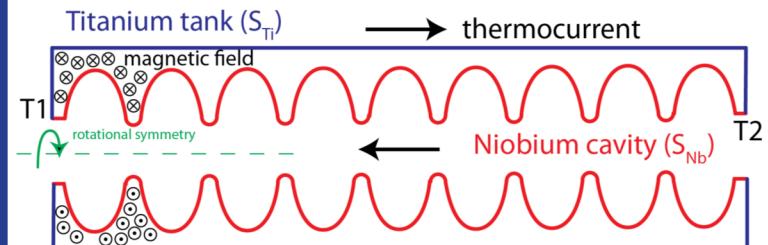


Kugeler et al, talk presented at SRF 2013



Thermocurrents

- Cavity forms thermoelement
- Different Seebeck coefficients for Nb and Ti

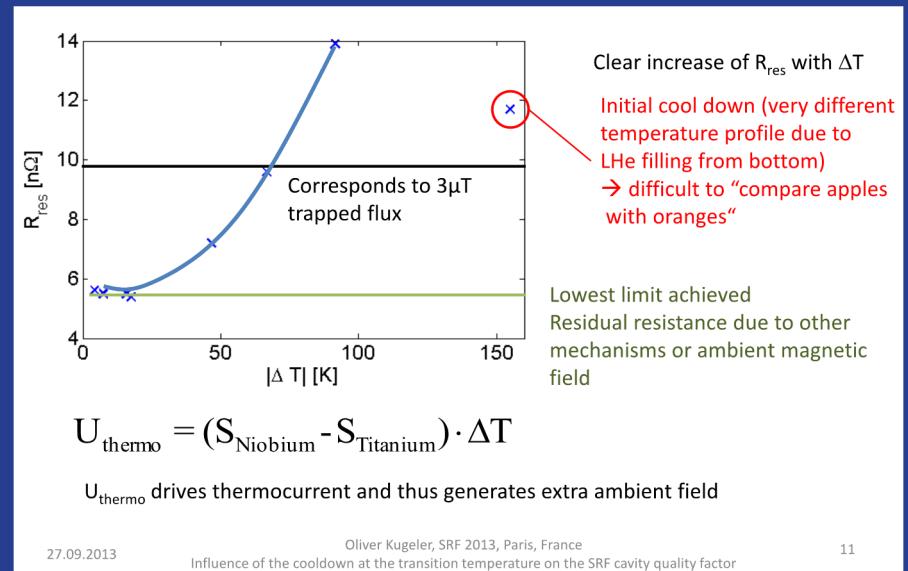


$$U_{\text{thermo}} = (S_{\text{Niobium}} - S_{\text{Titanium}}) \cdot \Delta T$$

27.09.2013

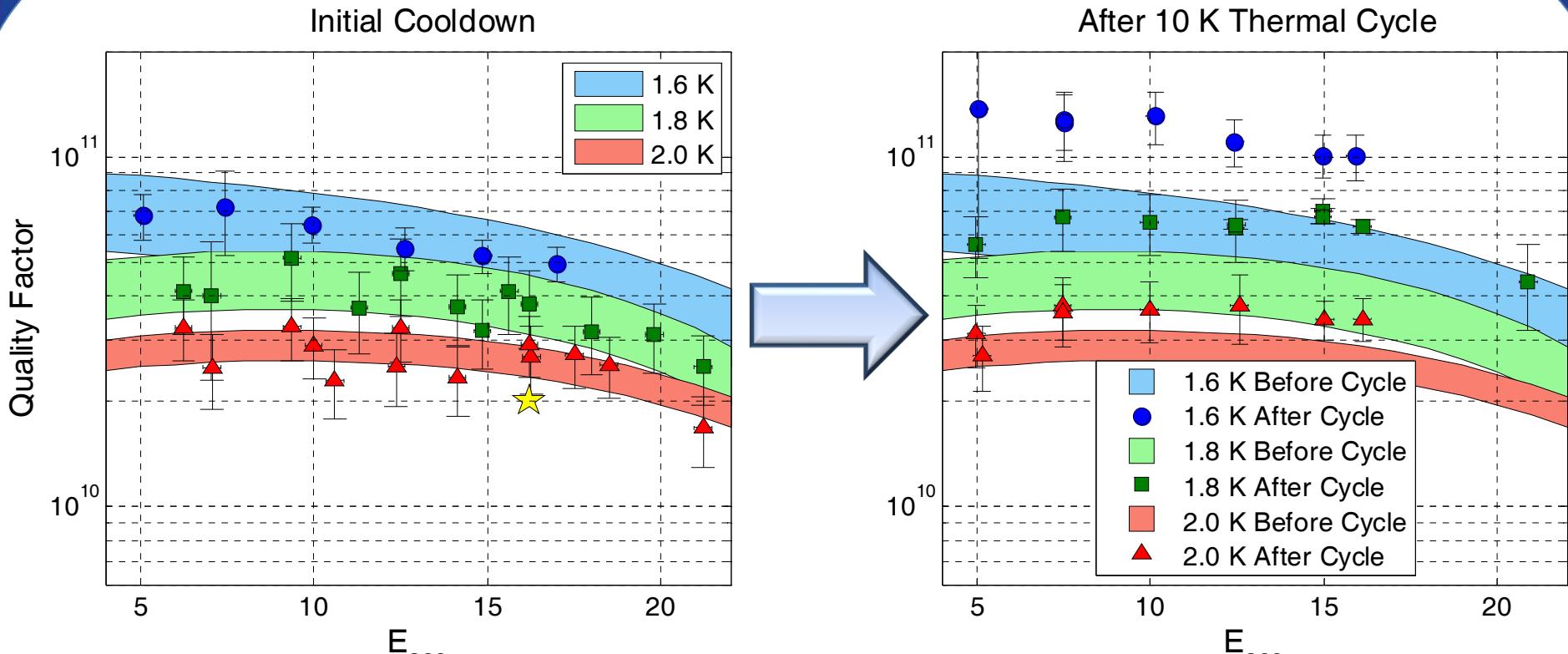
Oliver Kugeler, SRF 2013, Paris, France
Influence of the cooldown at the transition temperature on the SRF cavity quality factor

8



Cornell studies of minimization of trapped flux in cryomodule

N. Valles – High Q Cavity Operation in the Cornell HTC – TTC Topical Meeting on CW-SRF 2013



Initial Cooldown at 16.2 MV/m

$$Q(2.0\text{ K}) = 2.5 \times 10^{10}$$

$$Q(1.8\text{ K}) = 3.5 \times 10^{10}$$

$$Q(1.6\text{ K}) = 5.0 \times 10^{10}$$

10 K thermal cycle at 16.2 MV/m

$$Q(2.0\text{ K}) = 3.5 \times 10^{10}$$

$$Q(1.8\text{ K}) = 6.0 \times 10^{10}$$

$$Q(1.6\text{ K}) = 10.0 \times 10^{10}$$

Conclusions

- Tremendous progress in understanding and finding of new cures in the last two years, since high Q effort started
- The doping with interstitials discovery shows that Nb is NOT at the end of the road! And that new exciting horizons are open for the SRF technology!
- R&D largely pays off, with relatively small effort we have already produced results which translate into huge savings for SRF based accelerators planned worldwide

The FNAL work presented comes from a team effort:

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B.Stone, C.Baker, Y. Pischalnikov

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